



Advances in Nanoscale Analysis of Hf doped Nb₃Sn wires using Atom Probe Tomography

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Requirement: High Performance Nb₃Sn for FCC

- Operational J_c in Nb₃Sn superconductor: 1500 Amm⁻² and RRR > 150 at 16 T (4.2 K)
- Needs radical improvements in performance of Nb₃Sn filaments



Schematic of the FCC www.home.cern/science/accelerators/future-circular-collider

Optimising the Superconducting Properties

 For high J_c at 16 T the pinning force function will require both grain boundary and secondary point pinning

$$F_P = A_{GB} \left(\frac{H}{H_{IRR}}\right)^{0.5} \left(1 - \frac{H}{H_{IRR}}\right)^2 + A_{PD} \left(\frac{H}{H_{IRR}}\right) \left(1 - \frac{H}{H_{IRR}}\right)^2$$

- Grain boundaries and optimised point defects are on the scale of the coherence length (3-4 nm in Nb₃Sn)
- Nanostructural analysis is required to visualise grain boundaries, secondary phases and local chemistry changes on this scale

 $\frac{m}{n} \approx 2eV \left(\frac{t_{flight}}{L}\right)^2$

What is Atom Probe Tomography?

- 3-Dimensional characterisation technique
 - High spatial and chemical resolution
 - Sensitivity down to ppm

Hf



Additions for Point Defects

- Addition of Group IVB elements Zr and Hf to produce oxide nanoparticles [1,2]
- Ta known addition to increase the upper critical field
- Obvious shift to point pinning function seen in Hf doped sample

1.0 - (b) 3.01 K 4.2 K 6 K 0.8 8 K 10 K 12 K ,/F_{pMax} 9.0 Nb-Ta4 sample 14 K **GB** function – PD function 0.2 0.0 2 H/H_{Max} 1.0 - (b) 3.01 K 4.2 K - 6 K 0.8 8 K Nb-Ta4-Hf1 10 K р/F_{рМах} 90 12 K - 14 K sample **GB** function -··- PD function 0.2 0.0 2 H/H_{Max}

[1] S. Balachandran et al. 2019 Supercond. Sci. Technol. 32 044006
[2] X. Xu et al. 2020 Scripta Materialia 186 317-320.
[3] C.Tarantini et al .2019 Supercond. Sci. Technol. 32 124003

Sample studied in this work



BSD image of reacted wire



Are nanoparticles present in the Nb₃Sn region?

Atom Probe Tips from Nb₃Sn layer



Tip 3

20 nm

Nb₃Sn region: Location of oxides



- Cu is located at grain boundaries
- Additional isolated Cu regions are present within grains

Oxide Source

- <u>No oxygen</u> added to the alloy
- Are HfO₂ nanoparticles present within the Nb-Ta-Hf alloy before reaction?



Pre-heat treatment

Post-heat treatment

Pre-Heat Treatment Nb-Ta-Hf alloy: Hf distribution









Post Heat Treatment Nb-Ta-Hf alloy

Now the unreacted regions of metallic alloy do contain HfO₂ nanoparticles



Post-heat treatment



Oxygen Content

Pre-heat treatment Nb alloy (Oxygen at%)	Post-heat treatment Nb alloy (Oxygen at%)	Nb ₃ Sn (Oxygen at%)
2.59	3.66	0.49
1.95	3.61	0.87
3.83	1.36	0.16

Oxygen content in pre-heat and post-treatment Nb-Ta-Hf alloy is very similar, with far lower oxygen in the Nb₃Sn layer which is confined to purely oxide clusters.

We can also study the reaction process



Residual Nb₆Sn₅



- Nb₃Sn shown with the dark green surface
- The rest of the tip is Nb₆Sn₅

Nb₃Sn

Nb₆Sn

• HfO₂ clusters seen in Nb₆Sn₅ as well as the Nb₆Sn₅

TEM image of Nb₃Sn-Nb₆Sn₅

> S. Balachandran et al. 2019 Supercond. Sci. Technol. 32 044006

Cu preferentially partitioned into the Nb₆Sn₅

C. Segal, et al. IOP Conference Series:

Materials Science and Engineering. Vol. 279. No. 1. IOP Publishing, 2017.

Next Steps: Comparison of Oxygen content to a Commercial Alloy

Pre-heat treated Monofilament Nb-4Ta-1Hf alloy (at%)	Commercial Nb-4Ta-1Hf alloy (at%)
2.59	2.17
1.95	1.09
3.83	1.04

2 µm

BSD image of a commercial alloy

- Evidence of larger HfO₂ precipitates in the commercial alloy
- Next step is to compare oxygen content across alloys using EPMA

Conclusion

- HfO₂ nanoparticles are found in the post-heat treatment Nb-Ta-Hf alloy, the Nb₆Sn₅ and the Nb₃Sn layer
- The oxygen was originally dissolved in the Nb-Ta-Hf alloy, leading to the formation of these oxides during heat treatment
- Nanoscale Cu islands are also present in the Nb₃Sn (and may contribute to pinning)
- Nb₆Sn₅ contains a larger concentration of Cu than the Nb₃Sn that forms from it (approx. 2at%)
- Different Nb-Ta-Hf alloy compositions can be compared to determine the best starting material for producing optimal superconducting properties

