

Transient Liquid Assisted Growth (TLAG), a method for increasing coated conductors throughput and meeting future

Teresa Puig¹

R. Vlad¹, L. Saltarelli¹, D. Garcia^{1,2}, K. Gupta¹, C. Torres¹, A. Kethamkuzhi¹, E. Pach^{1,5}, C. Pop¹,
D. Sanchez³, S. Rasi¹, J. Banchewski¹, A. Queralto¹, S. Ricart¹, R. Yanez², J. Farjas³,
J. Gutierrez¹, C. Mocuta⁴, E. Solano⁵, X. Obradors¹

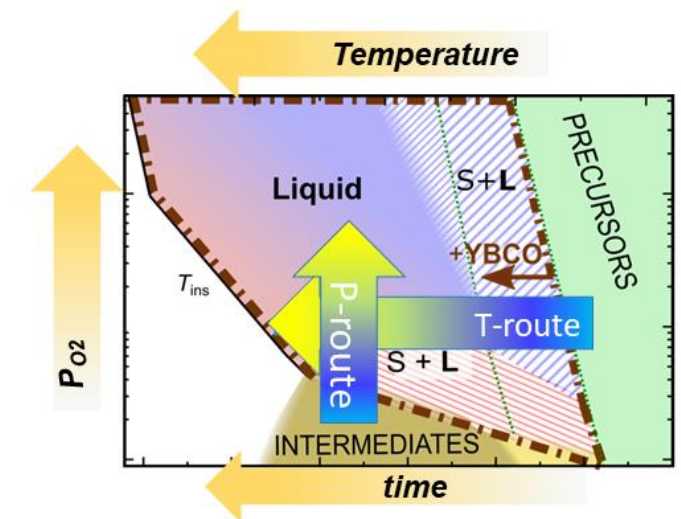
¹ Institut de Ciència de Materials de Barcelona, ICMAB-CSIC, Spain

² Departament de Química, Universitat Autònoma Barcelona, Spain

³ GRMT, Department of Physics, University of Girona, Spain

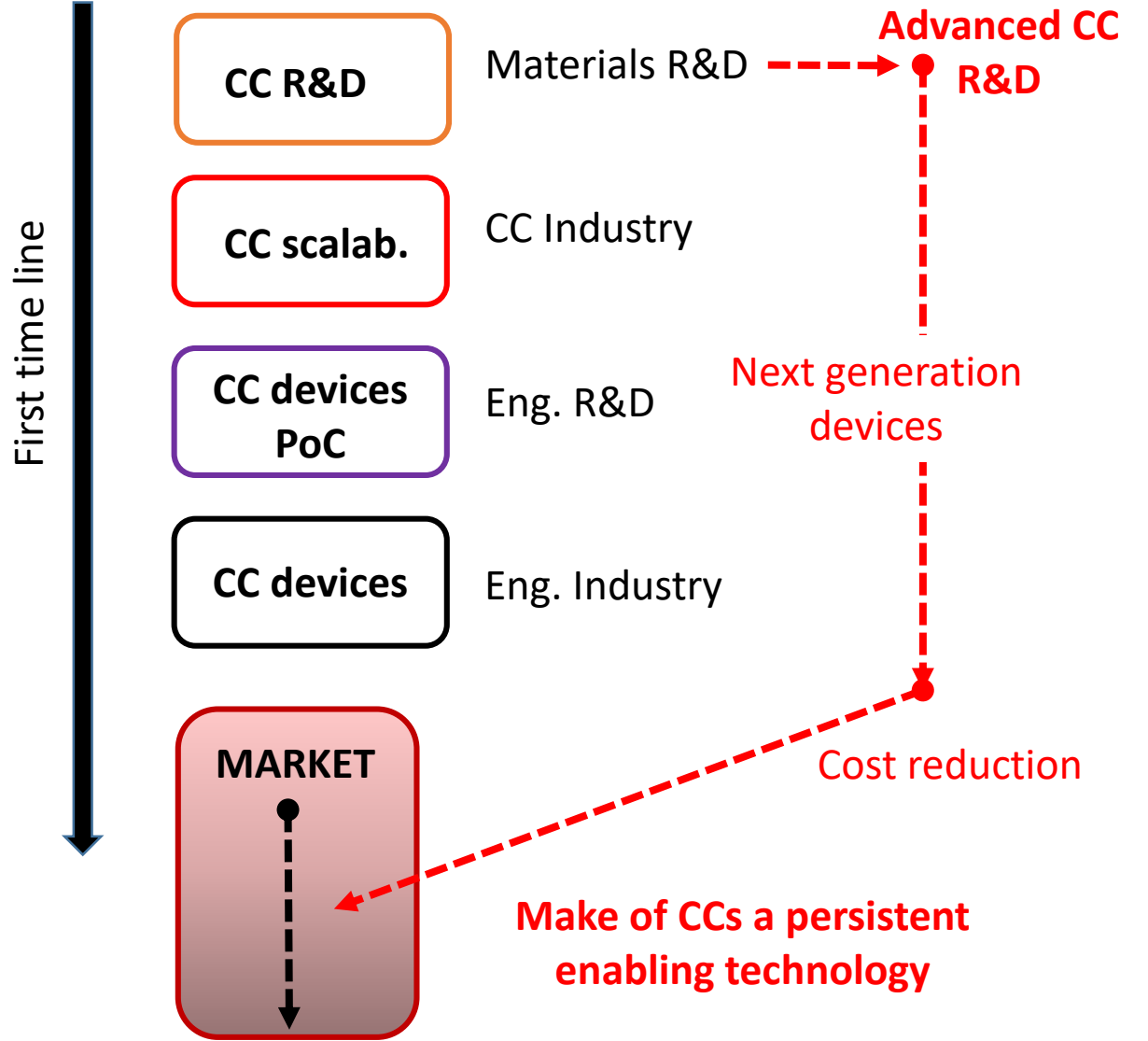
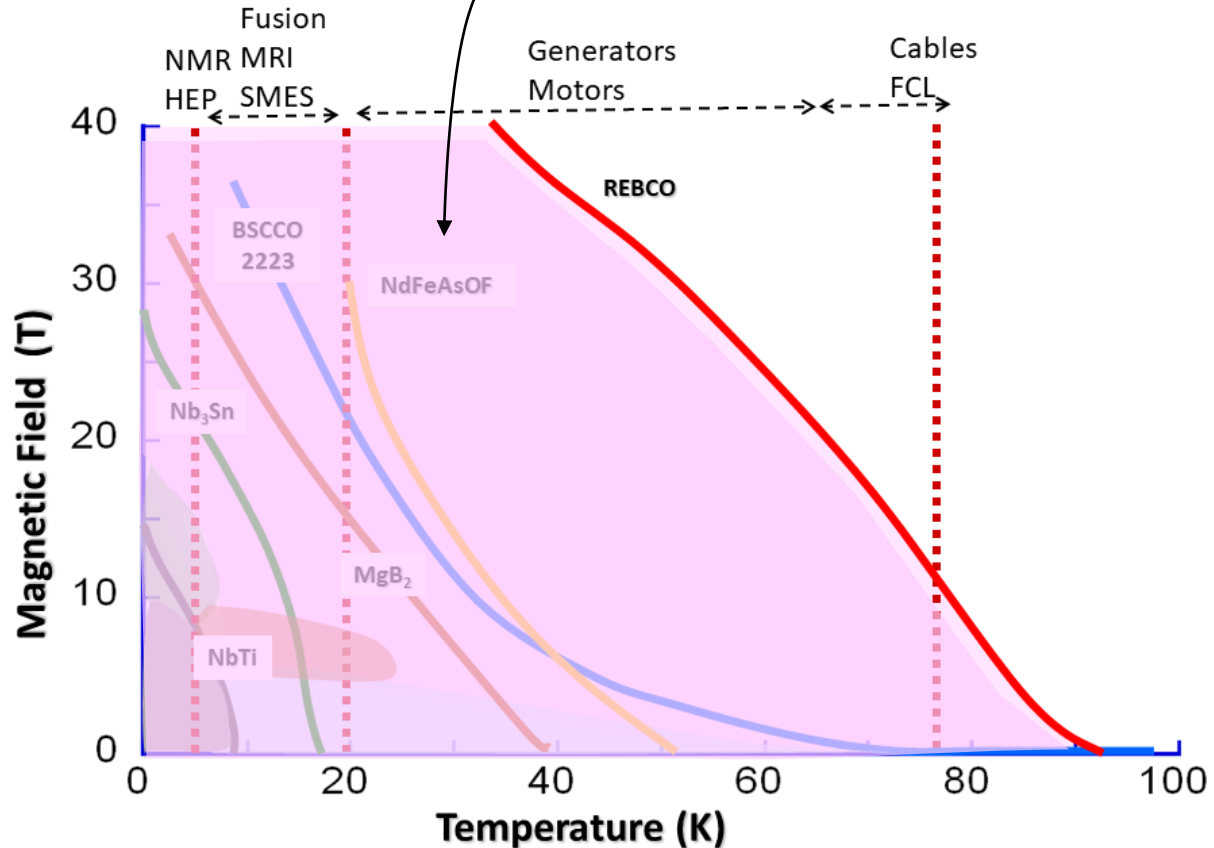
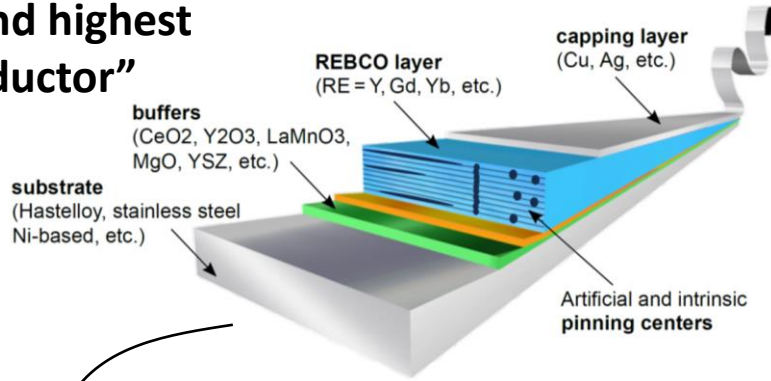
⁴ Diffabs beamline, Soleil Synchrotron, Paris, France

⁵ NCD-Sweet beamline, ALBA Synchrotron, Barcelona, Spain



REBCO COATED CONDUCTORS

“The highest Field and highest Temperature conductor”

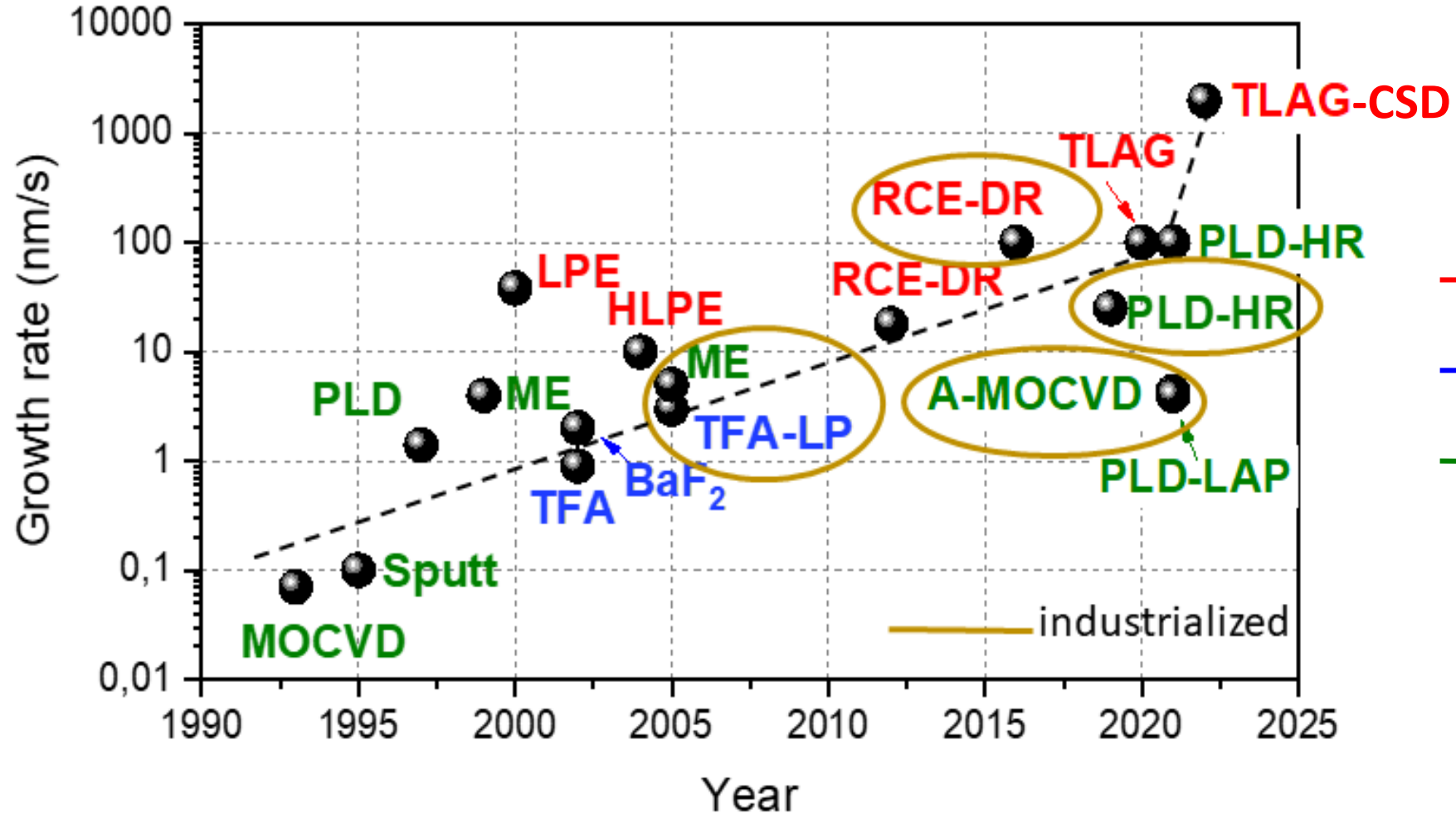


Make of CCs a persistent enabling technology

Reaching high Growth Rate: A path towards cost reduction

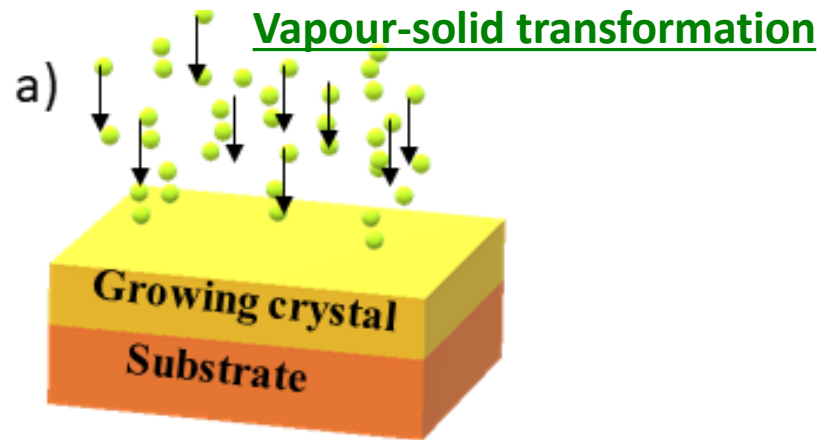
Figure of merit:
$$\frac{\text{Cost}}{\text{Performance}} = \frac{\text{total cost per year}}{G \times L \times W \times (I_{c-w}/d)} = \frac{\text{€}}{kA \times m}$$

G = growth rate
 W = tape width
 L = tape length
 d = tape thickness



REBCO growth processing

Supersaturation, σ , is the driving force for crystallization: $\sigma \propto G$ (growth rate)

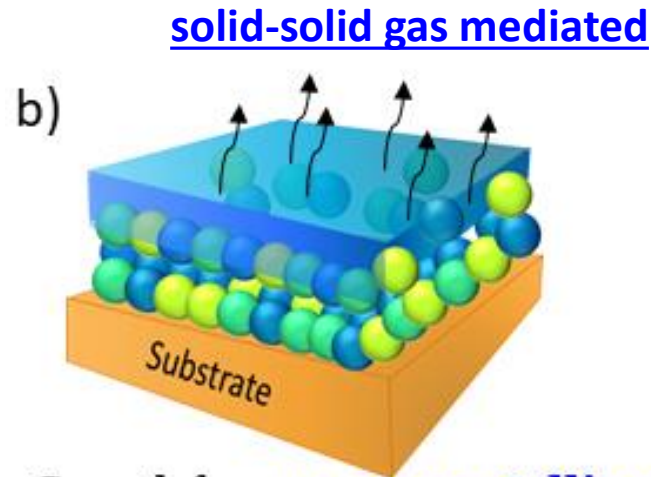


Growth from **vapour phase**
 PLD, MOCVD, ME, MBE, Sputt

$\sigma = (P_{ad} - P_{ad,e}) / P_{ad,e}$ Deposition rate
 High vacuum environ.

$P_{ad,e}$ = ad-atoms equilibrium pressure at surface growth front
 P_{ad} = ad-atoms pressure at surface growth front

Growth rate: $G = 0.5-25$ nm/s

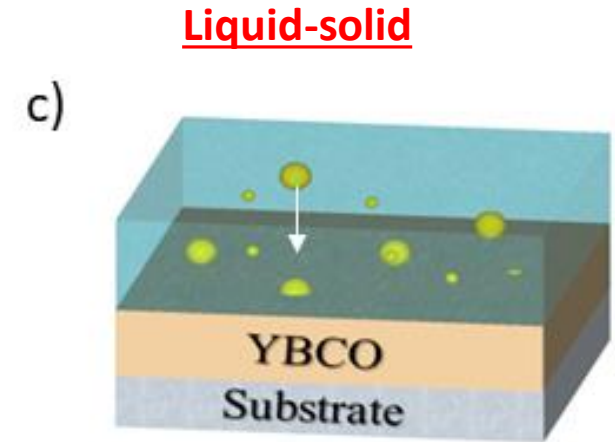


Growth from **nanocrystalline solids**
 TFA₇-MOD, BaF₂

$\sigma = f(\ln(P_{HF}^2 / P_{H2O}))$

P_{HF} = HF partial pressure
 P_{H2O} = water partial pressure

$G = 0.5-5$ nm/s



Growth from **liquid phase**
 TLAG-CSD, RCE-DR, HLPE, VLS

$\sigma = (C_{\delta} - C_e) / C_e$ RE solubility,
 Ba-Cu-O liquid

C_e = RE equilibrium concentration in the liquid
 C_{δ} = RE actual concentration

$G = 10-1000$ nm/s



Transient Liquid Assisted Growth (TLAG)



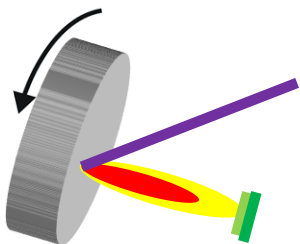
A new high throughput non-equilibrium kinetically controlled growth process



CSD

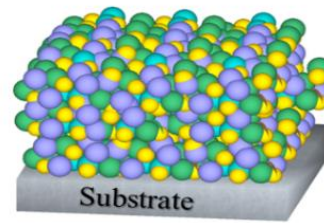
L. Saltarelli et al, ACS Appl. Mat. & Interf. (2022)

Patent EP22382741

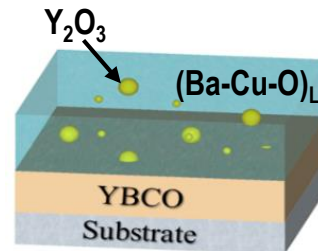


Low Temp PLD

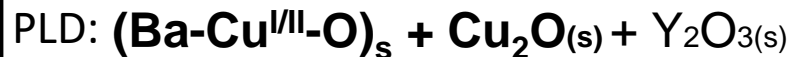
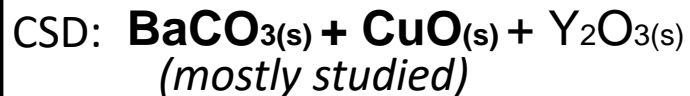
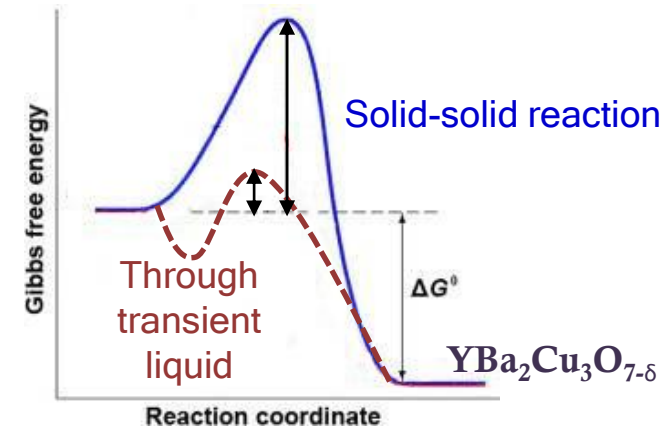
A. Quetalto et al, SUST (2023)



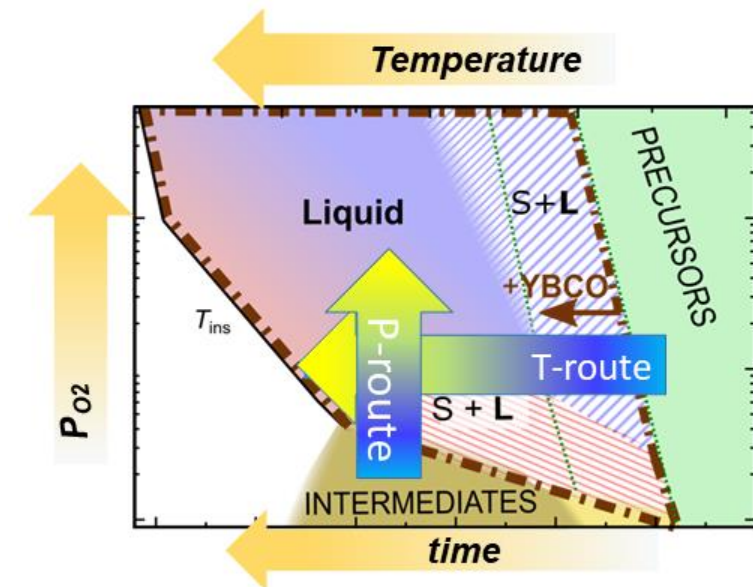
Nanocrystalline precursors



YBCO growth



- High throughput
- High growth rate (2300 nm/s demonstrated)
- High performance (2-5 MA/cm² at 77K)
- Uses simple reactor
- Large area processing
- Low cost/performance ratio



100 nm/s by **ultrafast-PLD** EuBCO/BHO
(transient liquid growth at high T PLD)

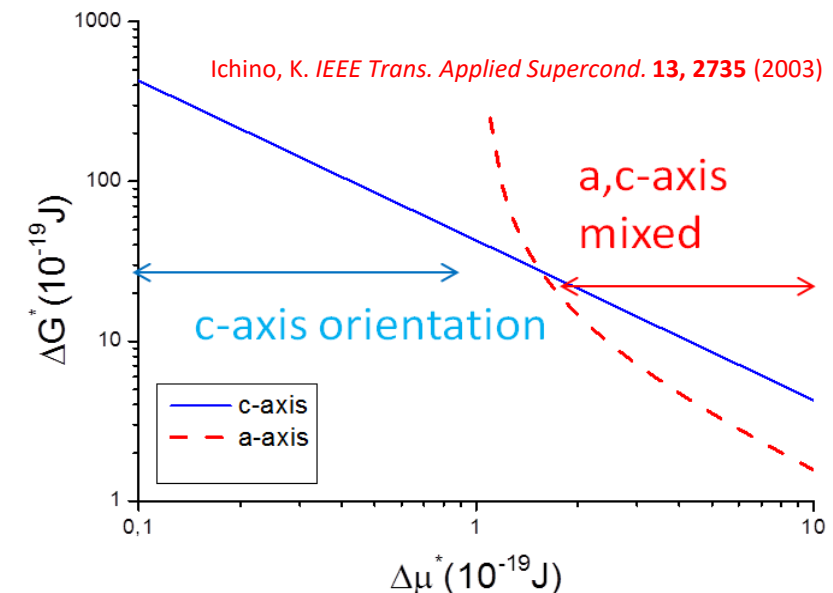
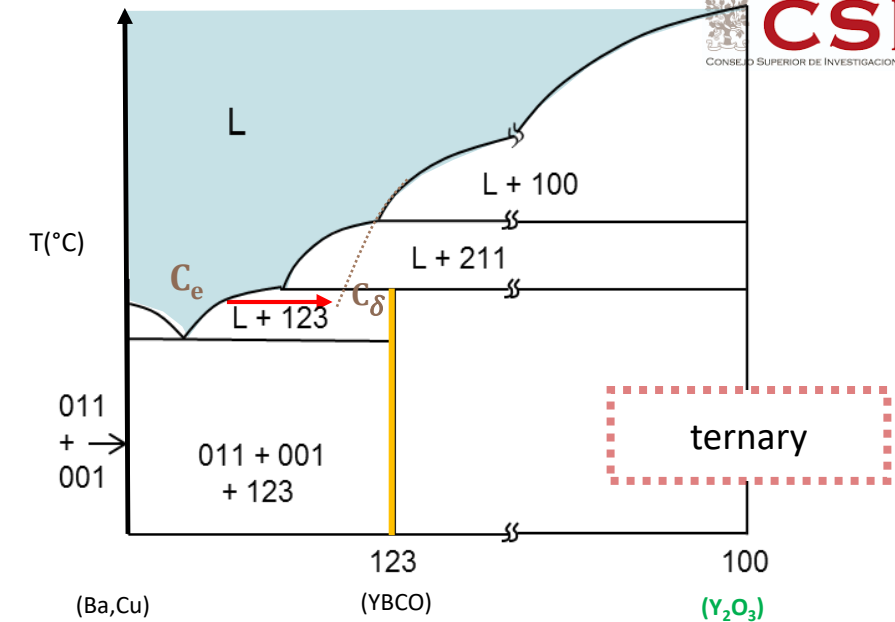
Y. Wu, Materials & Design 224 (2022)

WHAT DOES CONTROL TLAG ?

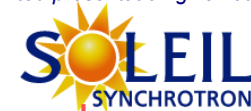
- **Supersaturation** $\sigma = (C_{\delta} - C_e) / C_e$
 C_e : RE equilibrium concentration
 - RE solubility (RE= Y, Gd, Sm, Yb, ...)
 - Ba/Cu Liquid composition
- Thermodynamic parameters: T, P_{O_2}
- Kinetic parameters: **heating ramp, pressure jump, gas velocity, P_{total}**
- **C-axis growth** is controlled by supersaturation:

$$\Delta\mu = kT \ln \frac{C_{\delta}}{C_e} = kT \ln(\sigma + 1)$$

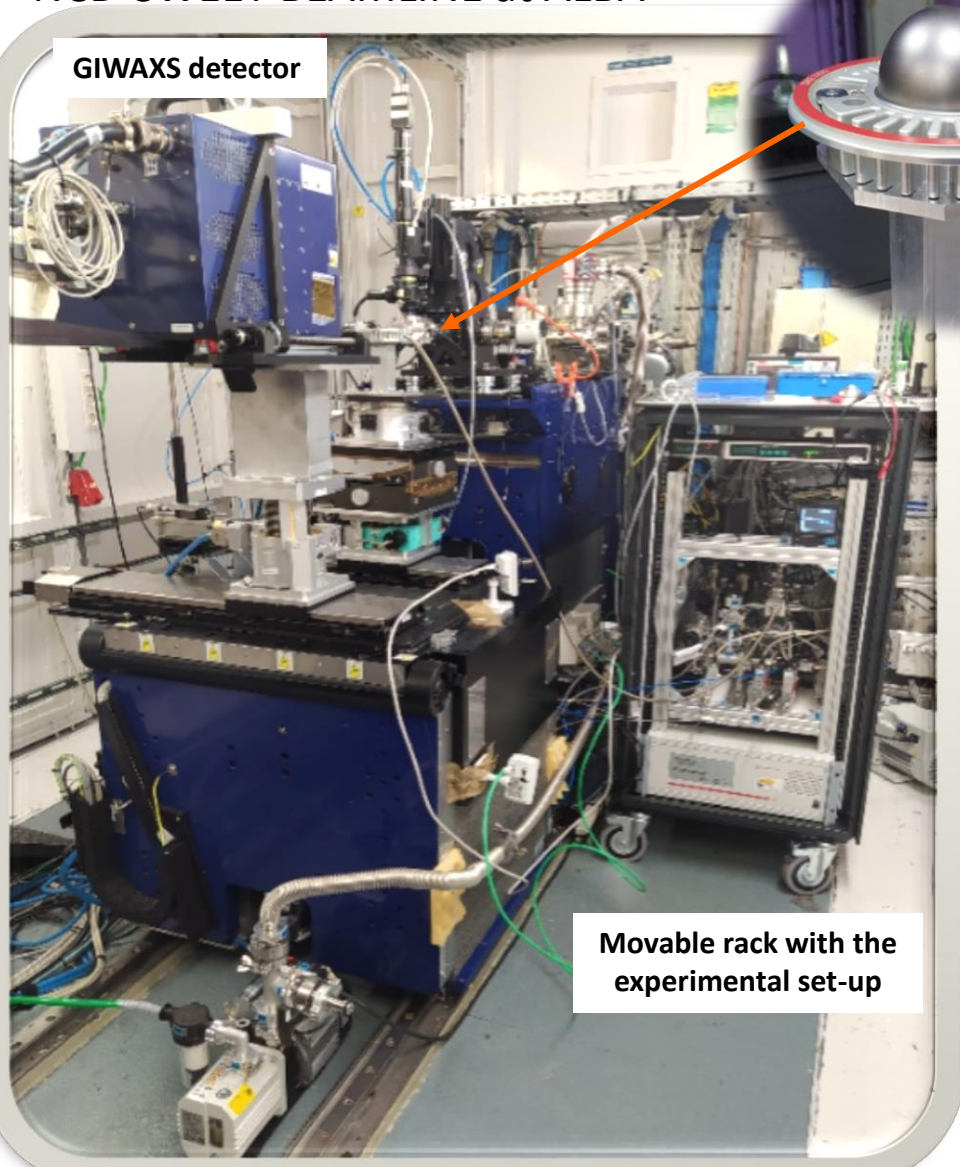
- TLAG is a non-equilibrium process of **high supersaturation** and **ultrahigh growth rate**
 - high density of pinning defects is generated
 - pre-formed nanoparticles can be embedded



Synchrotron installation



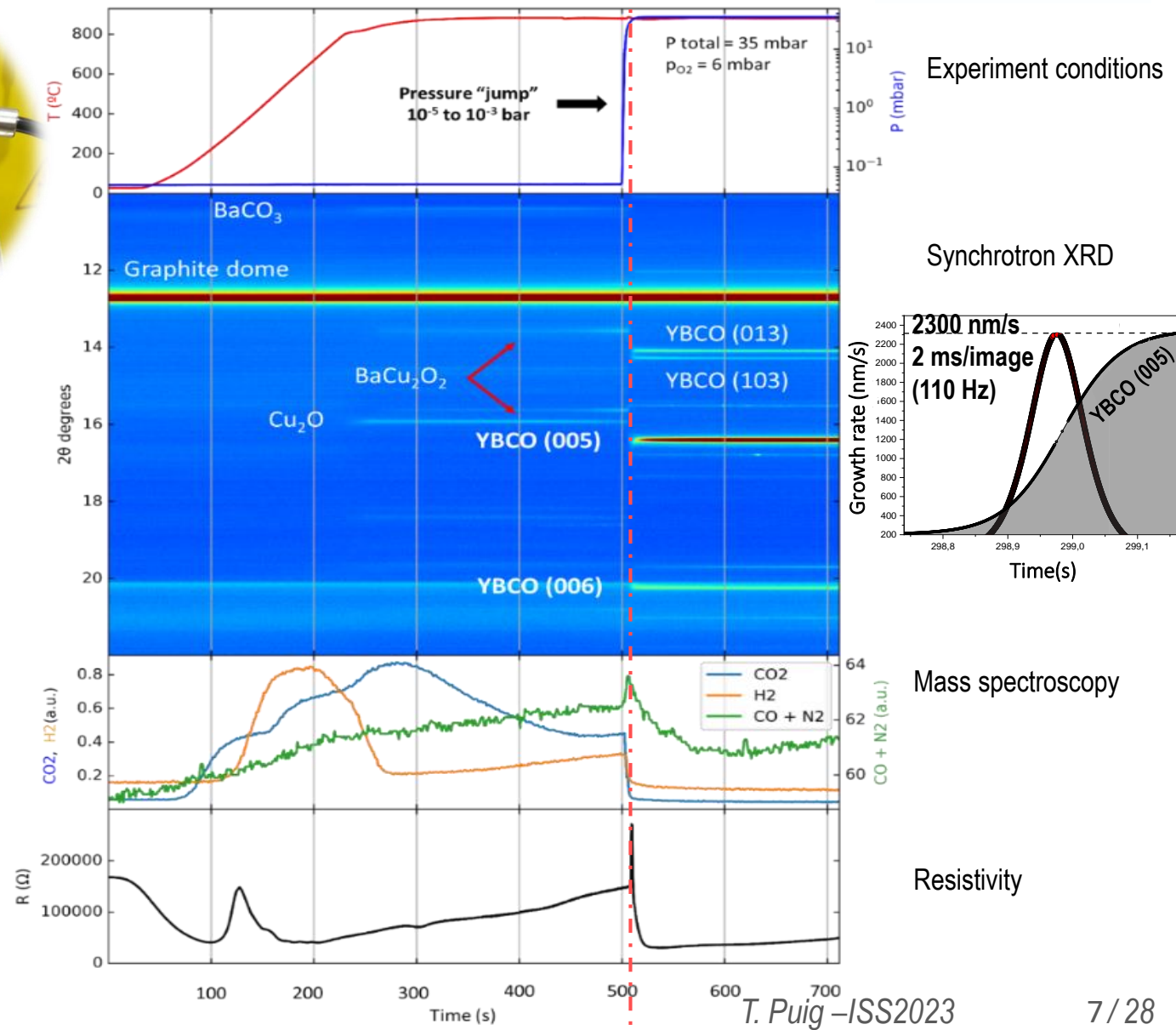
DIFFABS BEAMLINE at Soleil
NCD-SWEET BEAMLINE at ALBA



GIWAXS detector

Heating stage

Movable rack with the experimental set-up

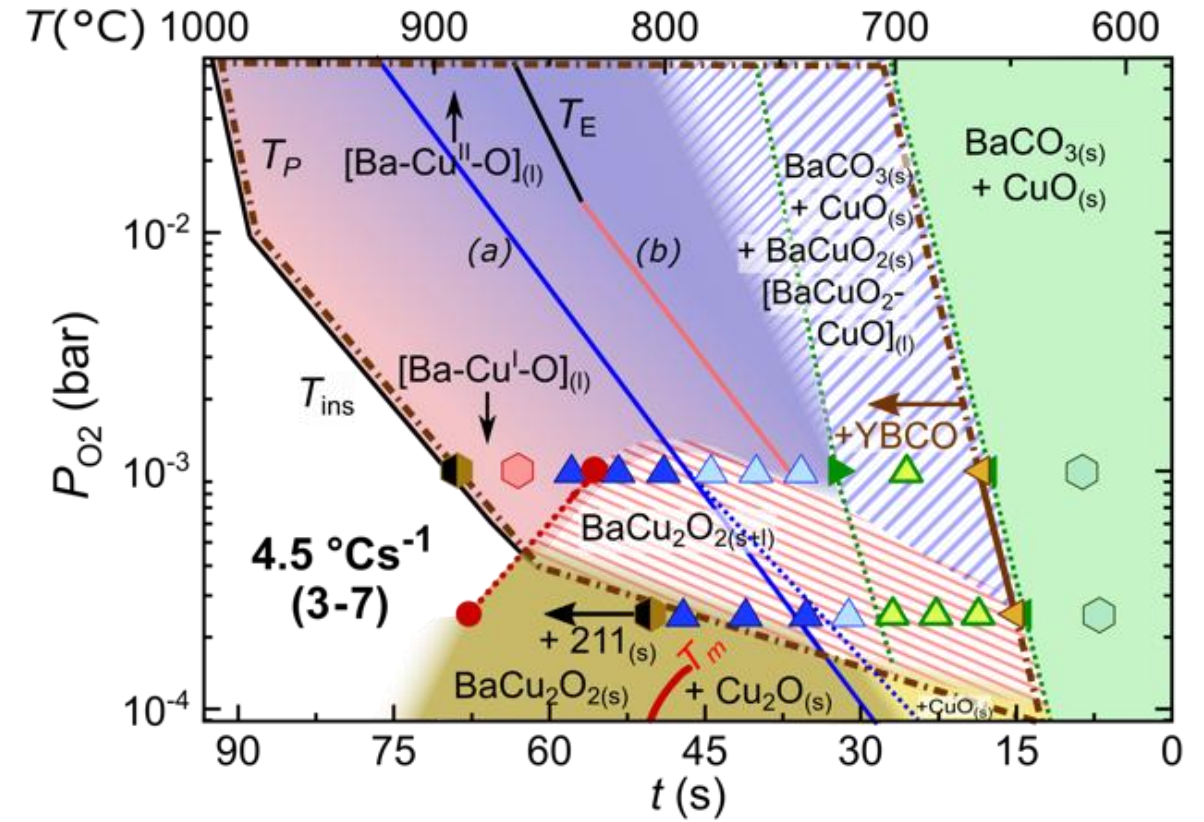
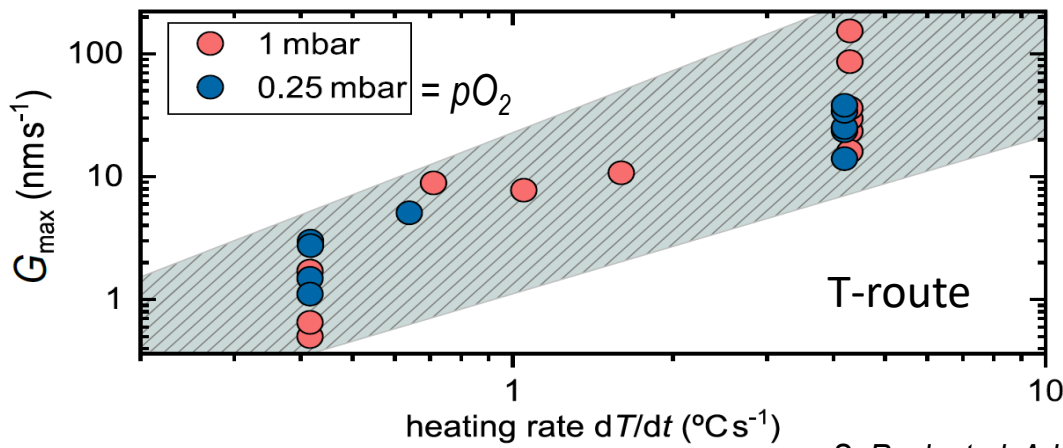
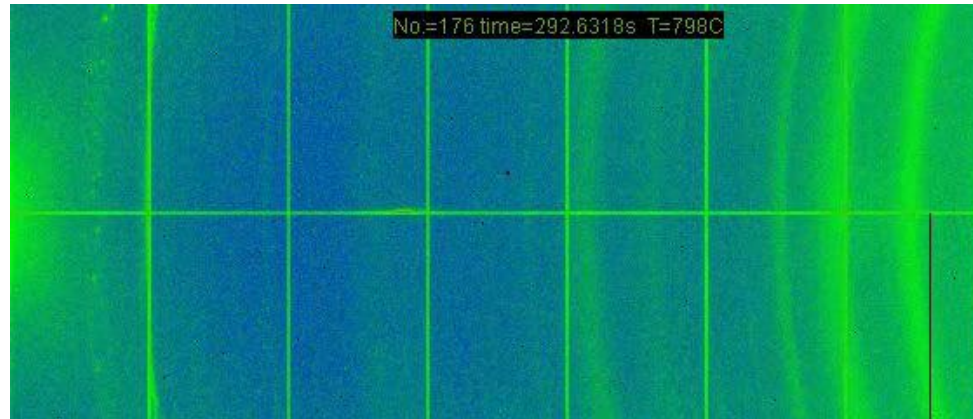
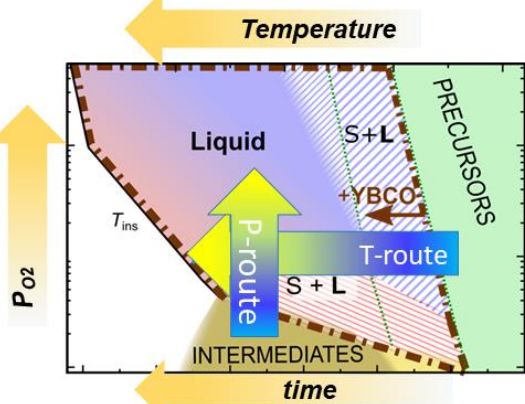


TLAG T-route

Kinetic process strongly differing from equilibrium reported phases



synchrotron
in-situ XRD

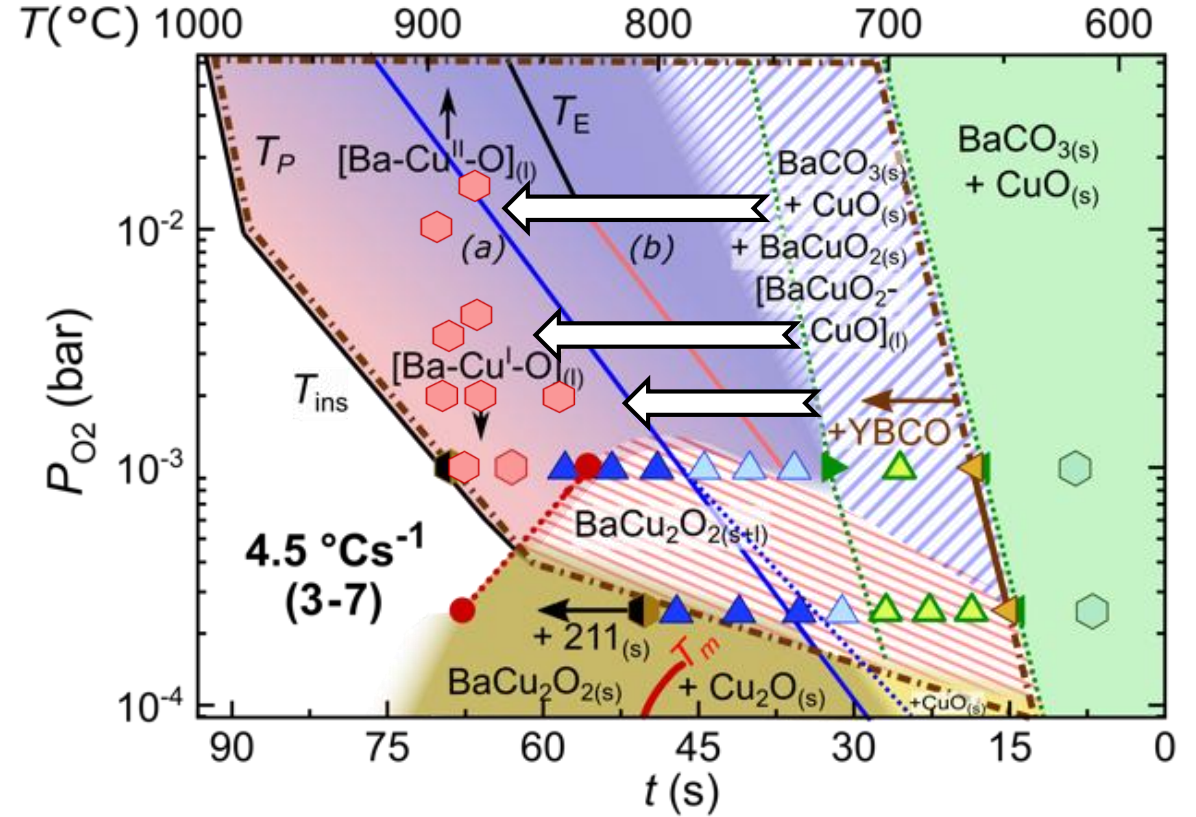
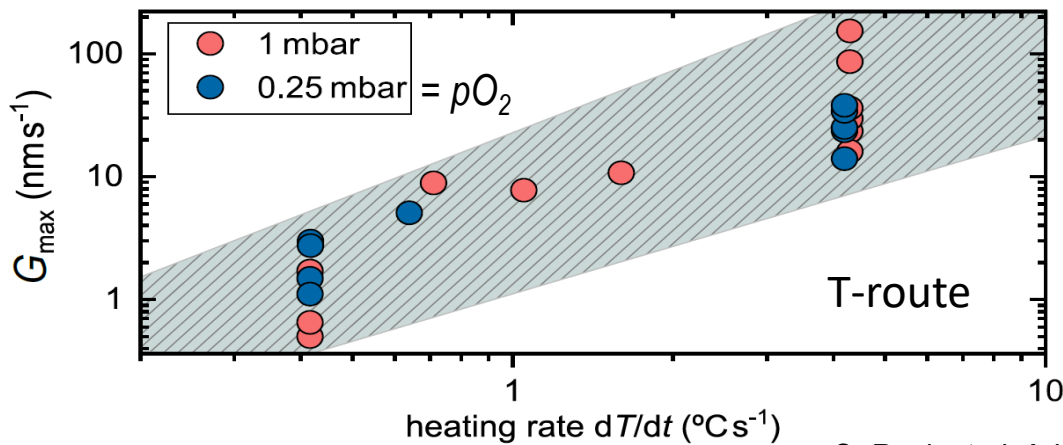
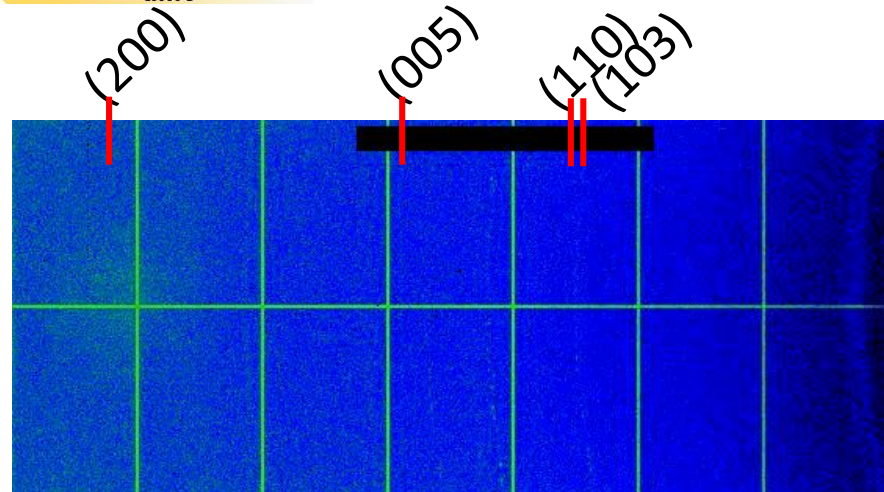
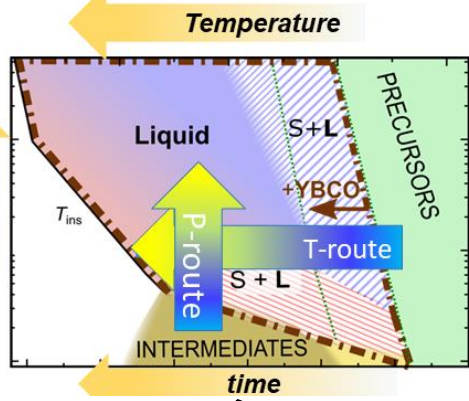


- ▲ YBCO + BaCO_{3(s)} + CuO_(s)
- △ YBCO + BaCuO_{2(s)} + CuO_(s)
- ▲ YBCO + CuO_(s)
- ▲ YBCO + Cu₂O_(s)
- YBCO
- YBCO decomp.
- T_E — BaCuO_{2(s)} + CuO_(s) → [BaCuO₂ + CuO]_(l)
- T_m — BaCuO_{2(s)} → [BaCuO₂]_(l)
- (a) — CuO_(s) → Cu₂O_(s)
- (b) — BaCuO_{2(s)} → BaCuO_{2(s)}



TLAG T-route

Kinetic process strongly differing from equilibrium reported phases



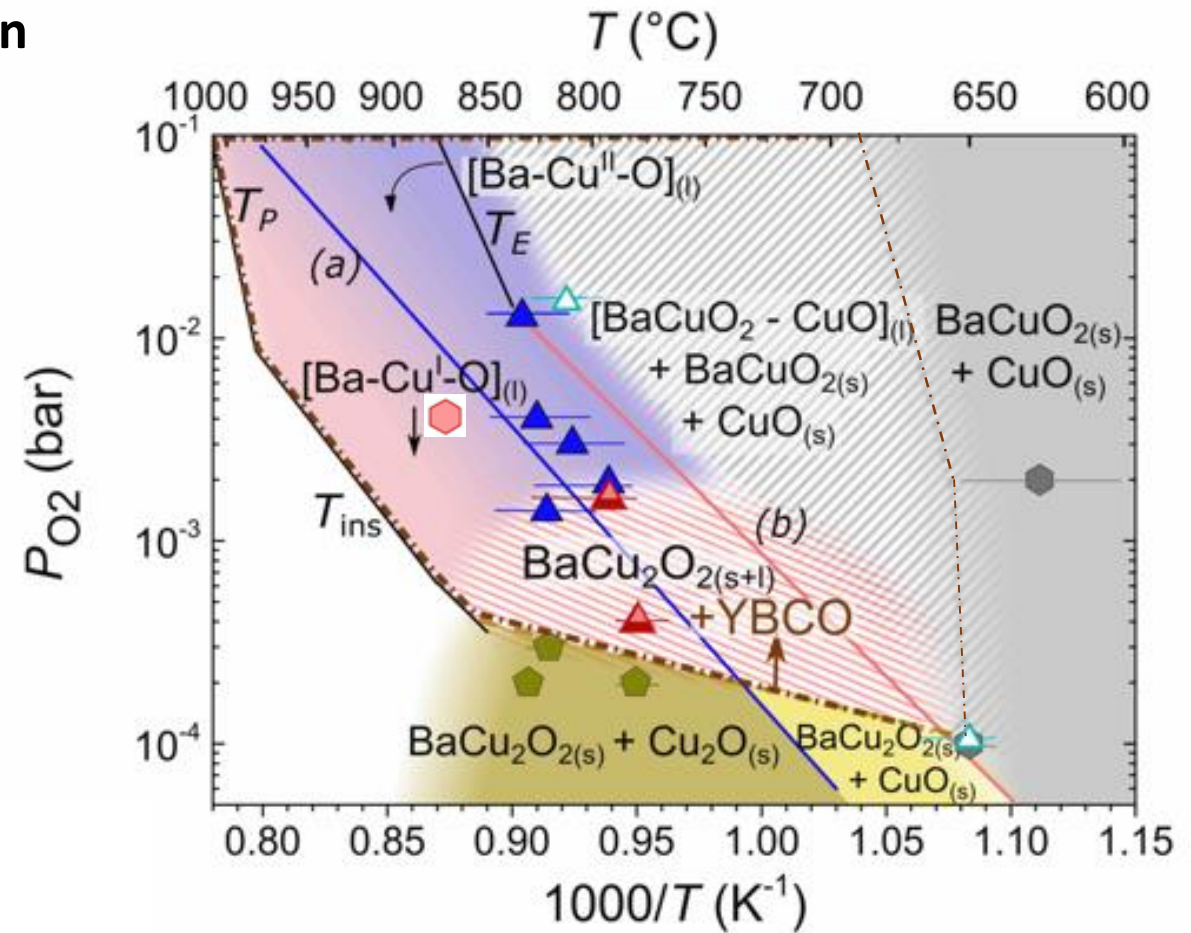
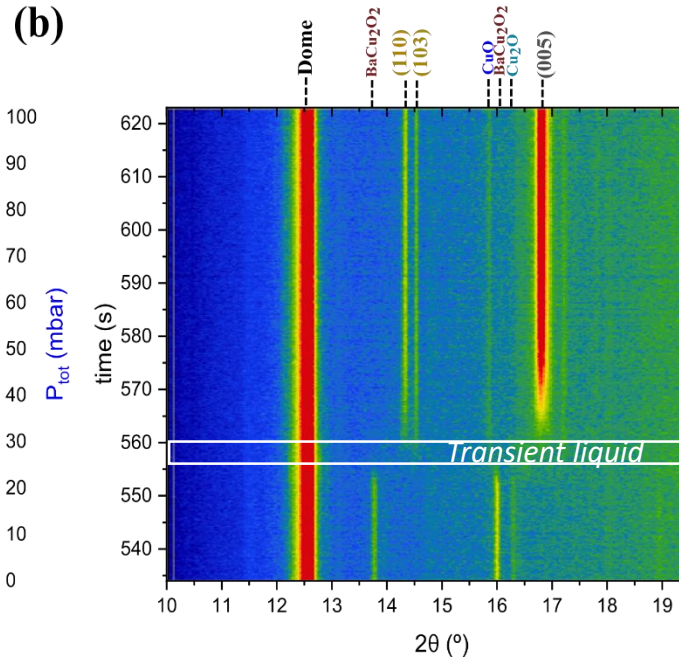
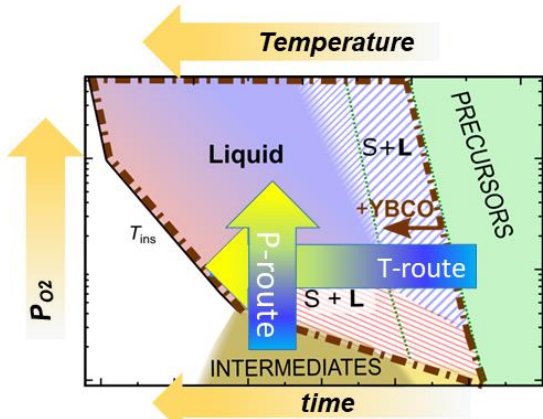
- ▲ YBCO + BaCO_{3(s)} + CuO_(s)
- ▲ YBCO + BaCuO_{2(s)} + CuO_(s)
- ▲ YBCO + CuO_(s)
- ▲ YBCO + Cu₂O_(s)
- YBCO
- YBCO decomp.
- T_E BaCuO_{2(s)} + CuO_(s) → [BaCuO₂ + CuO]_(l)
- T_m BaCu₂O_{2(s)} → [BaCu₂O₂]_(l)
- (a) CuO_(s) → Cu₂O_(s)
- (b) BaCuO_{2(s)} → BaCu₂O_{2(s)}

TLAG P-route

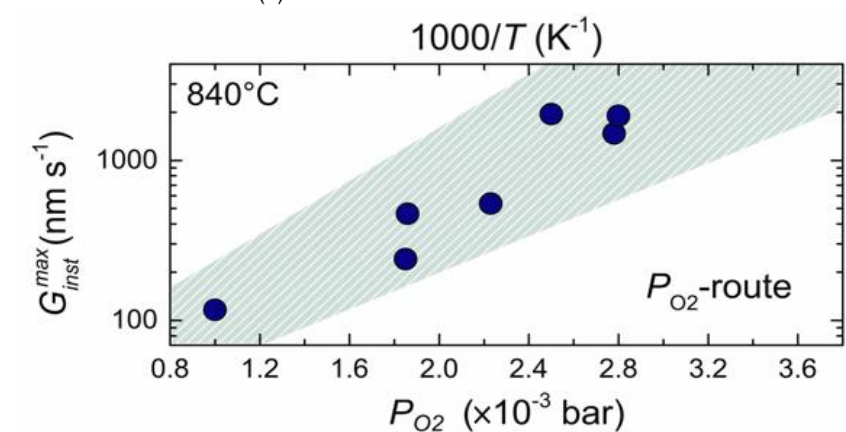
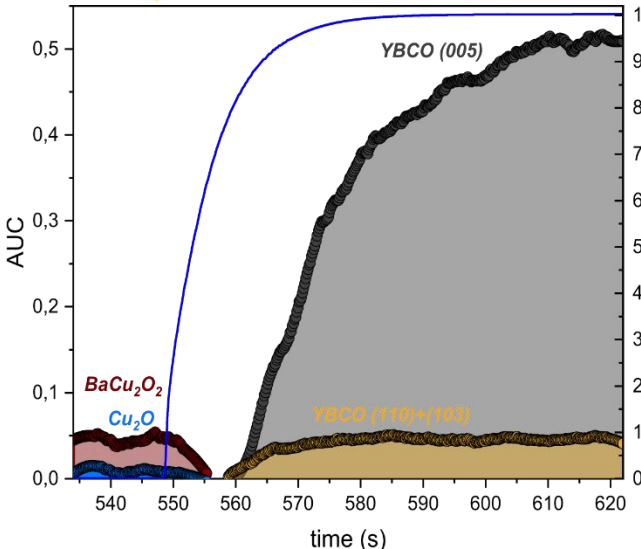


synchrotron
in-situ XRD

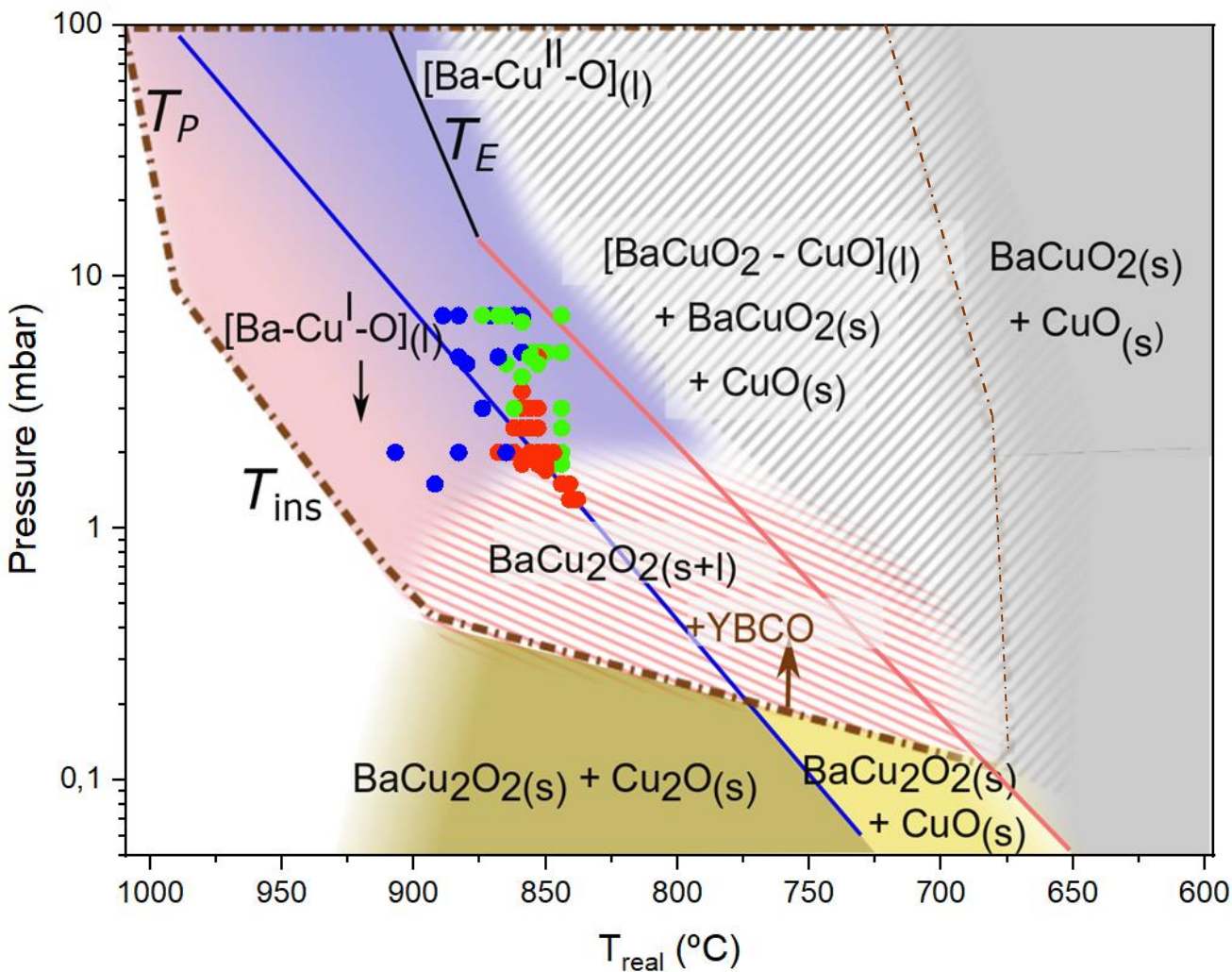
Kinetic process decoupling BaCO_3 reaction from YBCO formation



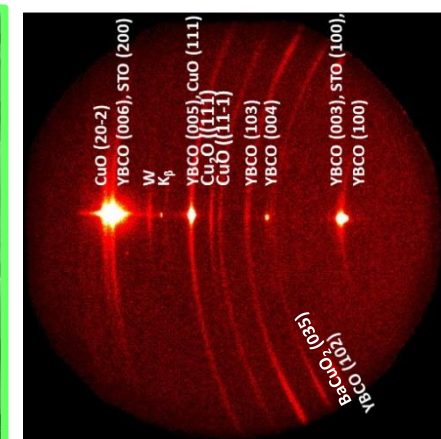
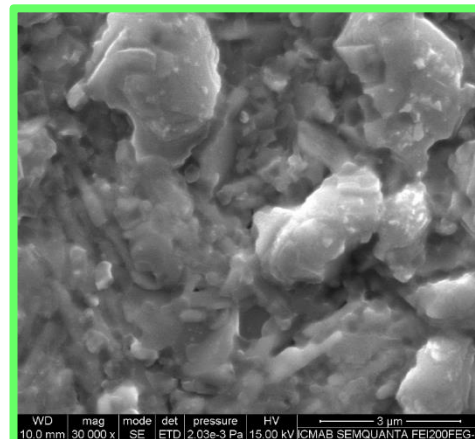
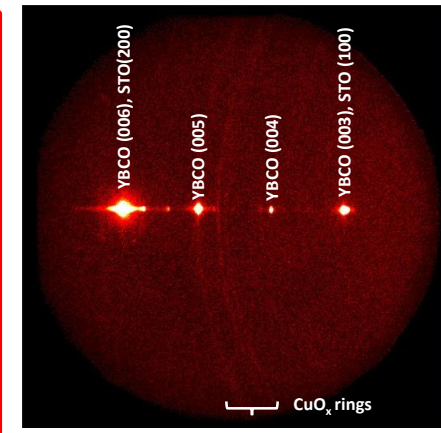
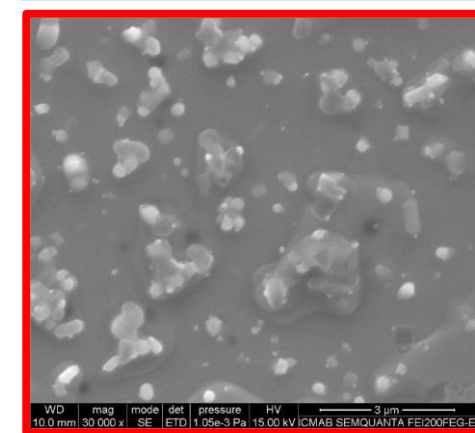
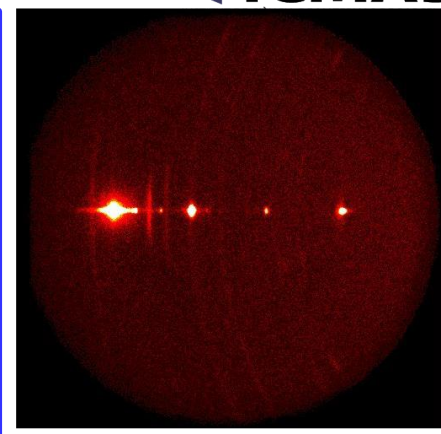
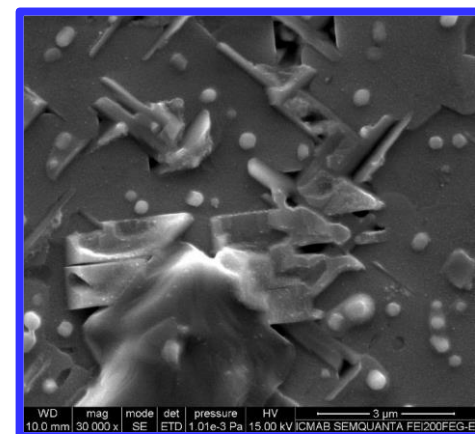
- △ $\text{BaCuO}_{2(s)} + \text{YBCO} + \text{CuO}_{(s)}$
- ▲ $\text{YBCO} + \text{Cu}_x\text{O}_{(s)}$
- ▲ $\text{BaCuO}_{2(s)} + \text{YBCO} + \text{Cu}_x\text{O}_{(s)}$
- ◻ YBCO
- T_E — $\text{BaCuO}_{2(s)} + \text{CuO}_{(s)} \rightarrow [\text{BaCuO}_2 + \text{CuO}]_{(l)}$
- T_m — $\text{BaCuO}_{2(s)} \rightarrow [\text{BaCuO}_2]_{(l)}$
- (a) — $\text{CuO}_{(s)} \rightarrow \text{Cu}_2\text{O}_{(s)}$
- (b) — $\text{BaCuO}_{2(s)} \rightarrow \text{BaCuO}_{2(s)}$



YBCO P-route epitaxial windows on STO

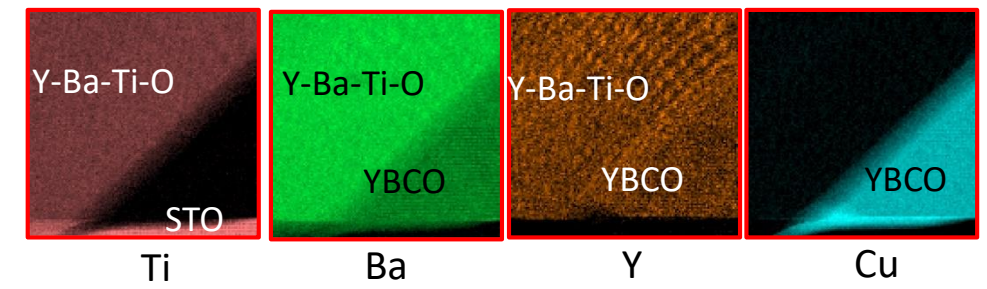
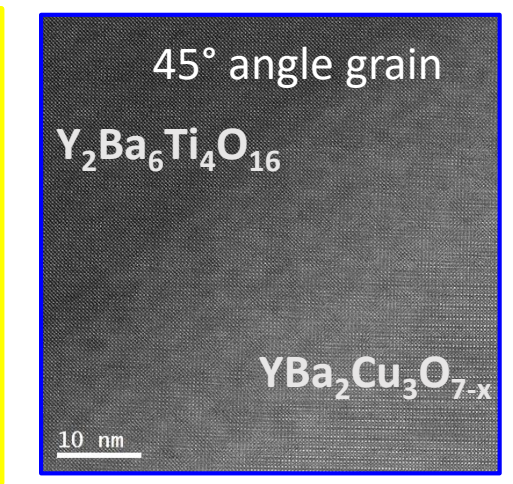
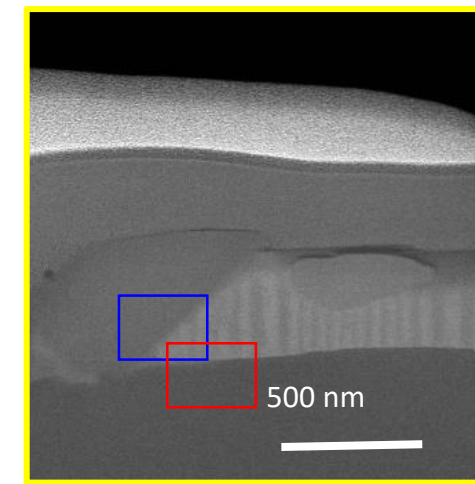
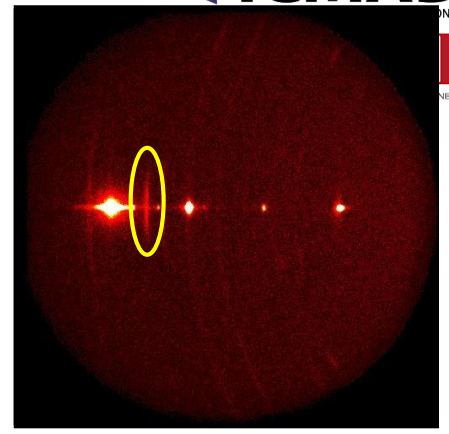
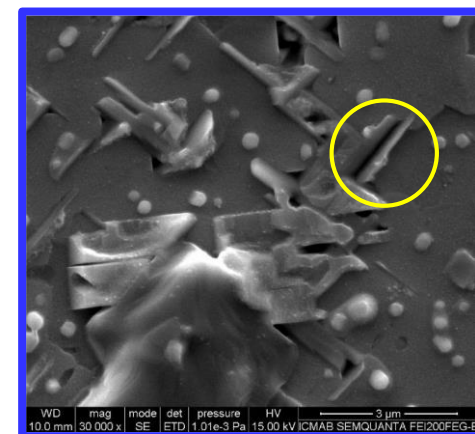
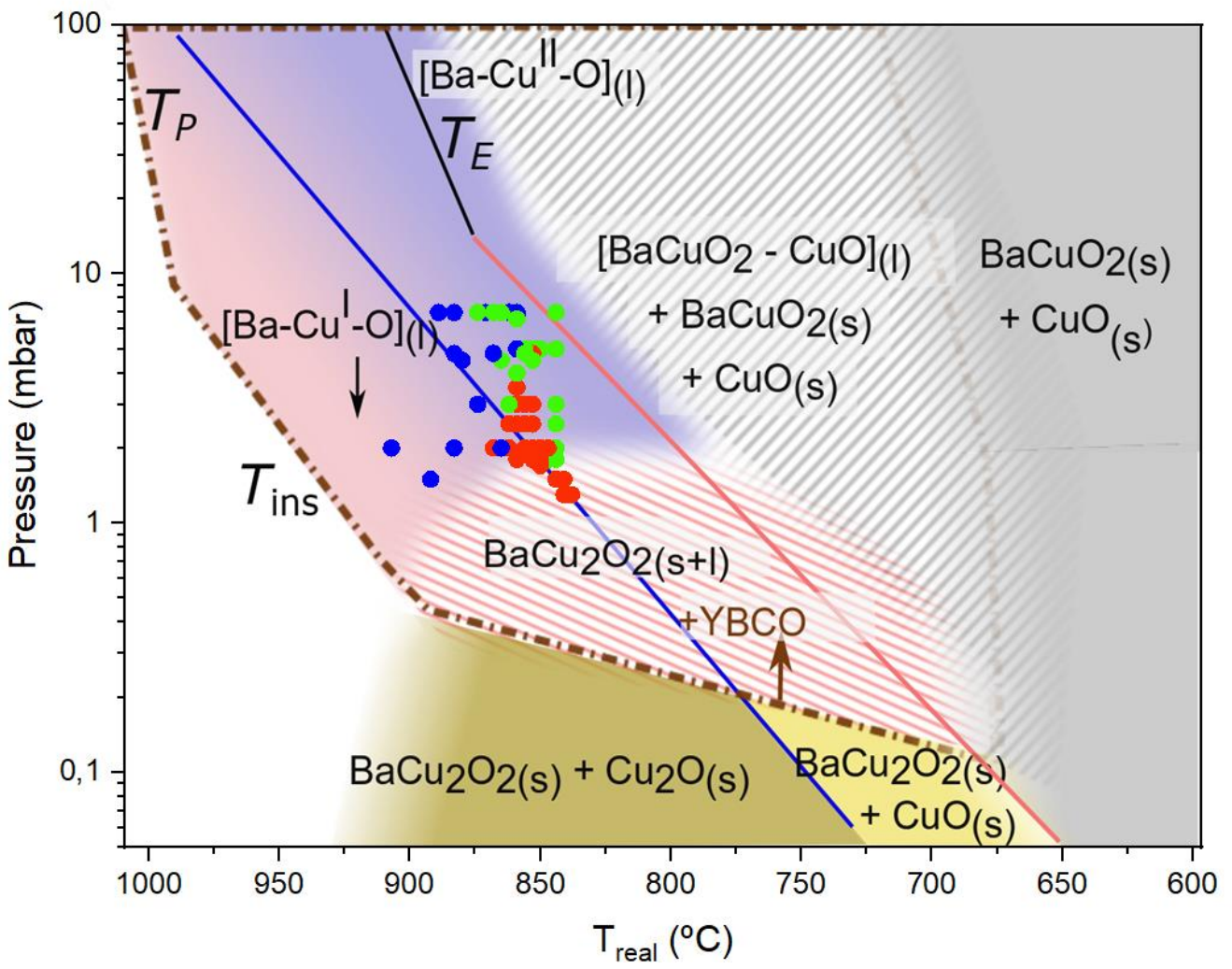


Reactivity
Epitaxy
Non fully epitaxy



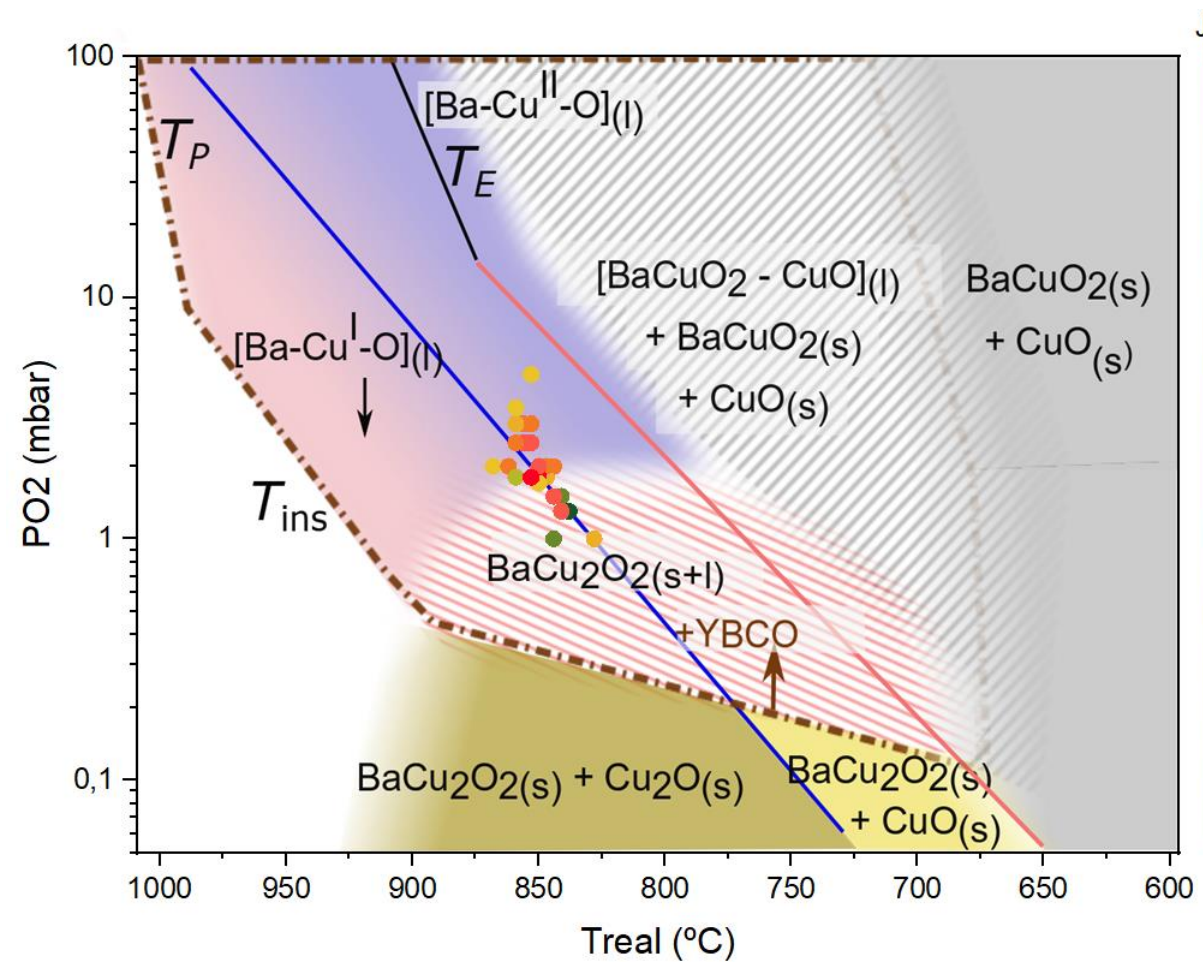
CuO segregation at the surface depends on the liquid composition chosen

YBCO P-route epitaxial windows on STO

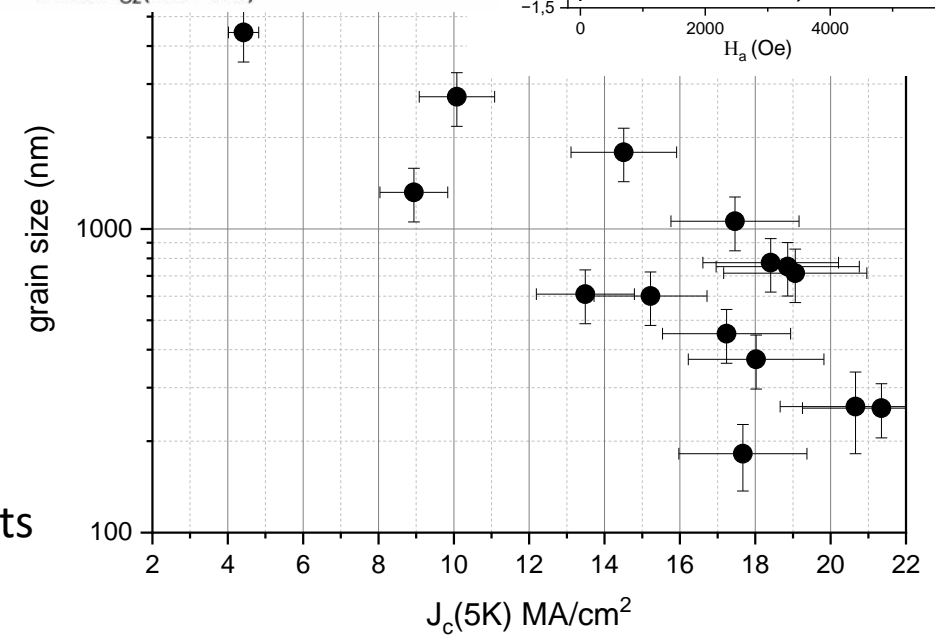
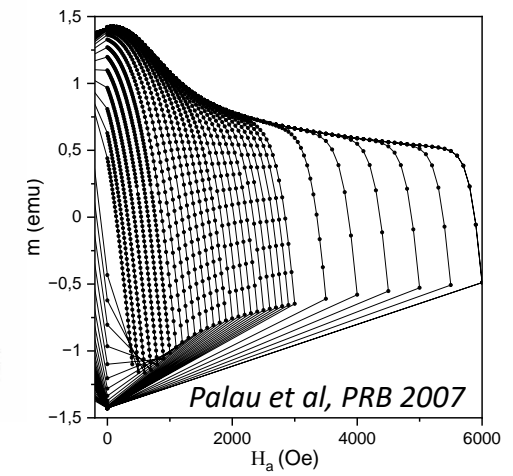
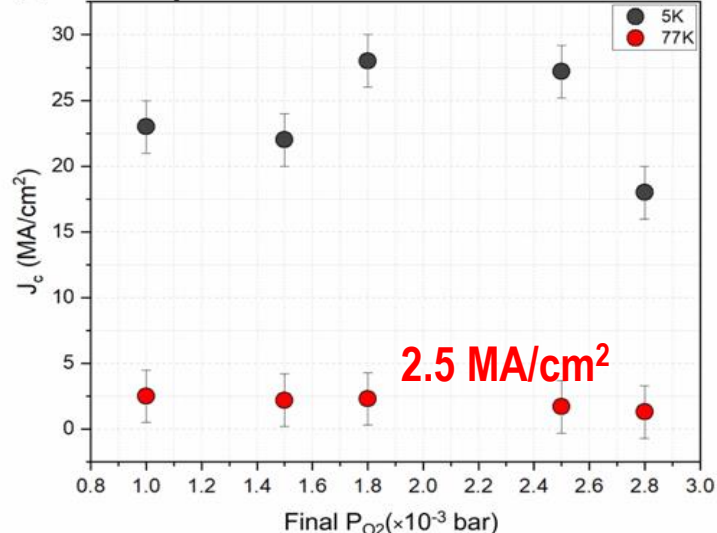


Liquid corrosion needs to be considered and minimized (buffer layers, RE, process conditions)

Critical currents at different growth conditions

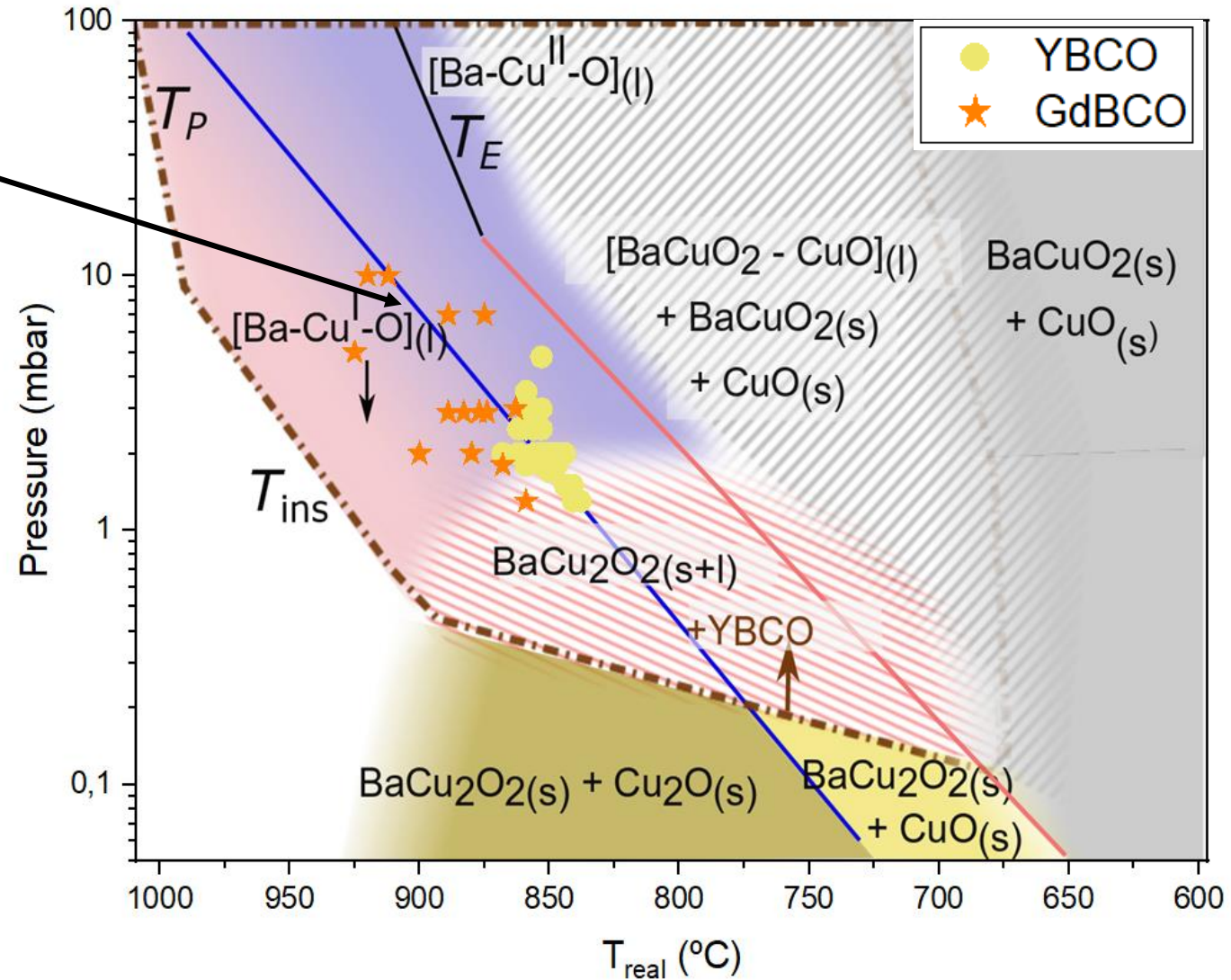
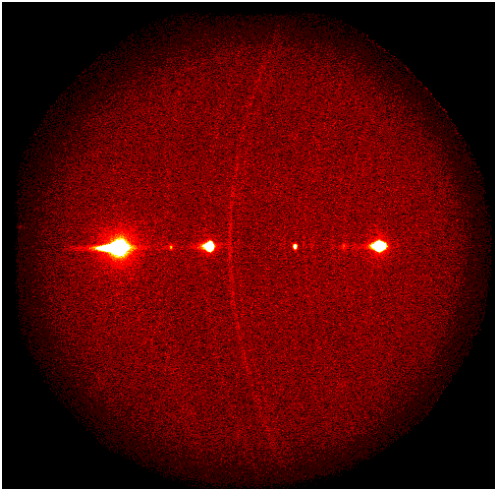
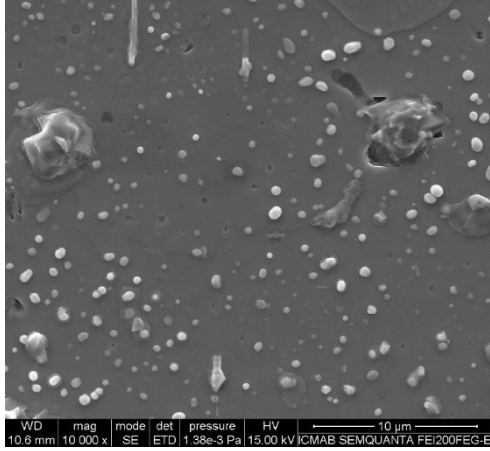


50 nm/s \longrightarrow 1500 nm/s



High nucleation densities (small grain sizes) favor high critical currents
 In agreement with BaF₂ methods (Solovyov et al, SUST, 2008)

GdBCO vs YBCO c-axis window on STO substrates in P-route



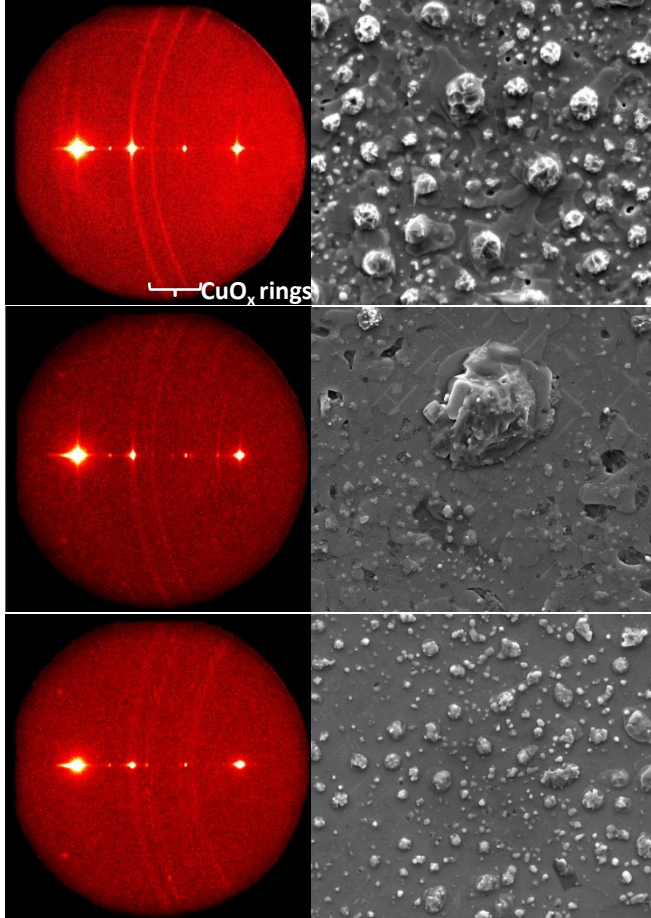
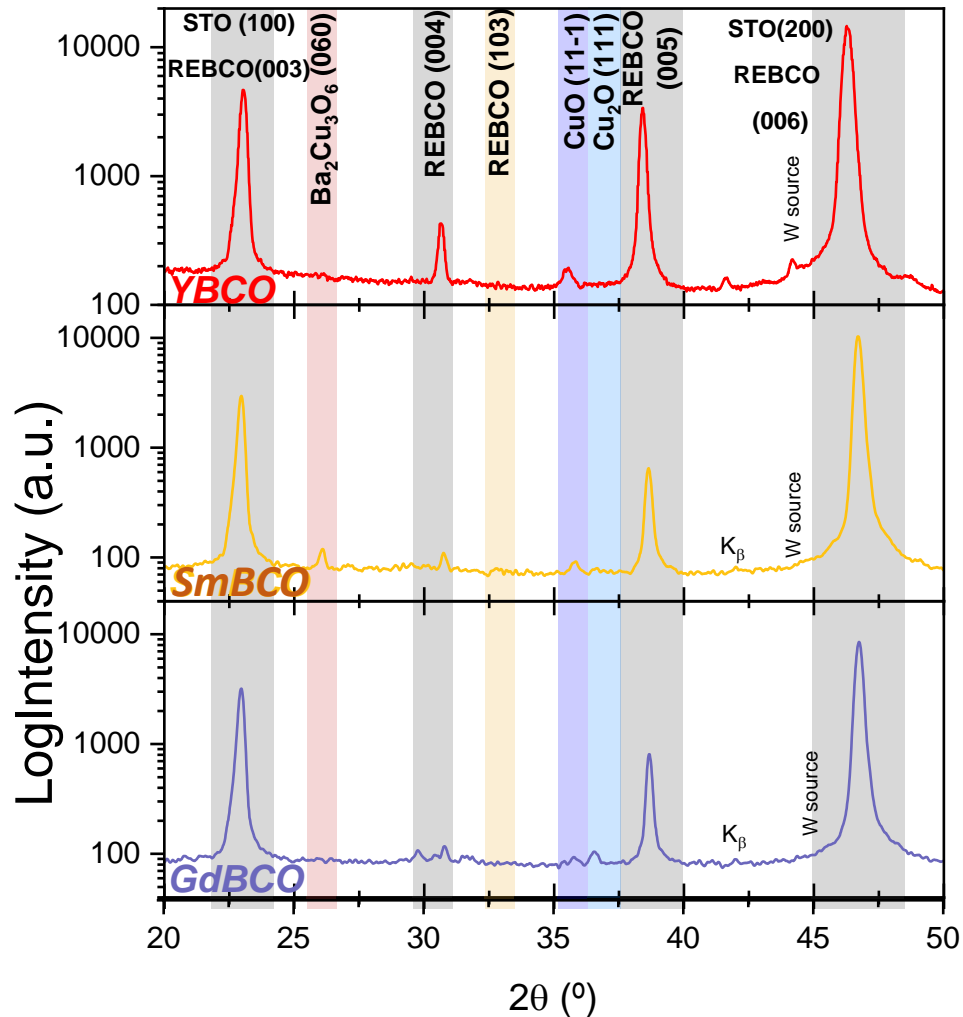
The RE modifies:

- Epitaxial windows
- Supersaturation value
- Solubility in the liquid
- Nucleation density
- Growth rate

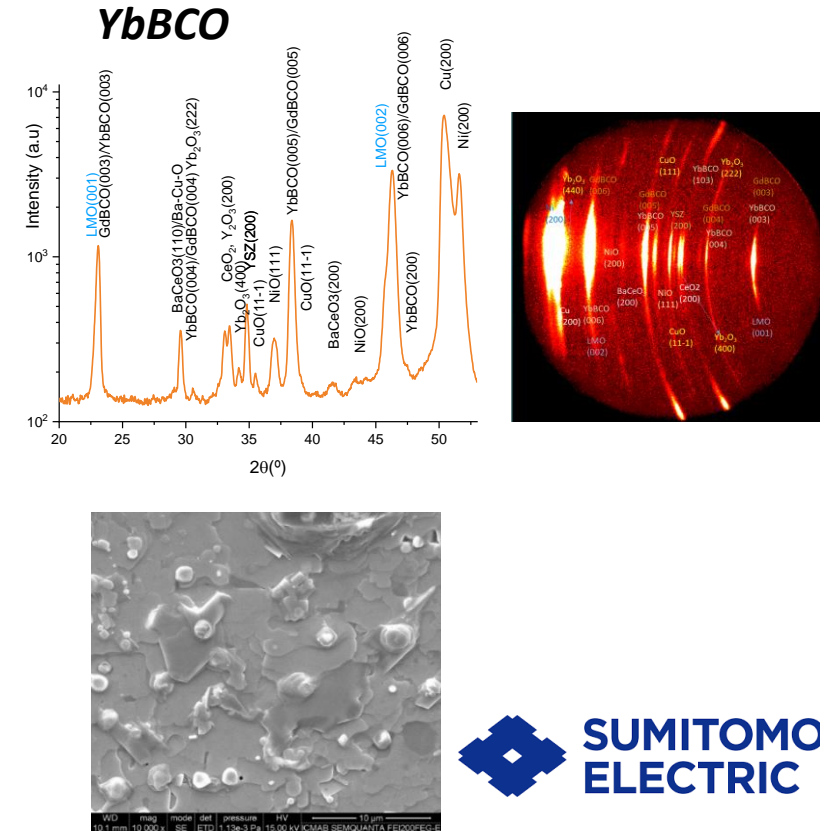
GdBCO c-axis window is larger than YBCO. It also avoids reactivity with the STO substrate
 This suggests a larger nucleation rate for GdBCO

REBCO films with RE = Y, Gd, Sm, Yb

P-route examples on STO



T-route example on CC

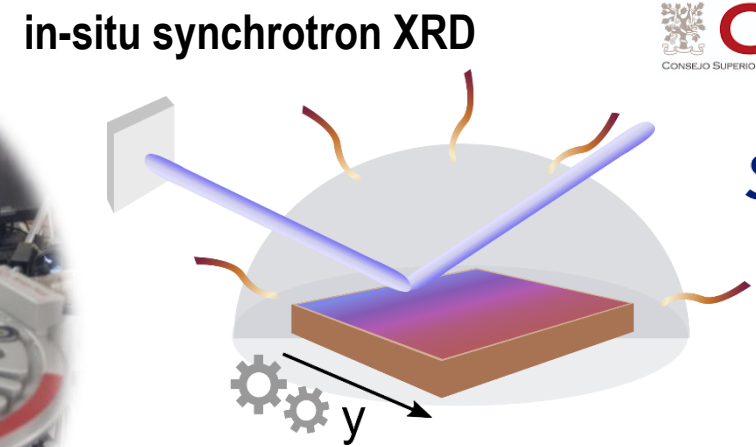
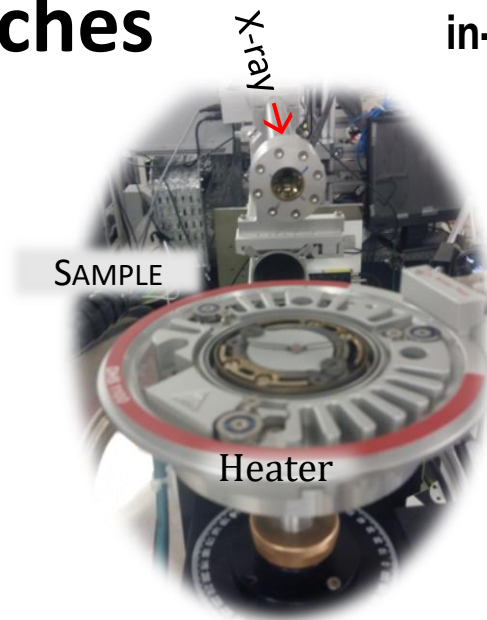
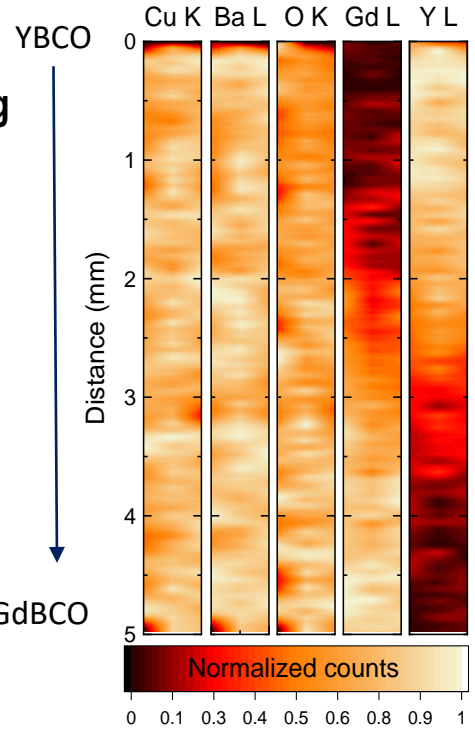
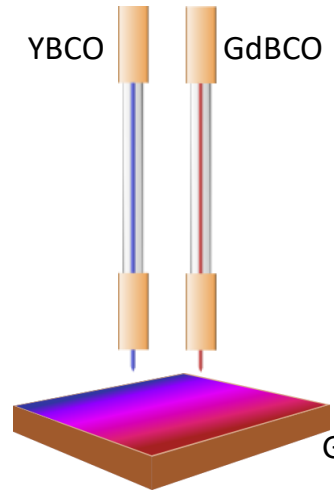


TLAG is compatible with REBCO film growth with different RE

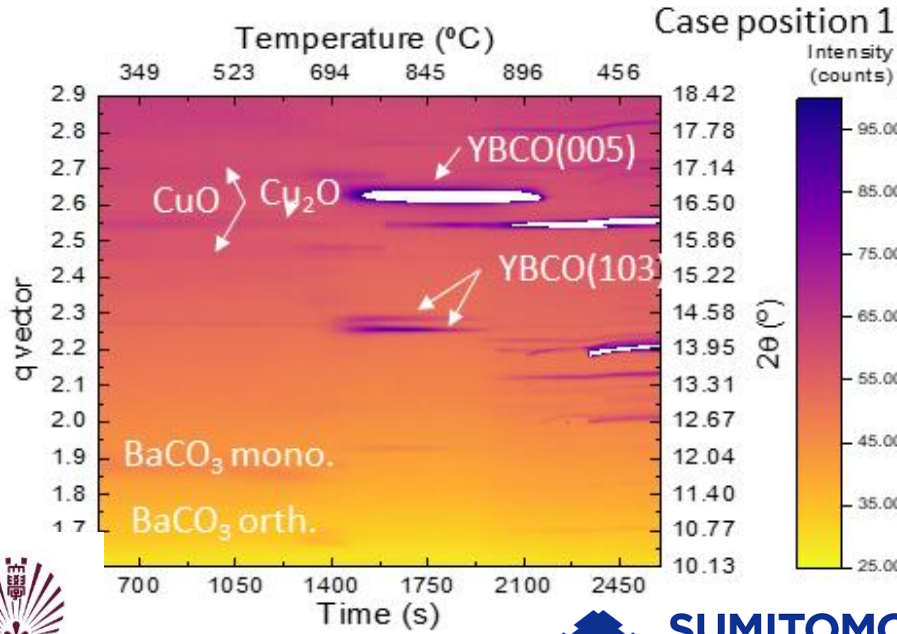
High Throughput Experimentation using Compositional Gradients enabling ML approaches



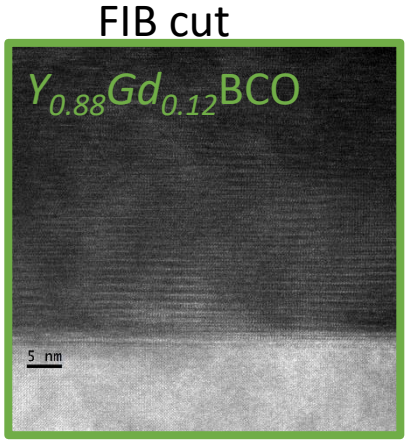
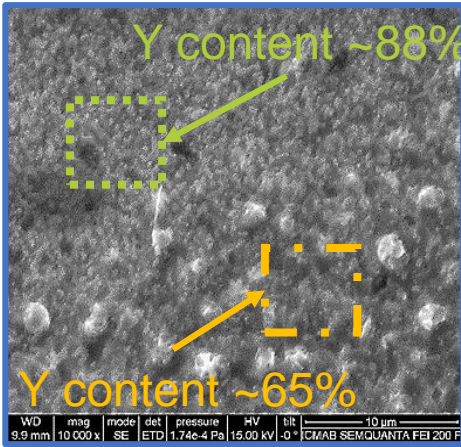
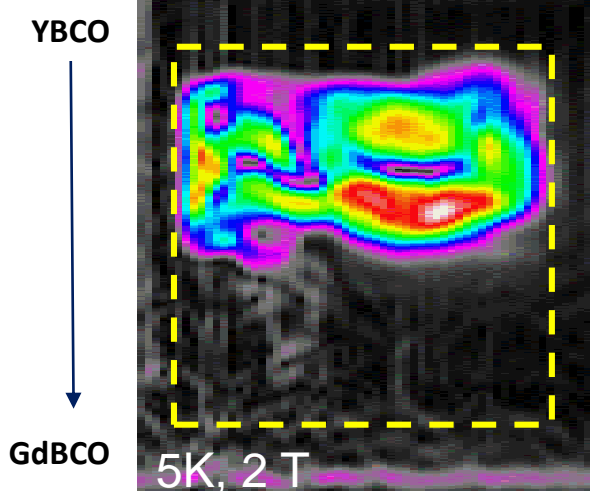
Combinatorial DoD Ink Jet Printing



Data is segmented by positions for analysis



Sheet current density, J_s (kA/m)

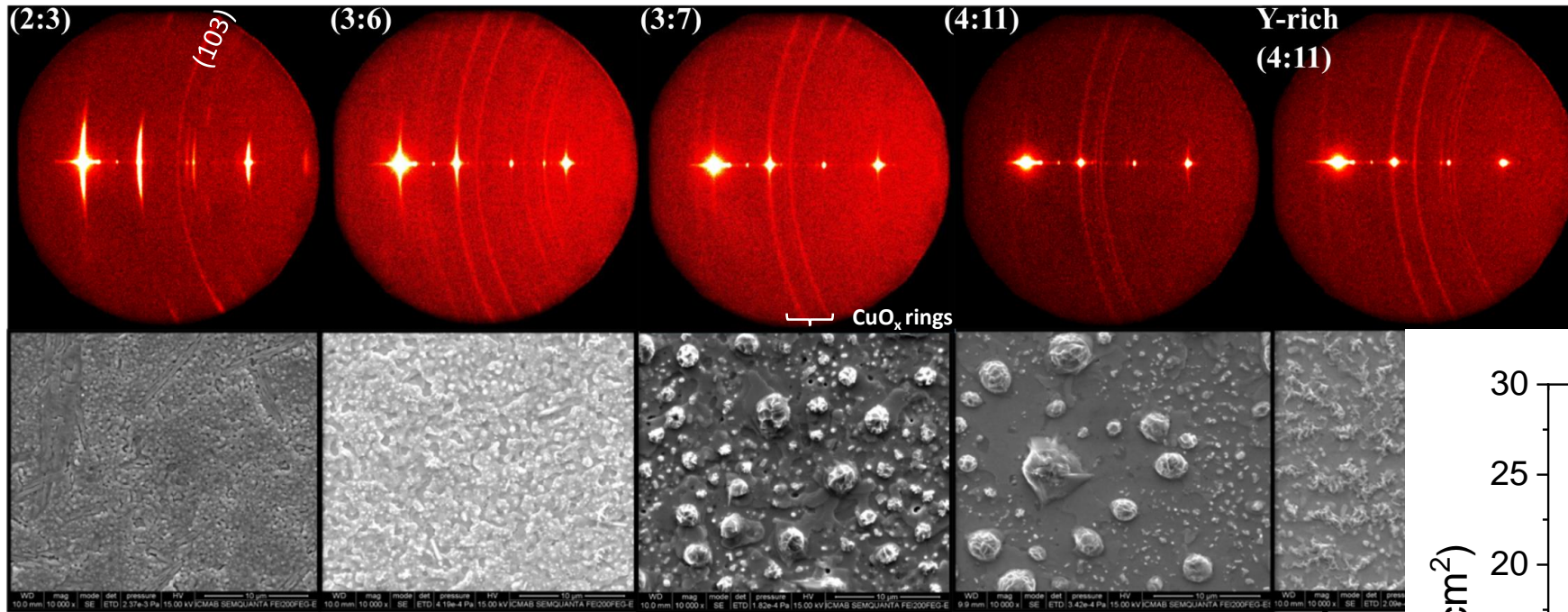


Collab. T. Kiss

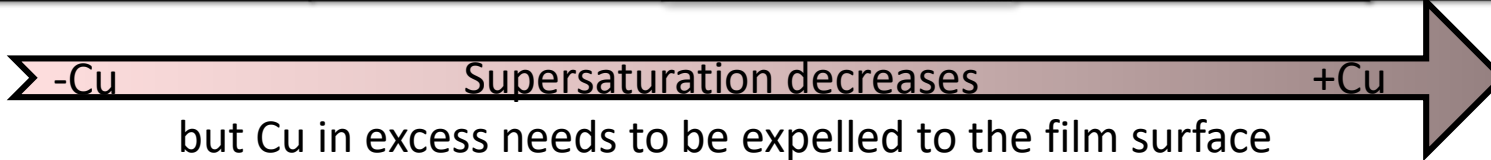
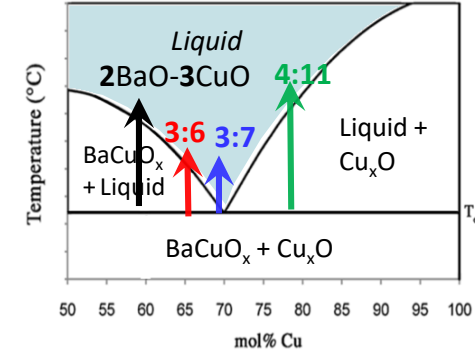


T. Puig -ISS2023

Ba:Cu liquid composition, a way to control the supersaturation

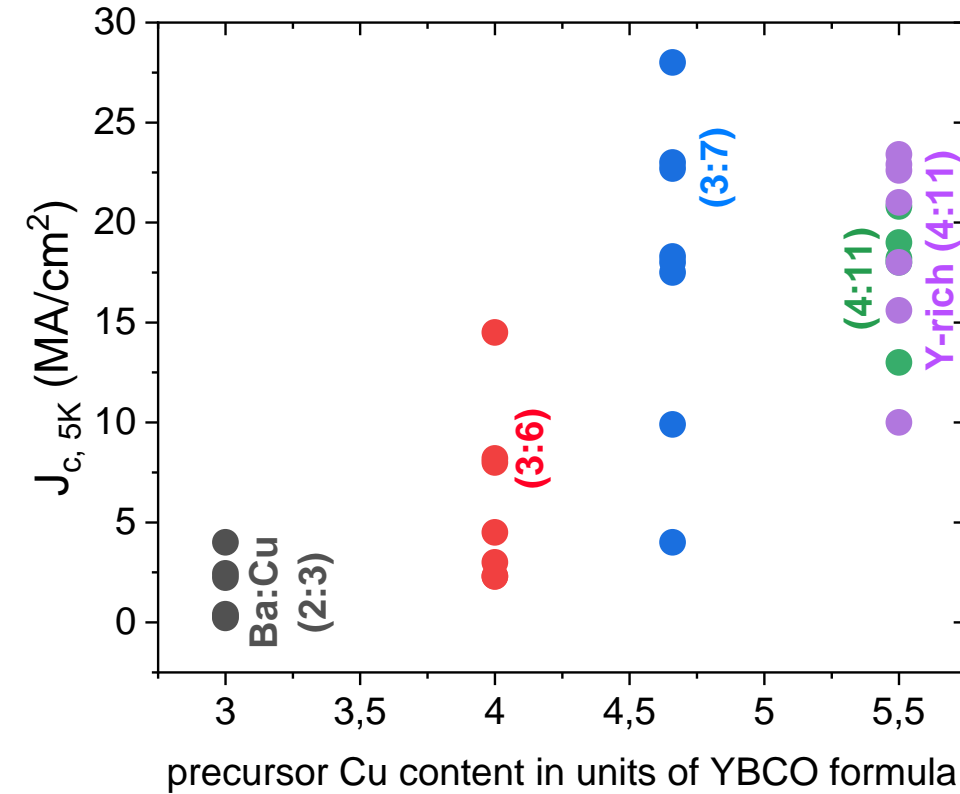


Experiments for YBCO composition



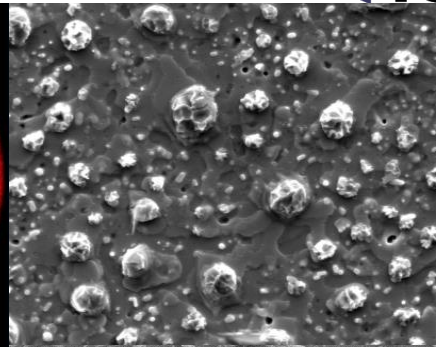
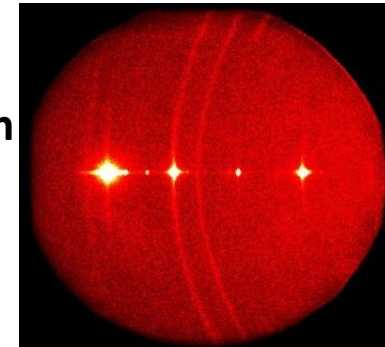
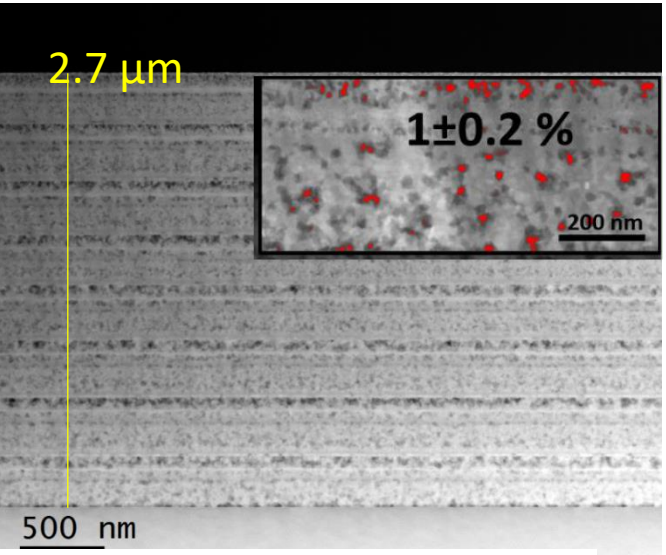
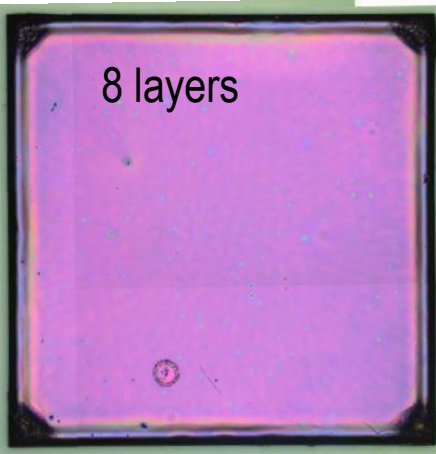
The (Ba:Cu) liquid composition:
 Determines the supersaturation
 c-axis growth
 Nucleation density
 Growth rate
 J_c at self-field

(3:7) is a good compromise
 (our mostly chosen composition)

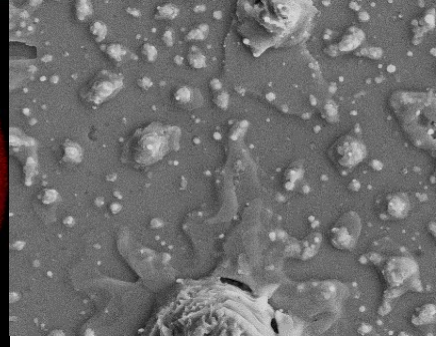
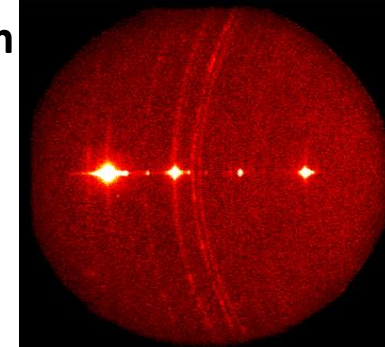


The thickness challenge

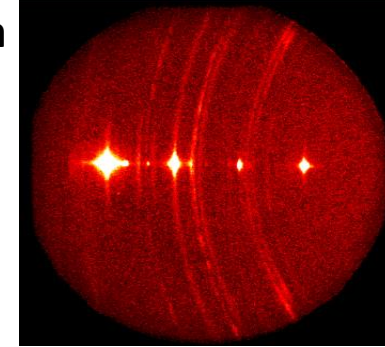
Case YBCO



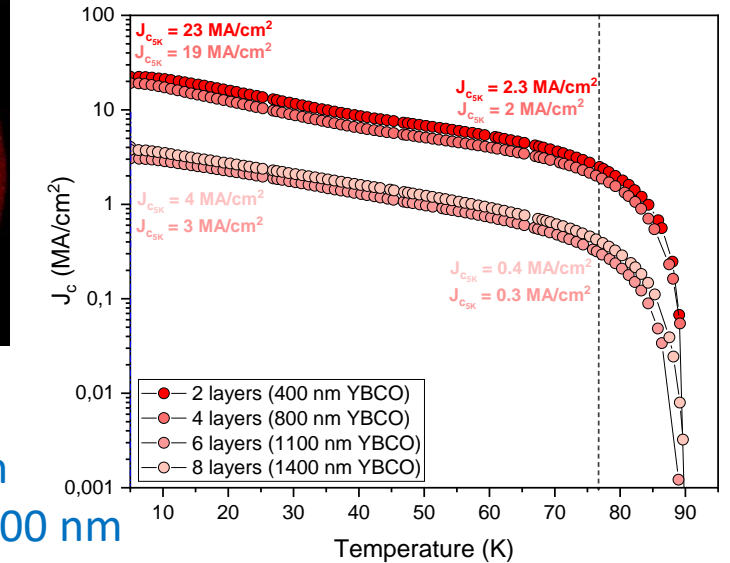
400 nm



800 nm

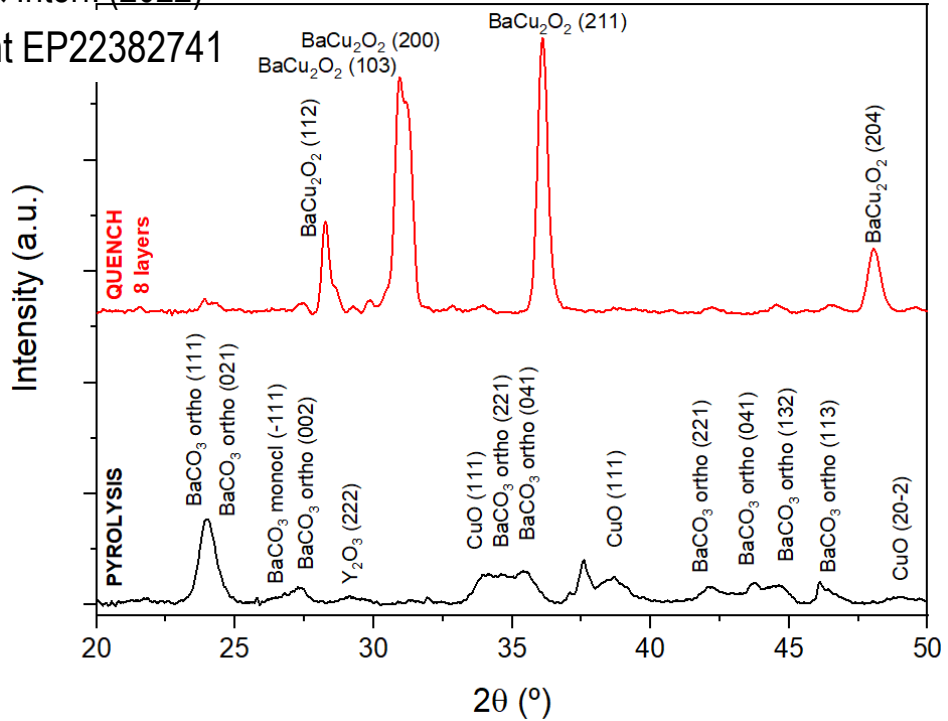


1100 nm



L. Saltarelli et al, ACS Appl. Mat. & Interf., (2022)

Patent EP22382741

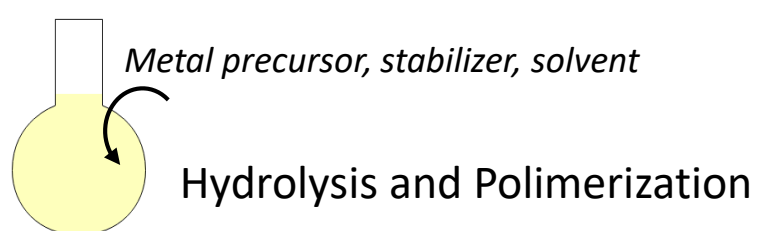


BaCO₃ is not an issue even at 2.7 μm
 TLAG growth now optimized up to 800 nm
 Interesting to test for other REBCO

TLAG nanocomposites with Multifunctional Colloidal Inks

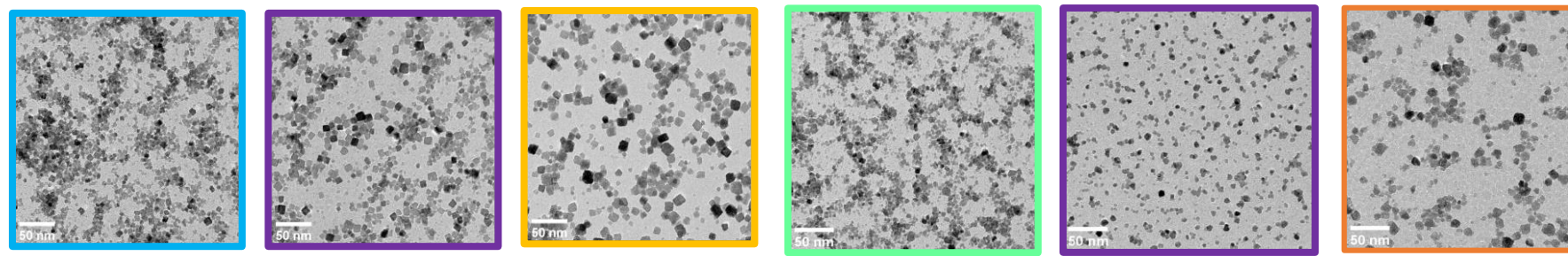
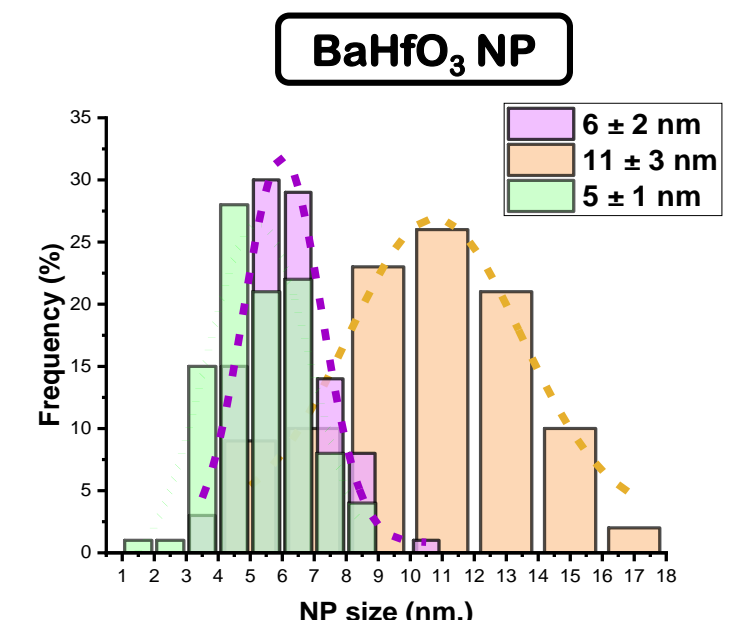
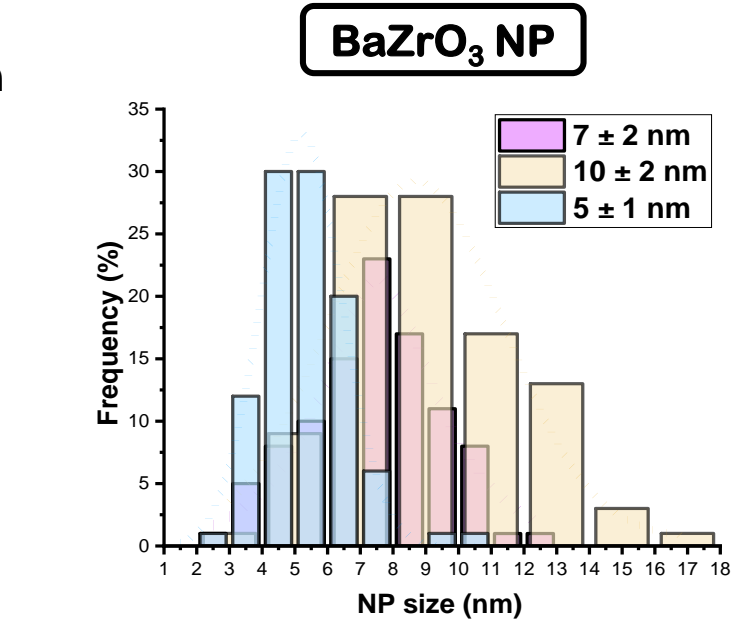
Hybrid Hydrolytic-Solvothermal Synthesis (H2S2) of Nanoparticles

1. Hydrolytic (sol-gel) step: nucleation



2. Solvothermal step: crystallization

Thermal activation

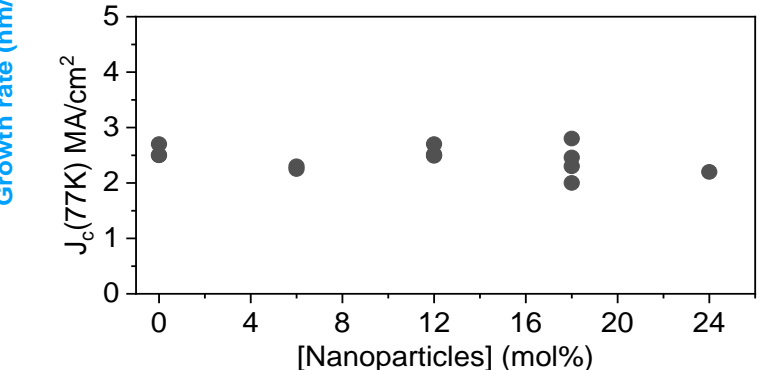
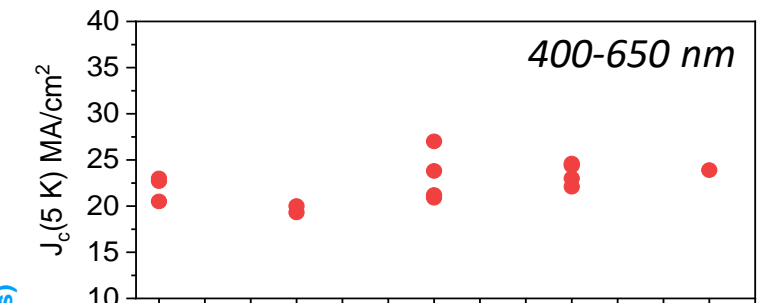
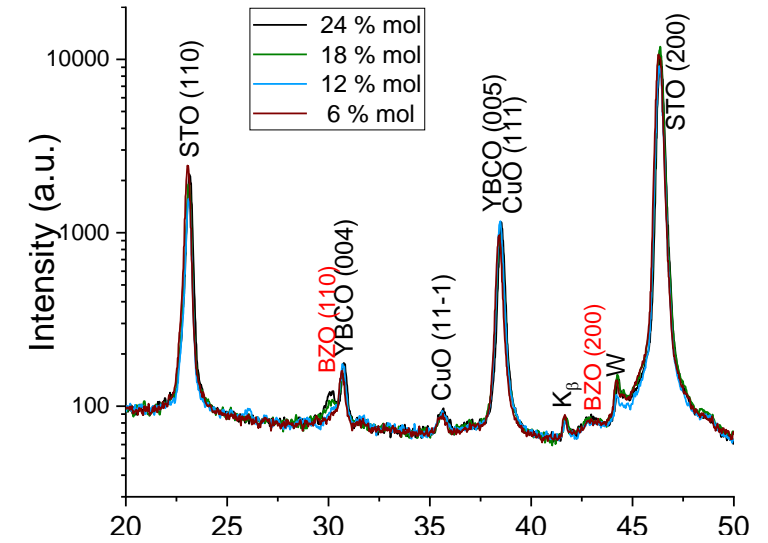
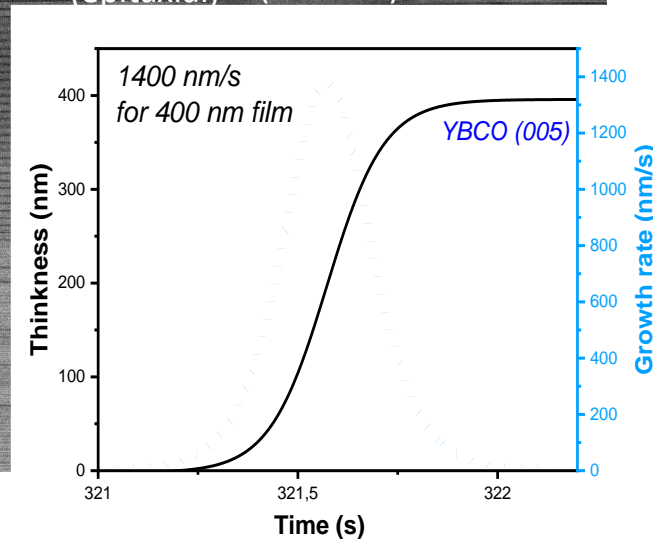
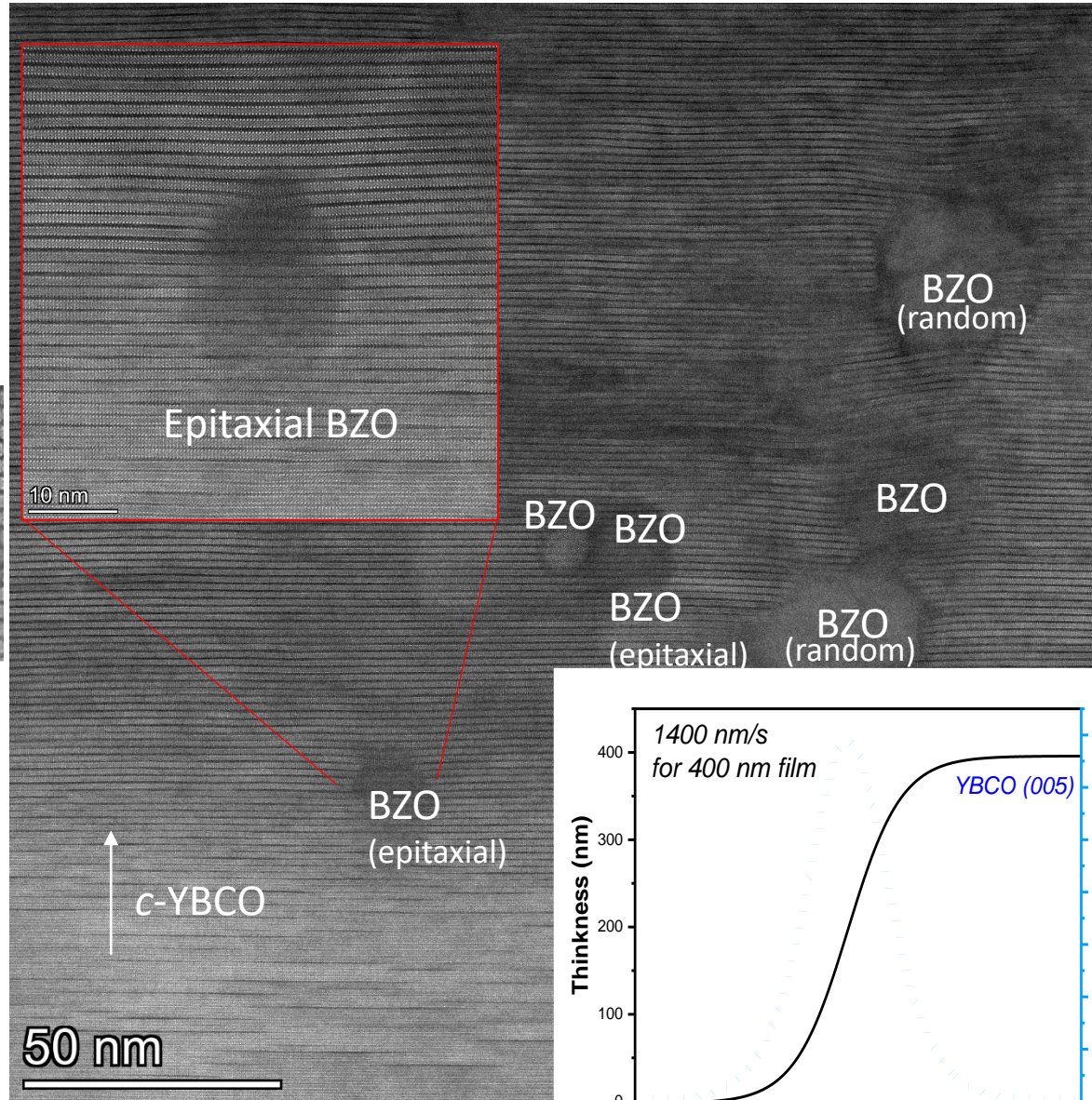
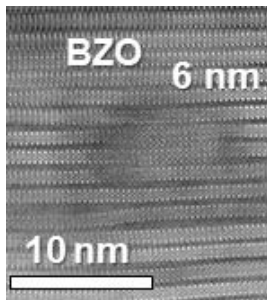
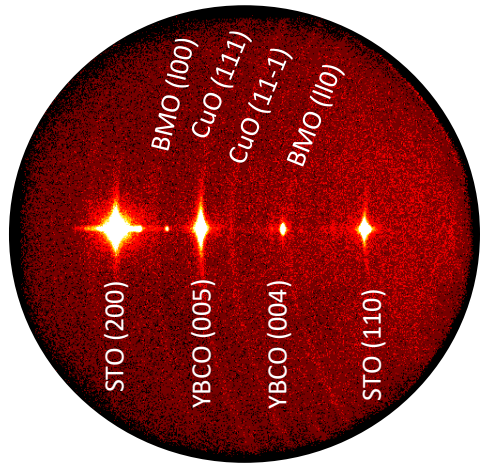


3. Stabilization in the REBCO precursor solution

Multifunctional colloidal ink
(Patent EP22382741)

Small crystalline NP with narrow size distribution at high concentrations (> 100 mM)
Stable in REBCO precursor solution for more than 3 months and scalable

YBCO TLAG-CSD NANOCOMPOSITE FILMS



Small NP are epitaxial (capability to rotate)
Larger Np are randomly oriented

Microstructure of TLAG pristine films at high growth rates

Main defects:

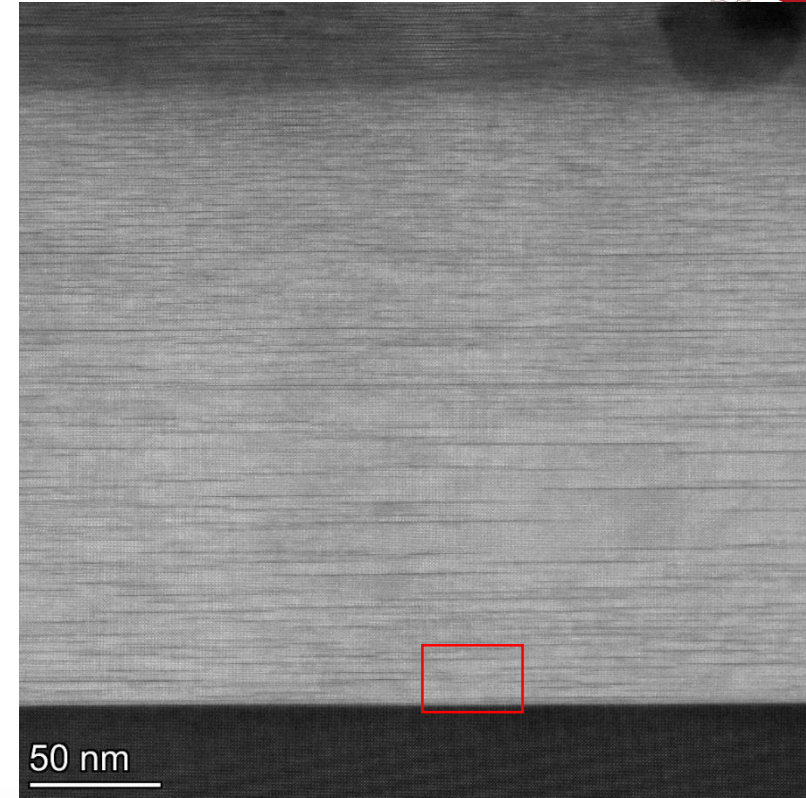
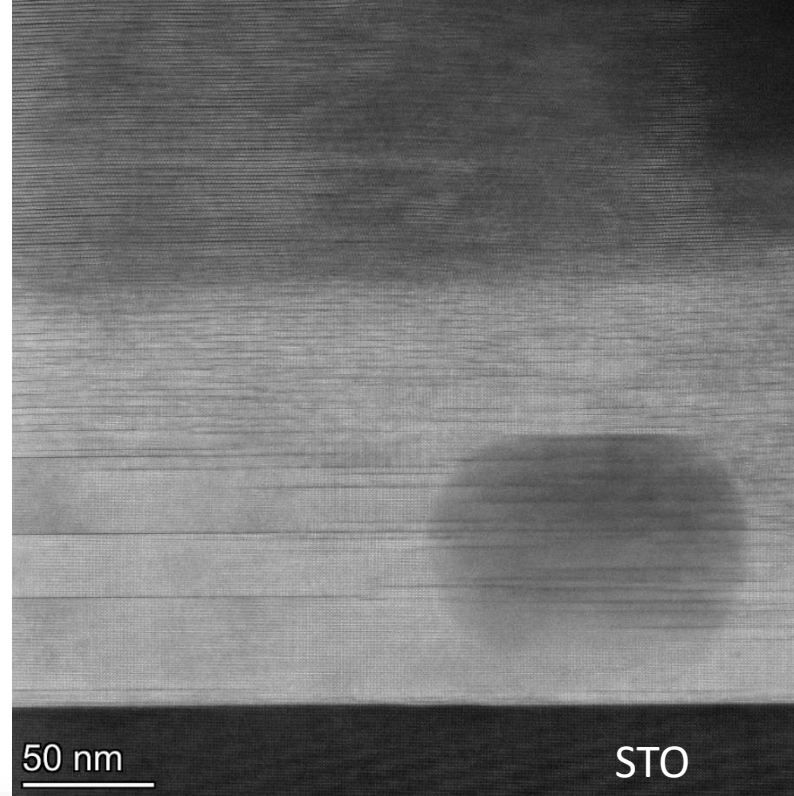
- SF
- partial disloc.
- strain
- TB
- oxygen and cluster vacancies

High density of **long** intergrowths (SF)

High density of **short** SFs

Low density of **long** SFs

STEM-HAADF



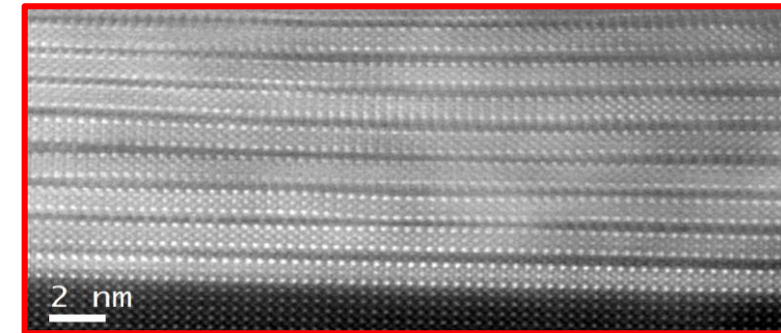
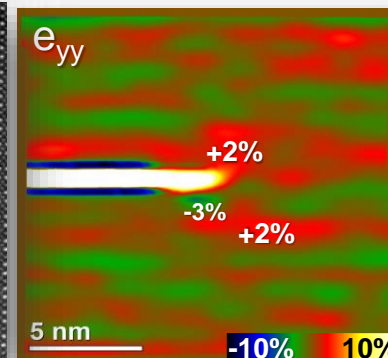
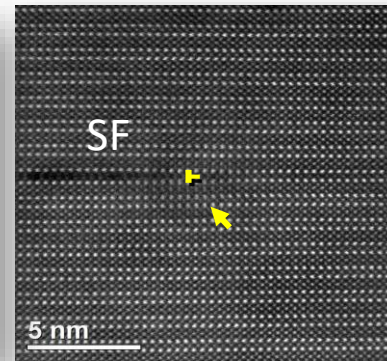
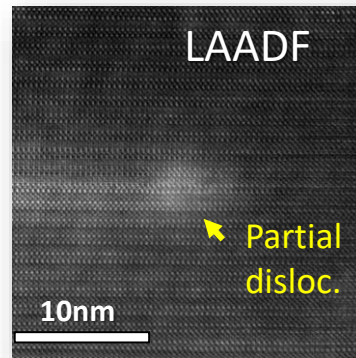
High density of **long** SFs

High density of **short** SFs

Medium density of **long** SFs

High density of **short** SFs

Strain accumulated at the partial dislocation surrounding the SF (**NANO**STRAIN)
 Also measured by XRD (Williamson-Hall)



Microstructure of TLAG nanocomposite films

Main defects:

- SF
- partial disloc.
- strain
- TB
- oxygen and cluster vacancies
- NP

High density of long SFs

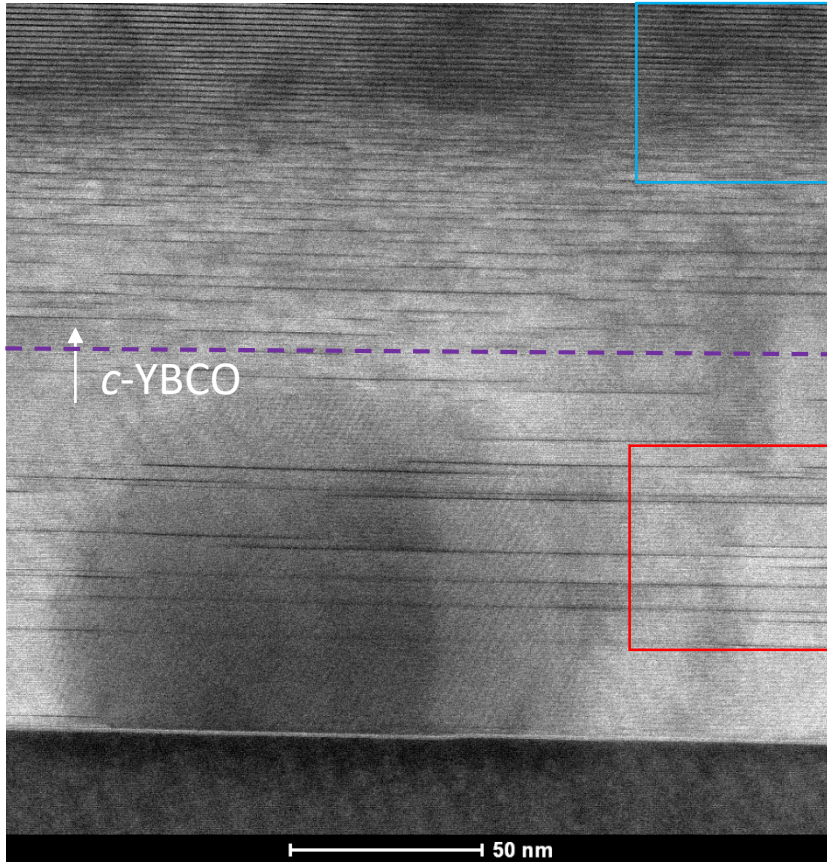
High density of short SFs

nanocomposite

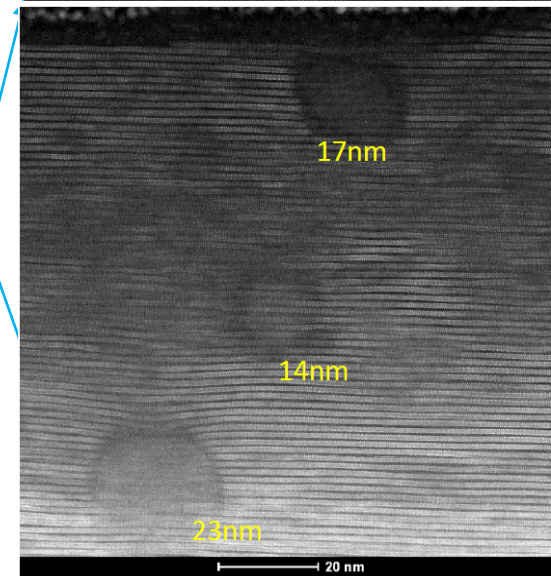
 pristine

Medium density of long SFs

STEM-HAADF



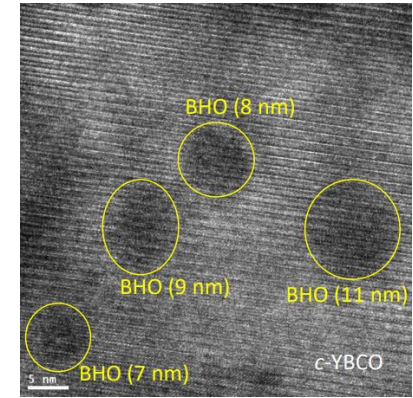
BZO NPs in YBCO 550 nm films



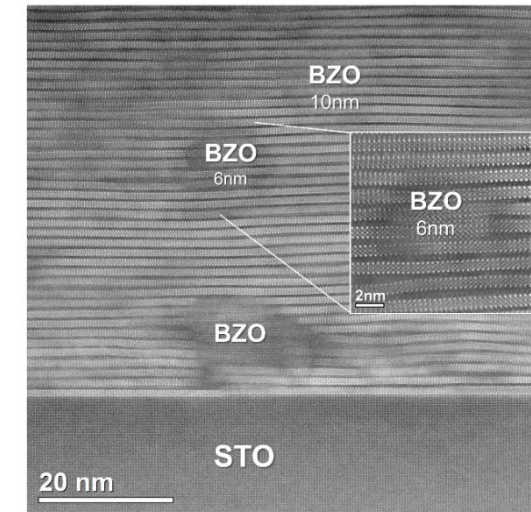
(14 nm BZO is epitaxial)

(23 nm is random)

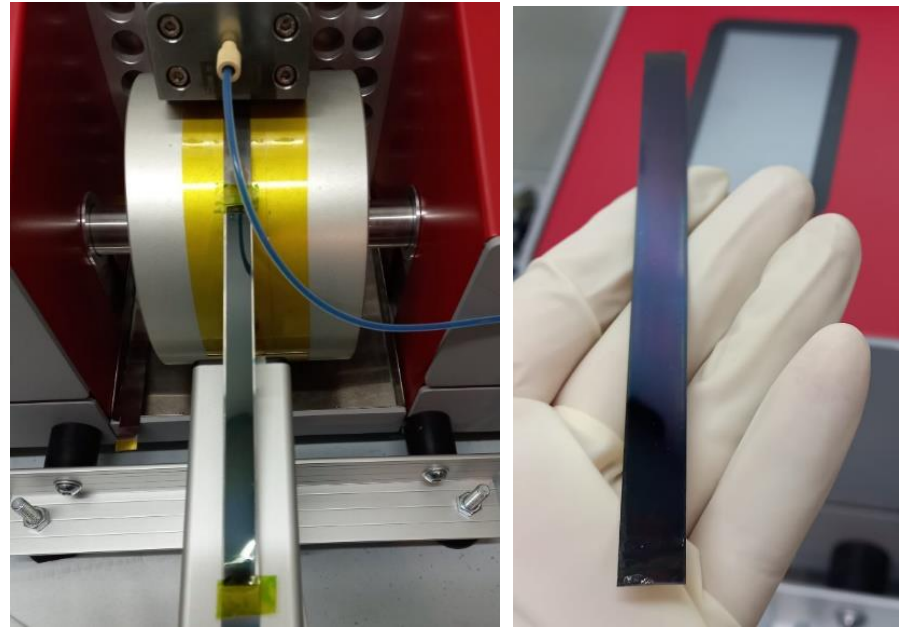
BHO NPs in YBCO



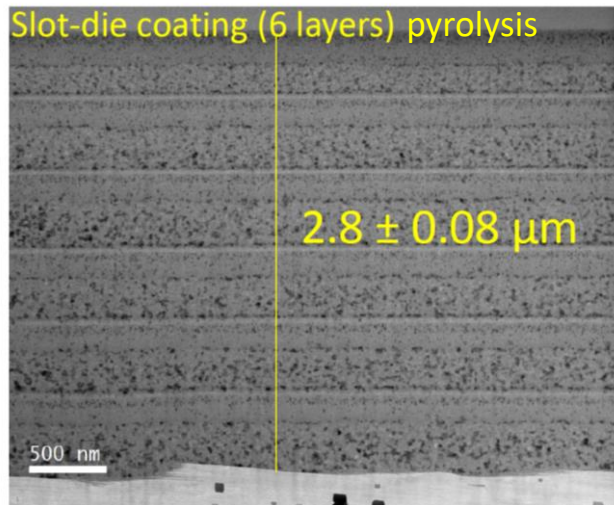
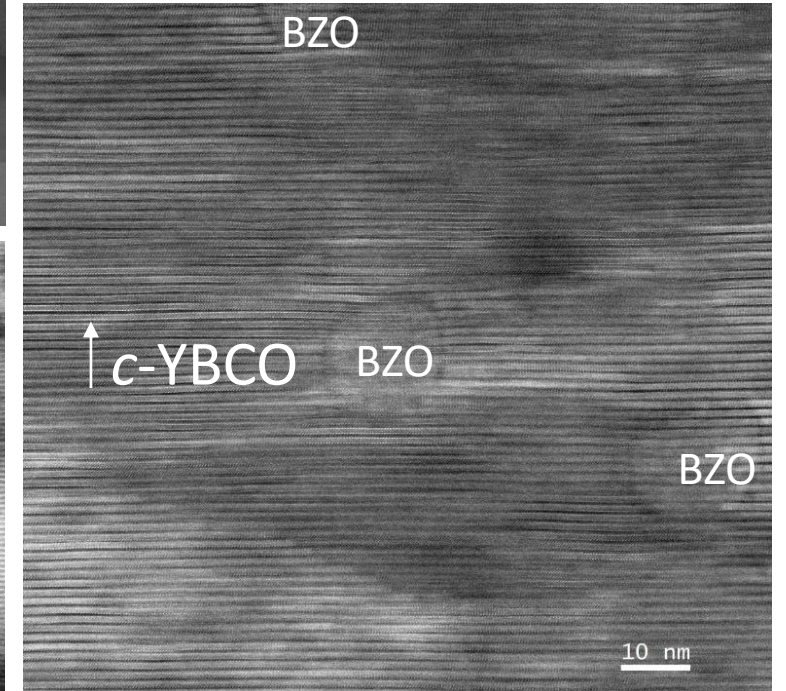
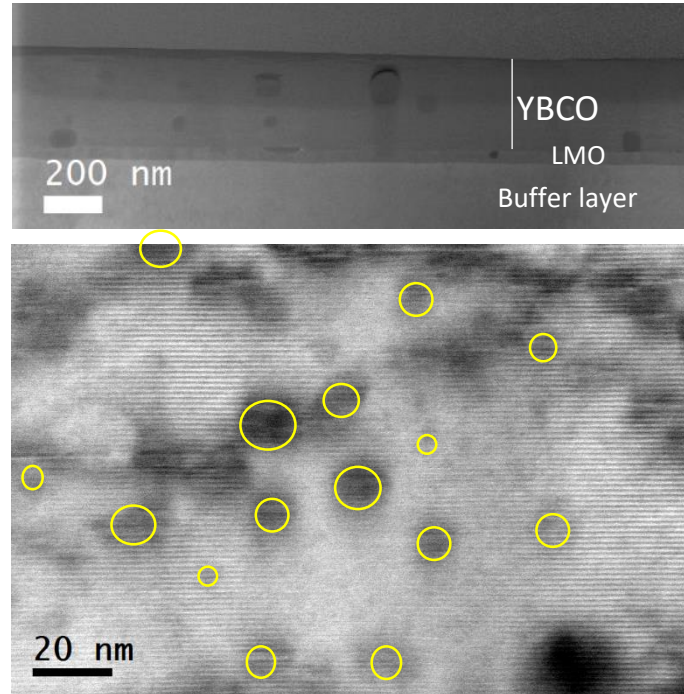
BZO NPs in YBCO



YBCO TLAG-CSD NANOCOMPOSITE CCs



Extended to technical substrates in collaboration with



Buffer layers

$J_c(5K) = 24 \text{ MA/cm}^2$, $J_c(77K) = 2 \text{ MA/cm}^2$ (450 nm)

$I_c(77K) = 130 \text{ A/cm-w}$ (750nm)

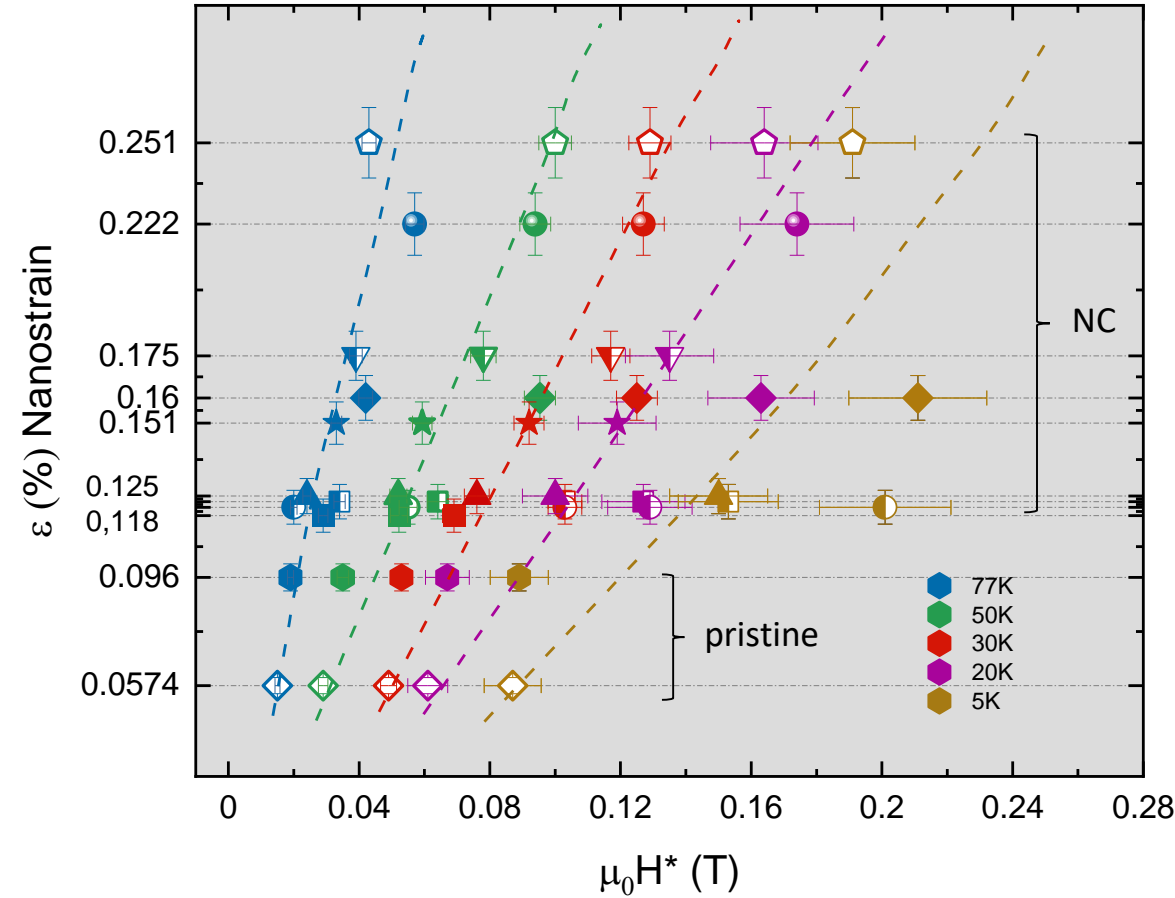
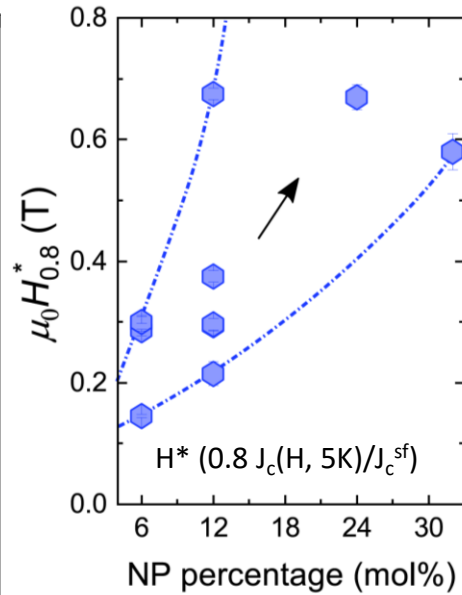
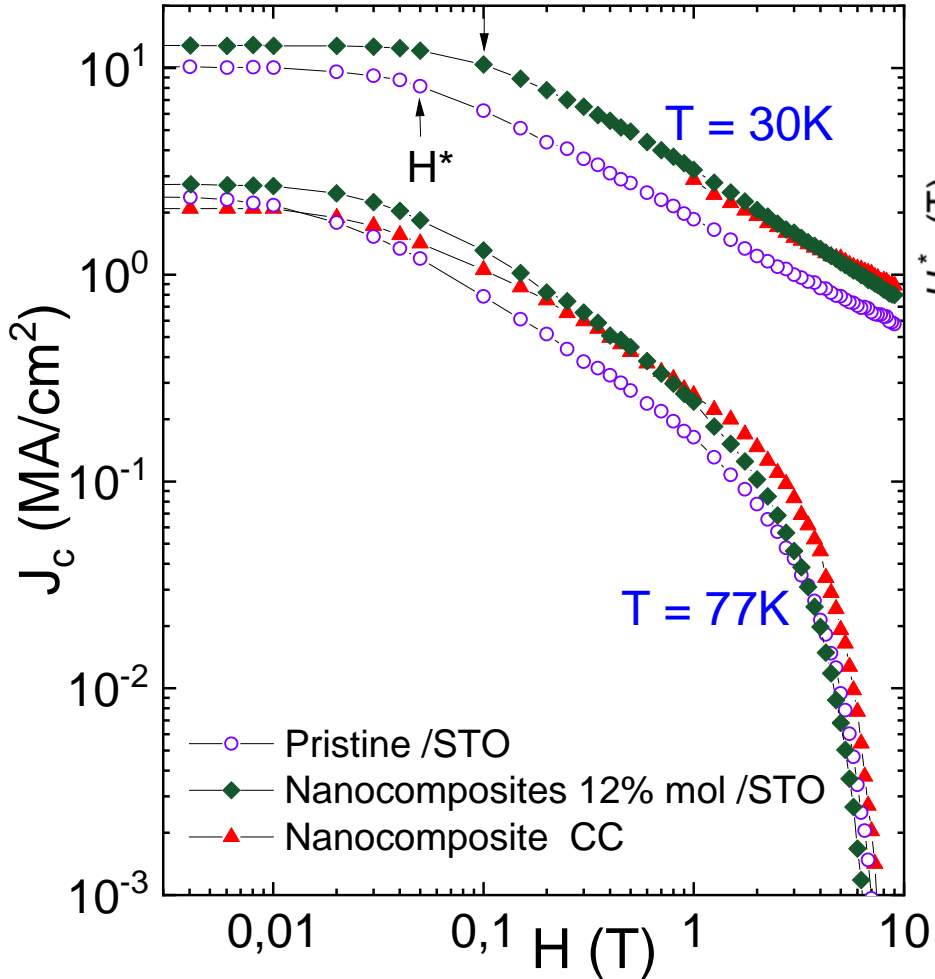
High density of defects is present as well as embeded nanoparticles

YBCO TLAG-PLD was extended to technical substrates in collaboration with



A. Quetalto et al, SUST (2023)

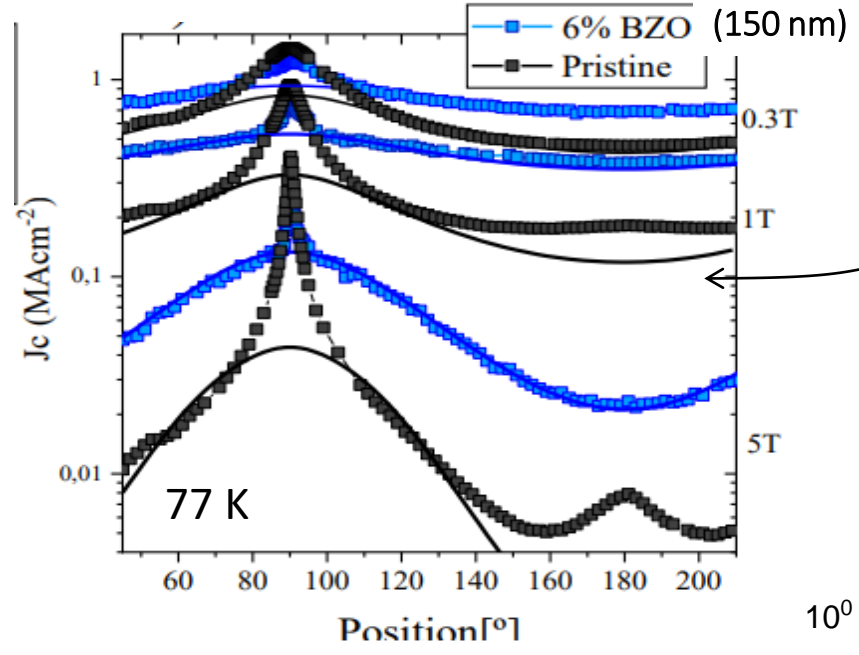
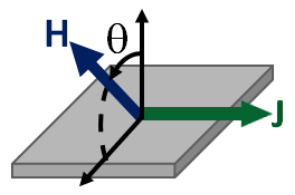
Vortex pinning consequences at high growth rate: TLAG-CSD



H^* : single vortex regime (*measure of the density of pinning centers*)

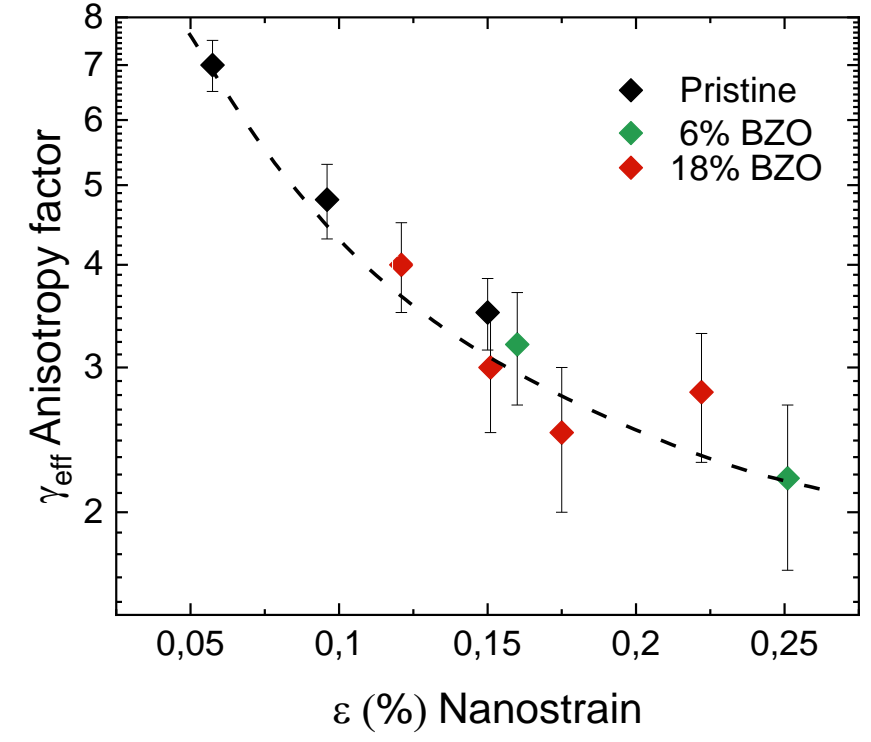
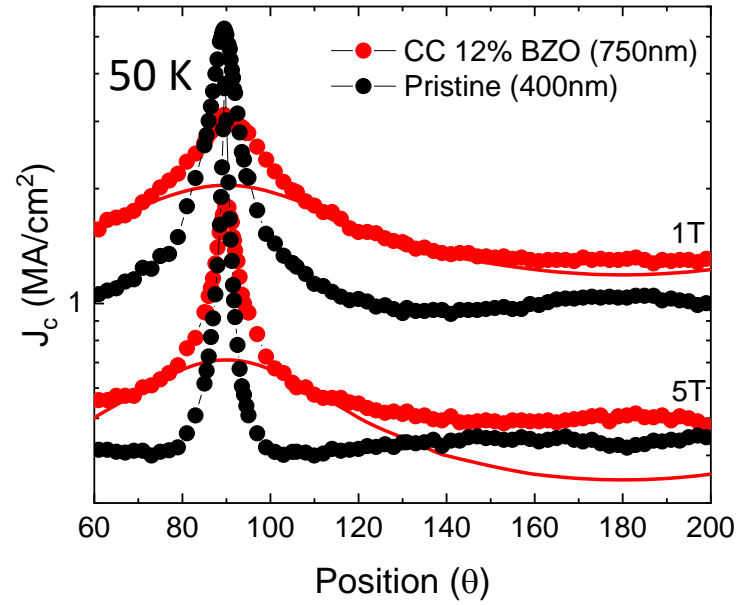
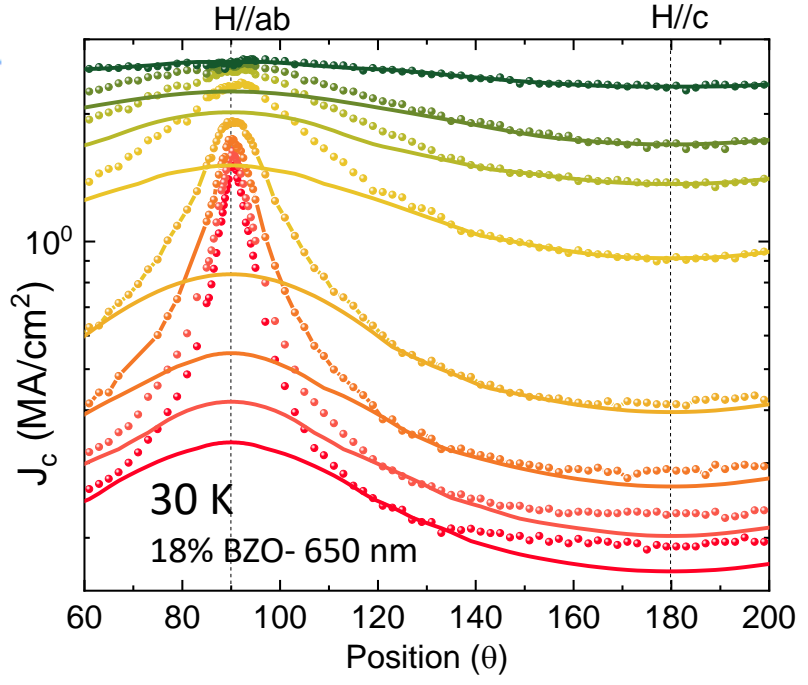
Strong correlation between **Nanostrain, nanoparticles and H^*** for many samples (pristine, nanocomposites (6- 18%) and several thickness)

Angular dependence of transport currents



Blatter scaling approach

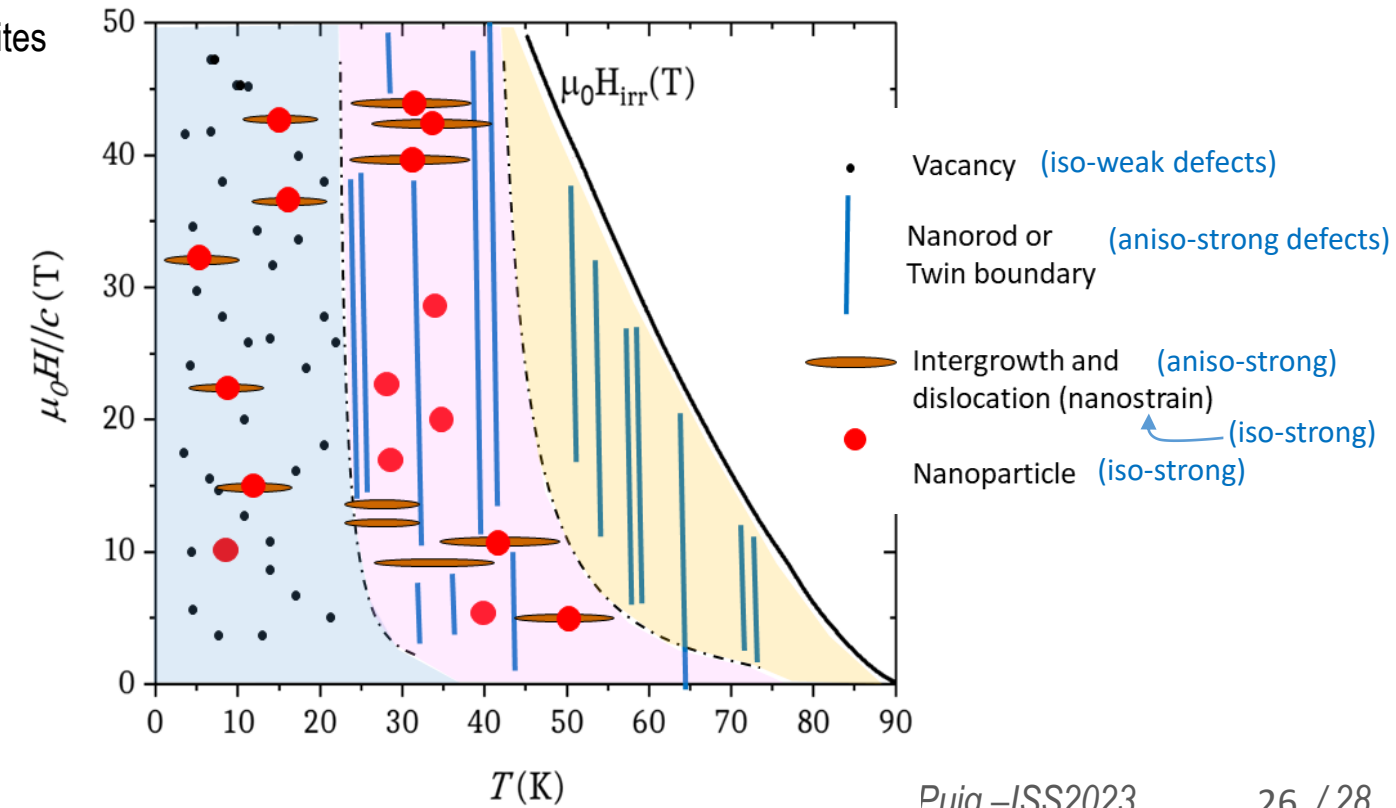
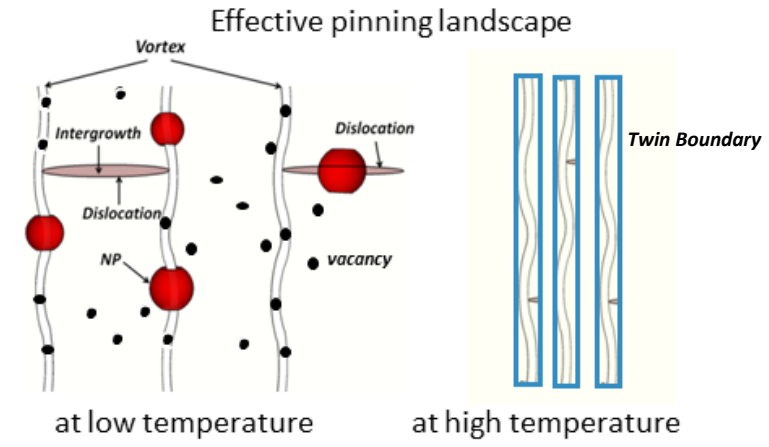
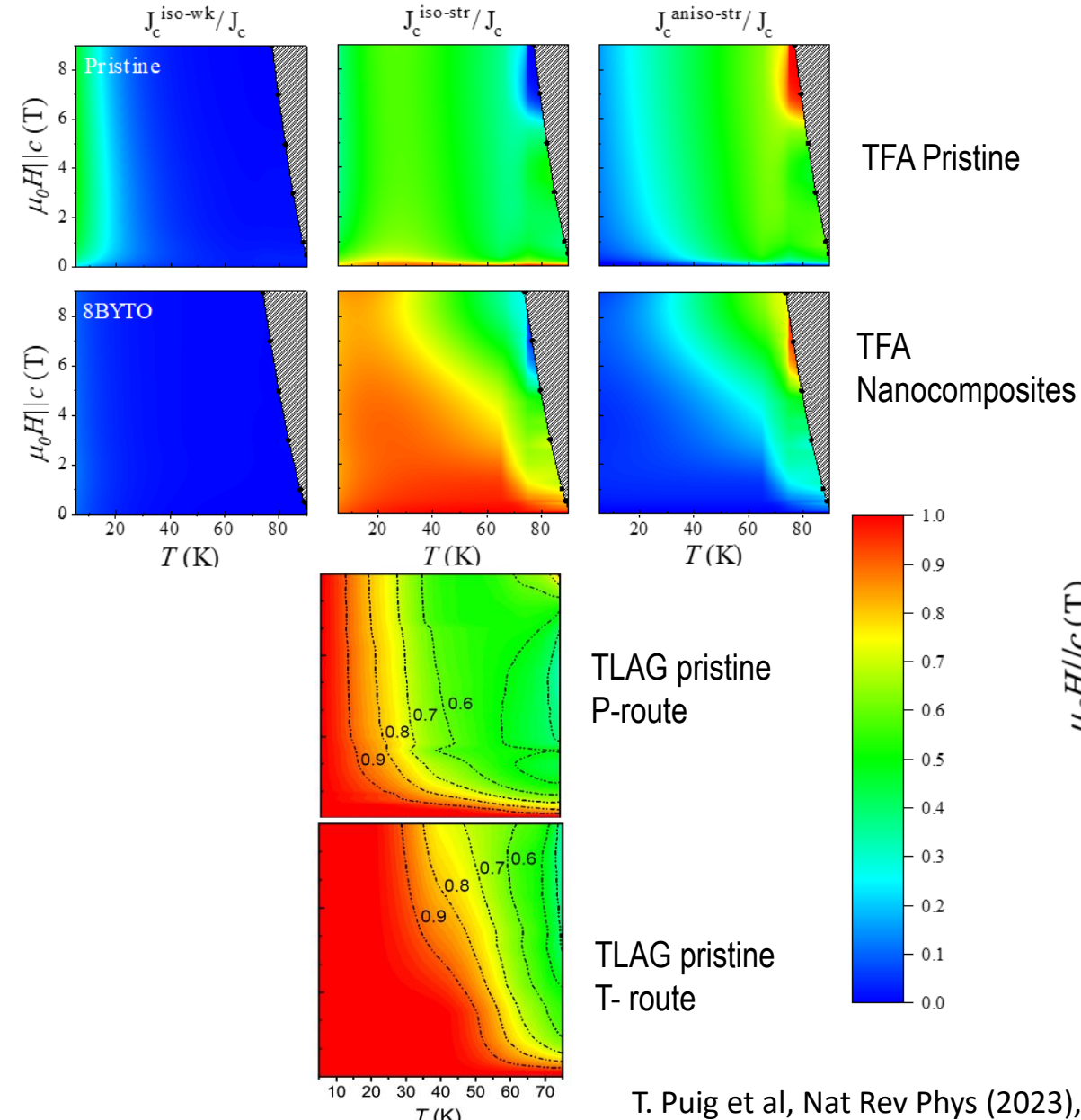
$$J_c^{iso} = J_c^{H//c} (\sin^2\theta / \gamma_{eff}^2 + \cos^2\theta)$$



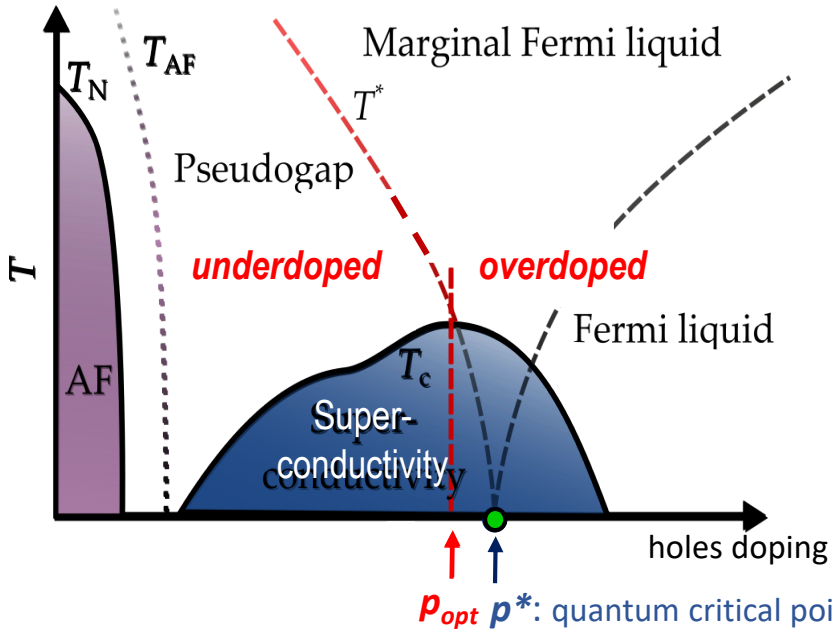
Strong correlation between effective anisotropy and Nanostrain

PINNING MODEL

$$J_c(T, H, \theta) = J_c^{\text{iso-wk}}(T, H, \theta) + J_c^{\text{iso-str}}(T, H, \theta) + J_c^{\text{aniso-str}}(T, H, \theta)$$

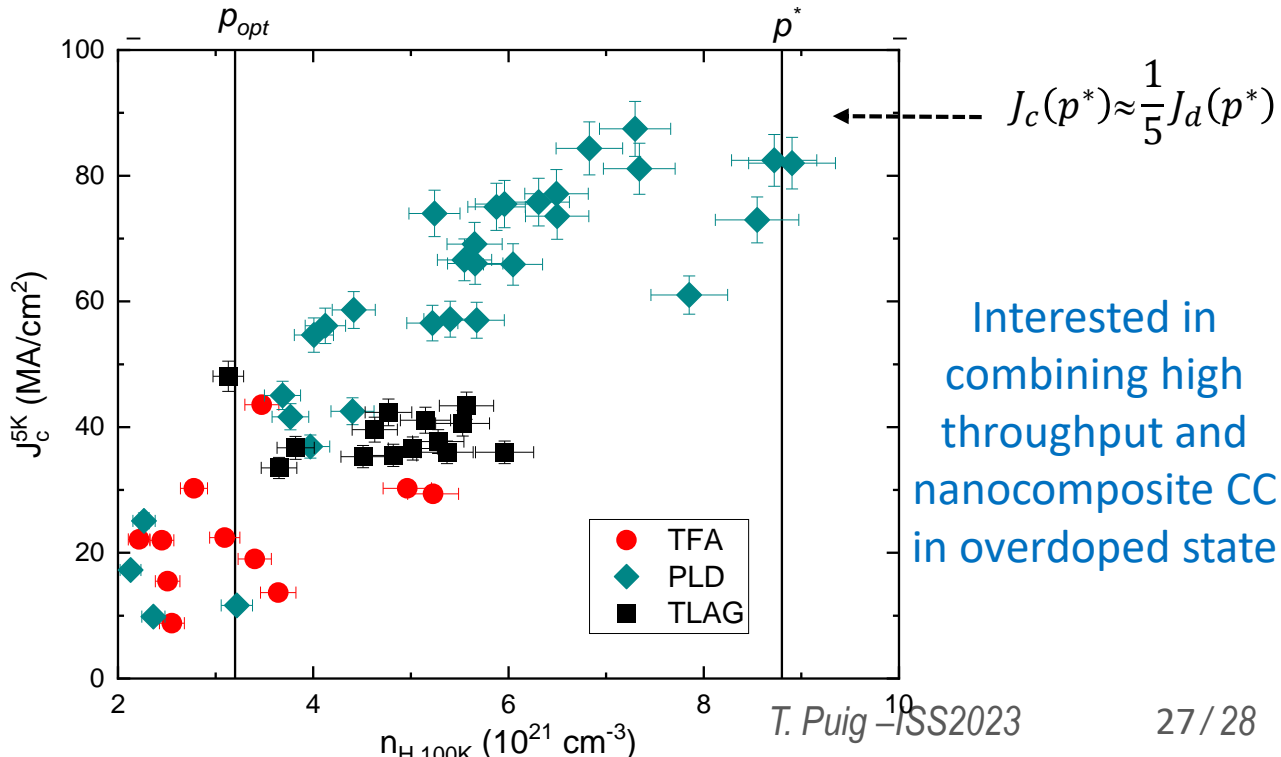
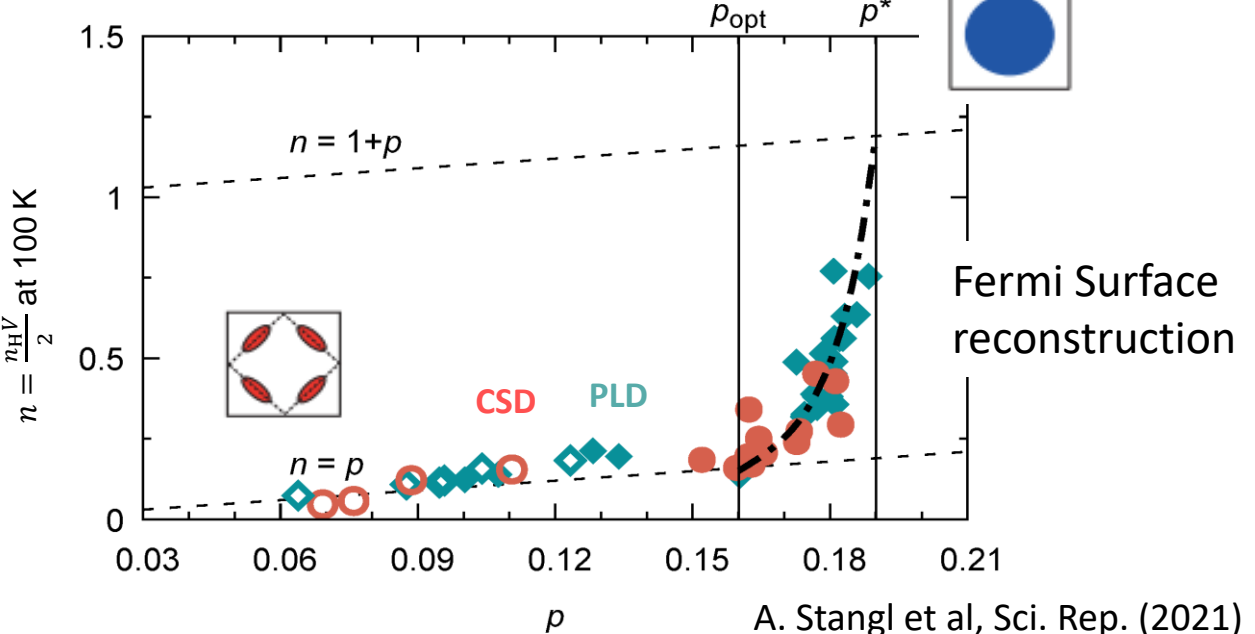
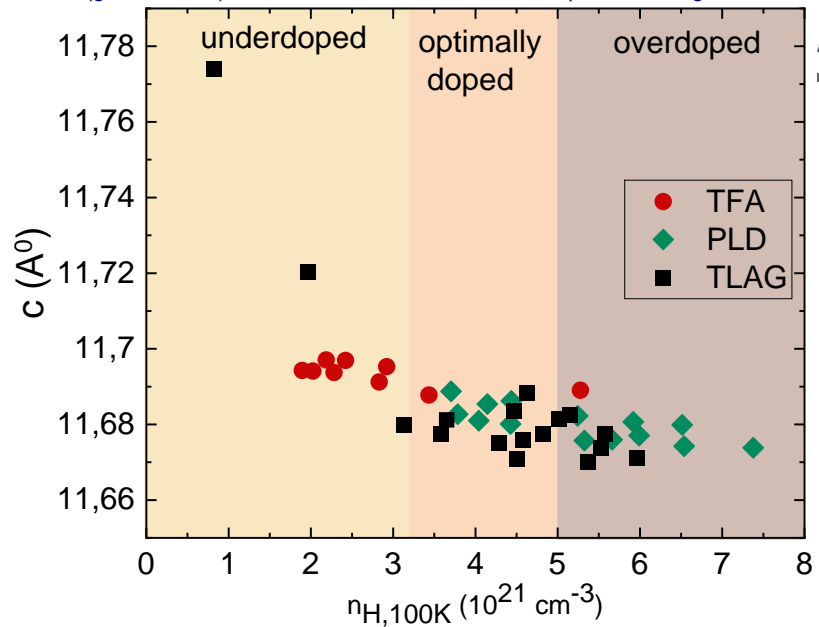


The Overdoped state



$$f_p \propto E_c \propto n_H$$

$$\text{If } n_H \uparrow \rightarrow J_c \uparrow$$

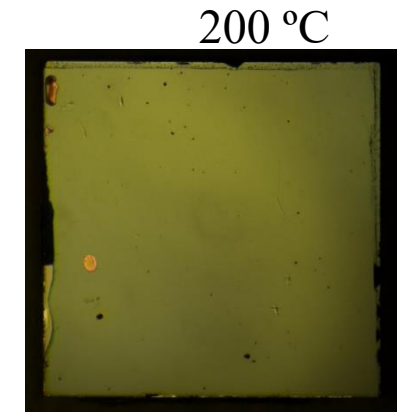
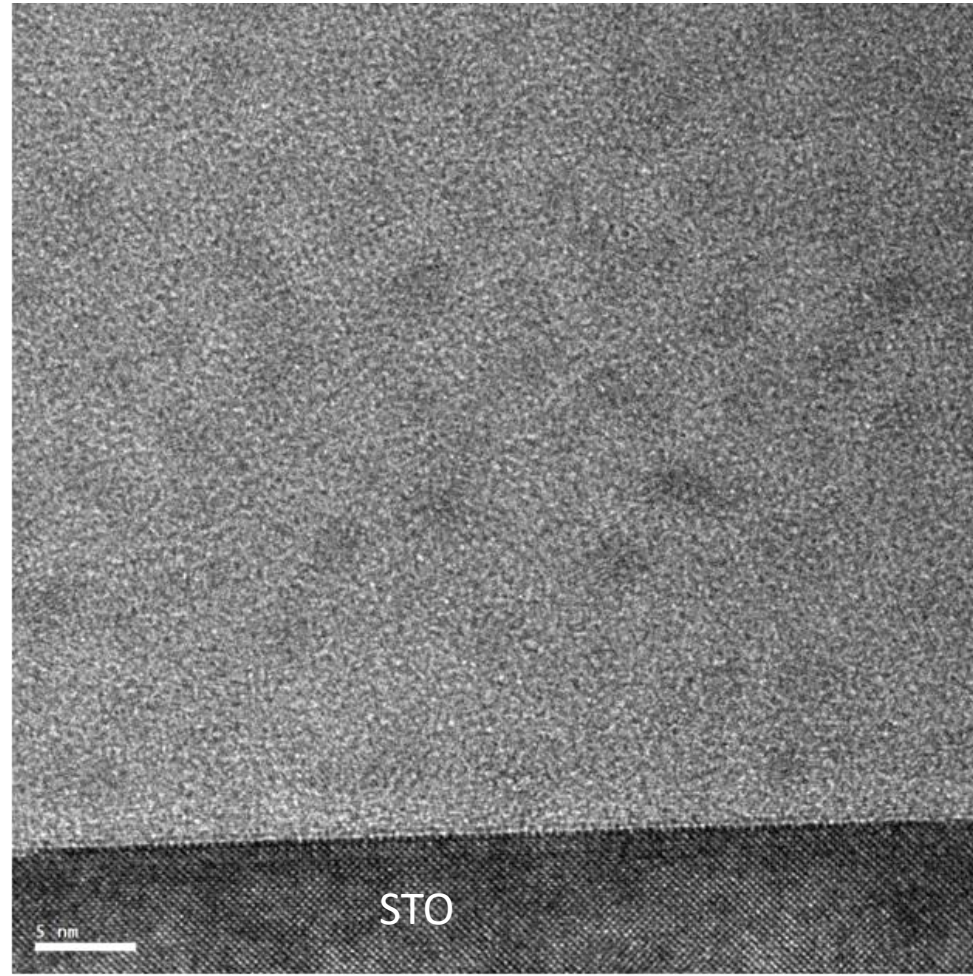
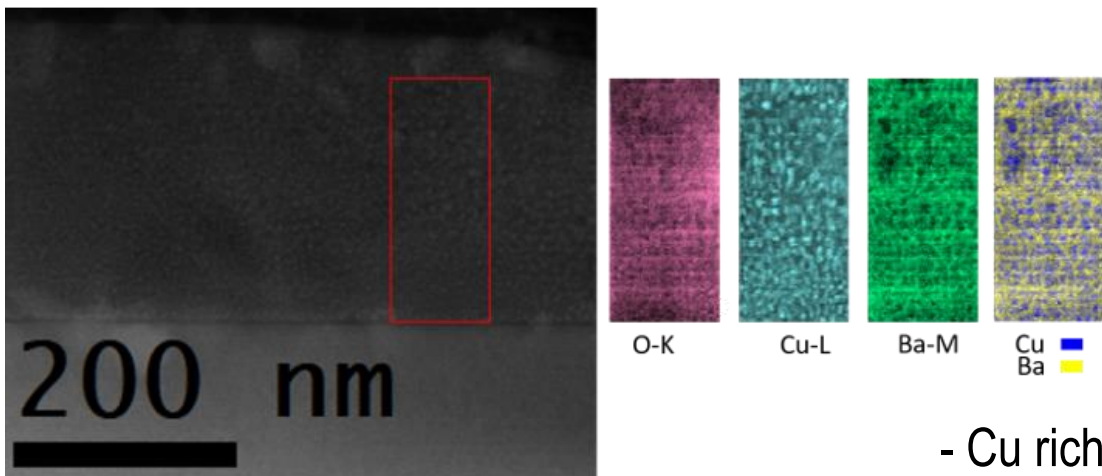
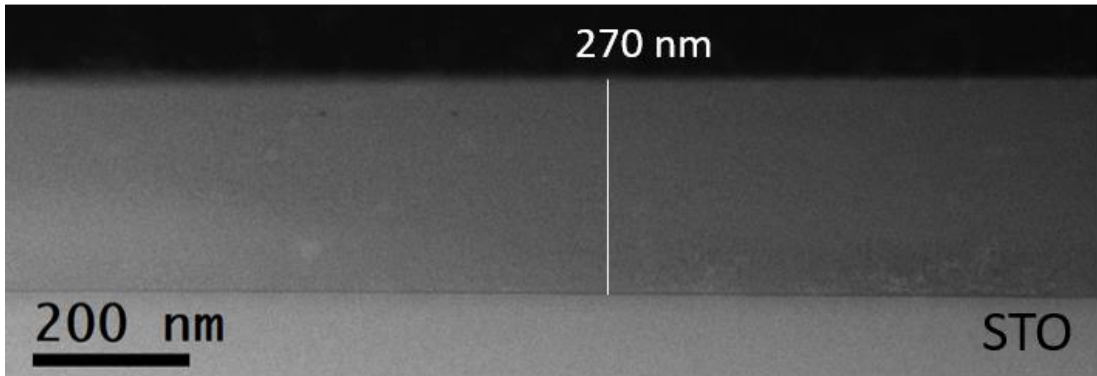
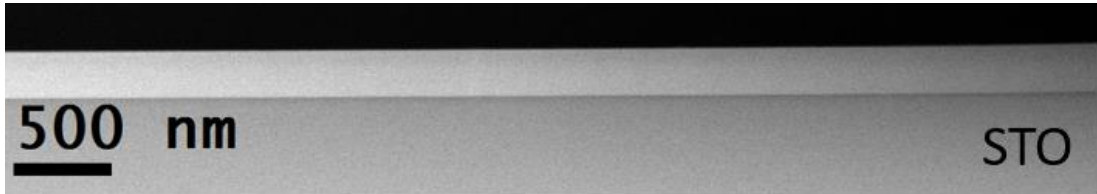


Interested in combining high throughput and nanocomposite CC in overdoped state

CONCLUSIONS

- **TLAG can contribute to make CCs a persistent enabling technology**
- Many opportunities/ challenges are being demonstrated
 - High throughput (high growth rate)
 - RE selection
 - High thickness
 - Nanocomposites
 - Coated Conductors
 - Rich vortex pinning
 - High performance
 - Overdoping
- Fast screening methodologies and machine learning should foster the High Throughput Experimentation required for its optimization
- **TLAG is an opportunity worth pursuing and expanding**

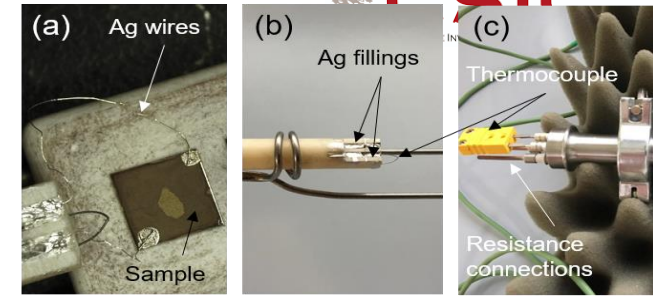
2. OXIDES PRECURSORS DEPOSITED BY PLD



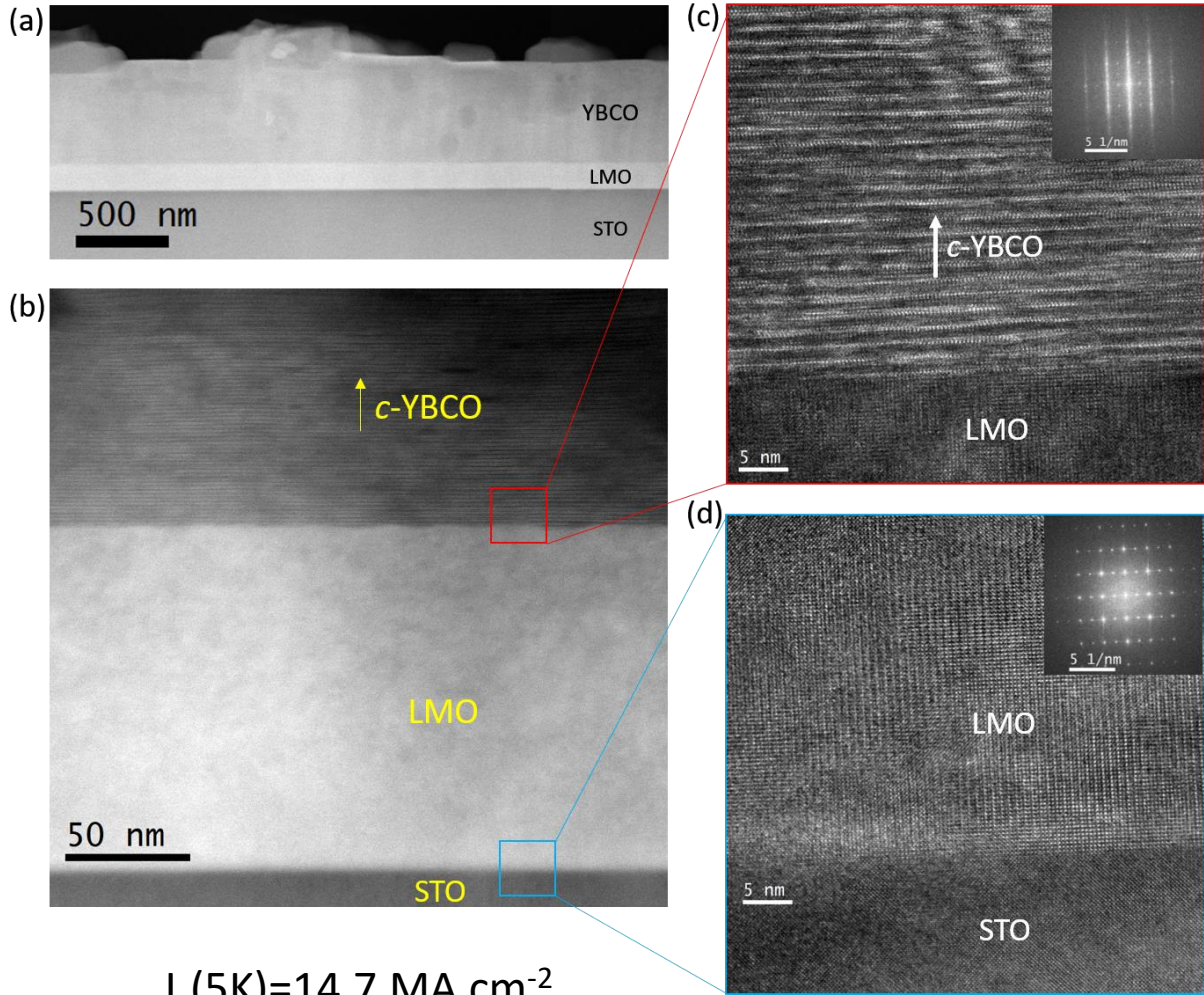
OM image

- Cu rich YBCO target
- Low temperature deposition (200-400°C)
- Amorphous and homogeneous PLD precursors

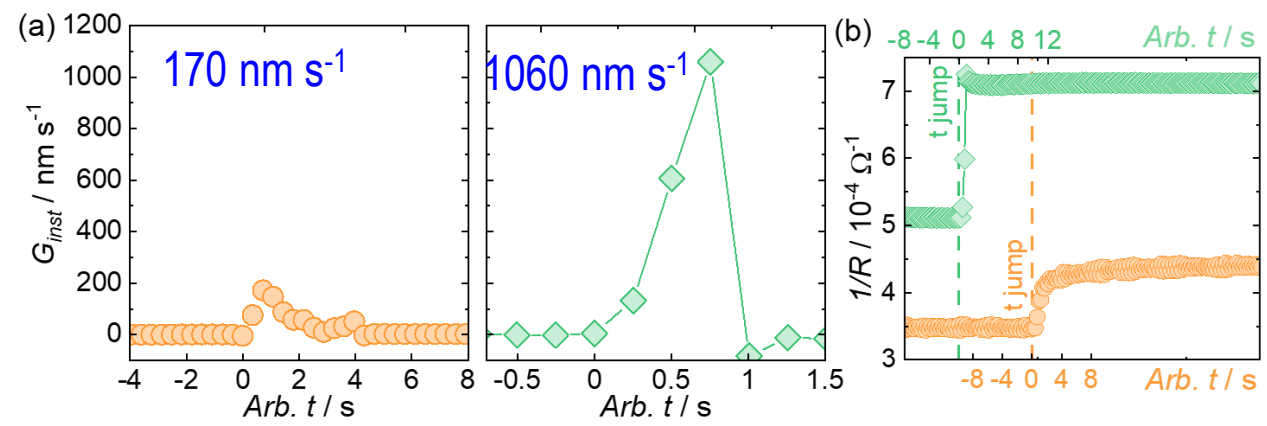
ULTRAFAST TLAG-PLD GROWTH



High defect density



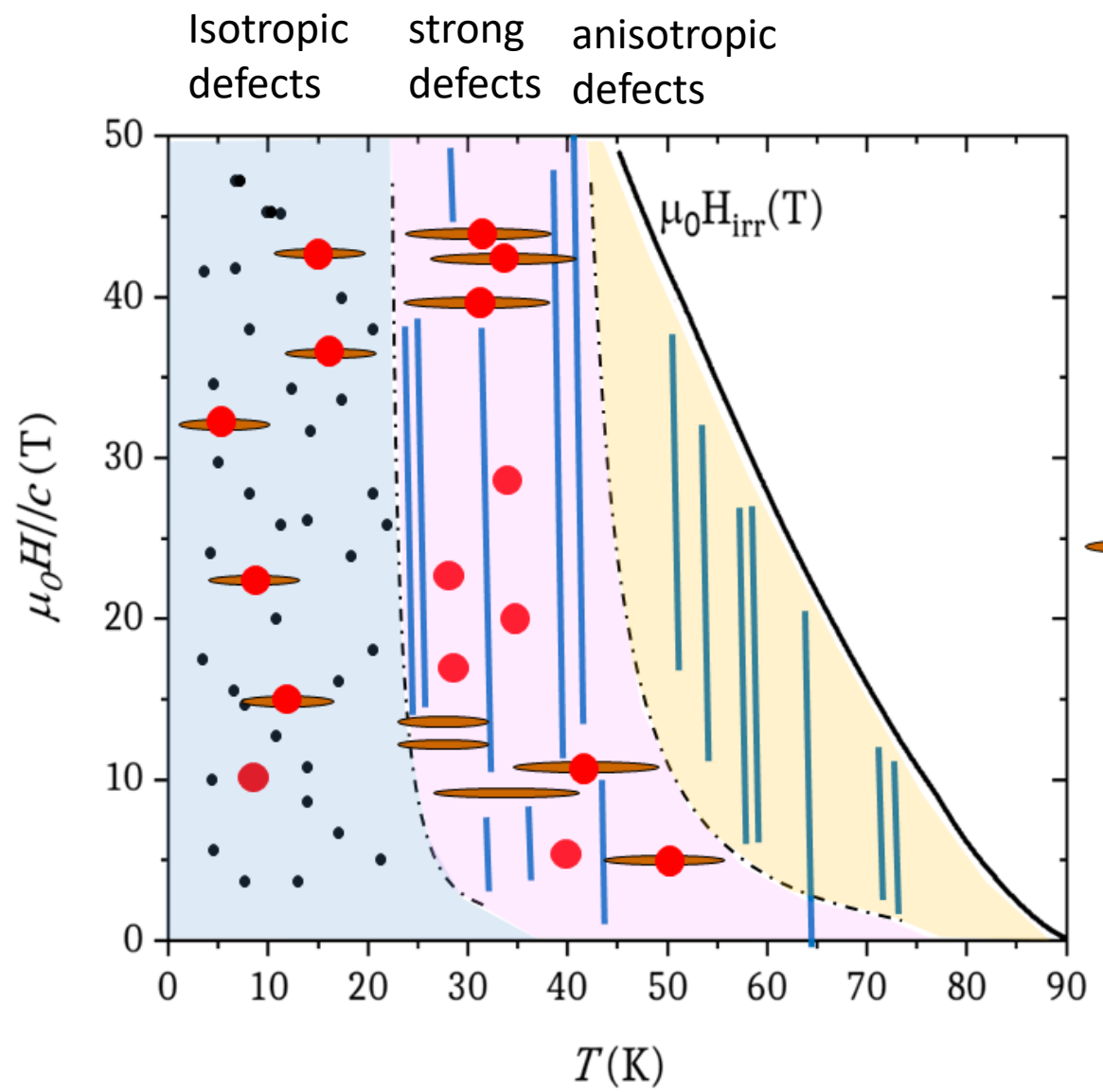
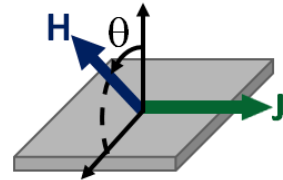
$J_c(5K) = 14.7 \text{ MA cm}^{-2}$
 $J_c(77K) = 1.7 \text{ MA cm}^{-2}$



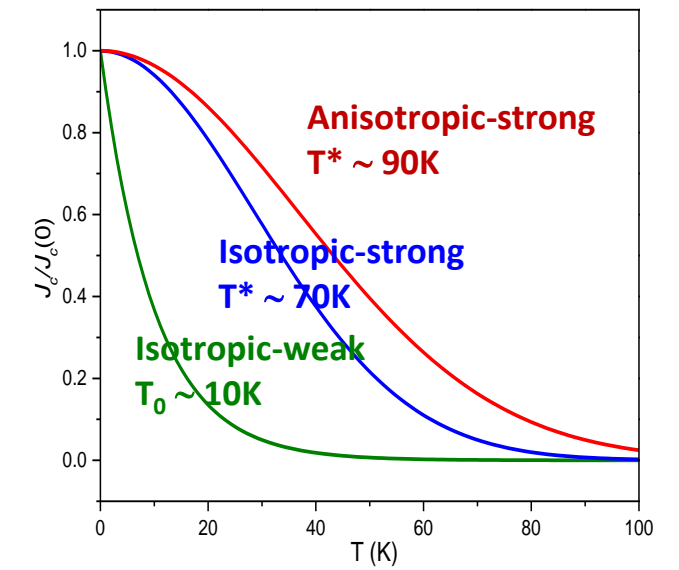
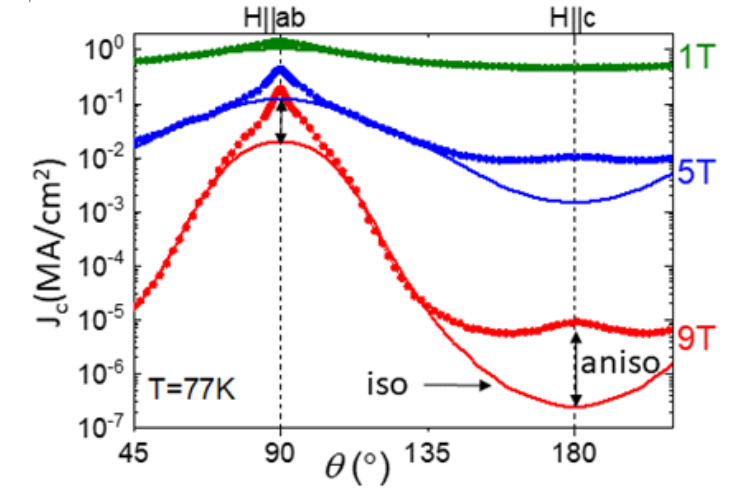
Ultrafast growth rate $\sim 1000 \text{ nm/s}$

Extended to technical substrates
 in collaboration with

The complex magnetic phase diagram of nanocomposites: correlation with microstructure



- Vacancy (iso-weak)
- Nanorod or Twin boundary (aniso-str)
- Intergrowth and dislocation (nanostrain) (aniso-str / iso-str)
- Nanoparticle (iso-str)



We need a simple route to classify APC for different processing methodologies