### Can Higher Critical Current of Powder In Tube (PIT) Nb<sub>3</sub>Sn be further developed without loss of RRR



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#### The Fraction Revel Estate of PIT

Small Grain A15 - 42%

Large Grain A15 - 13%

Remnant Core A15 - 2%

**Residual Core - 20%** 

- The diffusion barrier is fragile and the reaction must be carefully managed
- Only small grain A15 carries current.

# Are the large grains really contributing to current carrying capability?



LG A15 appears very disconnected

And here we can see Cu-rich phases penetrating between LG's

630C200h B29992 Longitudinal Cross Section

## How do we make **more current carrying** superconducting A15 phase?

### We must do this while maintaining...

- High Residual Resistance Ratio (RRR)
  - Keep diffusion barriers intact
- High grain boundary (GB) density
  - Produce small grains

#### Shape Analysis shows non-uniform deformation 1.18 1.16 1.14 **Aspect Ratio** 1.12 1.1 1.08 1.06 1.04 6 1 2 3 4 7 200 µm 1.04Ring Number 200 µm

### How RRR varies within filament pack

### -Serial etching experiment



### RRR varies depending on local Cu quality



There is clearly a zone more susceptible to leaks which is found in rings 6 & 7

Longer reaction time will not help! Where else can we make gains in J<sub>c</sub>?

### Two distinct filament types









### An intermediate Cu-Nb-Sn phase (Nausite) forms in internal tin strands<sup>[1]</sup>



Phase illustration by C. Sanabria

**[1] I. Pong**, L.-R. Oberli, and L. Bottura, "Cu diffusion in Nb<sub>3</sub>Sn internal tin superconductors during heat treatment," Supercond. Sci. Technol., vol. 26, no. 10, p. 105002, Oct. 2013.

### EDS results show Cu rich membrane

Residual Core

500 nm

element	AT%
Nb	71
Sn	23
Cu	4
Та	2

LG A15

Membrane

(likely ε phase)

 $\mathbf{X}$ 

	element	%	
<u>س ب</u>	Nb	95	
47	Sn	.5	676
a 14.	Cu	4.5	8
$\sim l_{c}^{2}$	Та	0	
100	ALC: NOT THE OWNER WATER		

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SG A15

#### Comparison of typical and atypical filaments



9	Nb	18.2	
	SG A15	46.3	
0	LG A15	10.2	
	Core	25.3	
	1.1		
	element	% chai	nge
	element Nb	% chai -20	nge
	element Nb SG A15	% chai -20 +12	nge
C	element Nb SG A15 LG A15	% chai -20 +12 -25	nge

element

**AT%** 

12% increase in the good stuff!

#### Comparison of end phase fractions of PIT and RRP

Condition and Component %	54/61 RRP <sup>®</sup> (Tarantini SuST 2014)	PIT 192 (CS evaln 2013)	PIT 192 (CS evaln 2014 –atypical filaments
HT	620C, 192h	650°C, 100 h	630°C, 240 h
RRR	377	177	177
A15 total %	58.8	56	56.7
A15 SG %	58.8	40	46
A15 LG %		16	10.7
Residual DB %	8.1	24.5	17.6
Residual core %	33.1	19.5	25.7

\* = 169 and 217 RRP<sup>®</sup> stacks need ~10% residual DB

### Increase in SG A15 without loss of RRR!

### Findings for discussion

- A Cu rich membrane to mediate the reaction appears VERY useful
  - Why does this membrane form and why does it positively affect A15 growth?
- The issue better control the reaction path:
  - Can we avoid large grain A15?
  - Consume all present Sn in package to produce current carrying SG A15
- The payoff?
  - Typical filaments have a 3:1 SG:LG ratio
  - Untypical filaments have a 4:1 SG:LG ratio!
  - Even more valuable, the total amount of SG A15 is enhanced from an average of 41% to 46%, a 12% increase surely beneficial for  $J_c$

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