



## Recent progress in Fe-based superconducting wires and tapes

#### Yanwei Ma

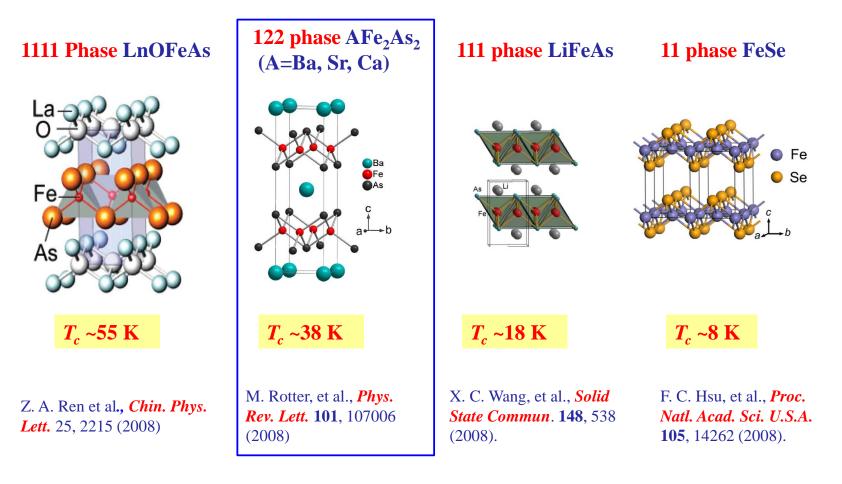
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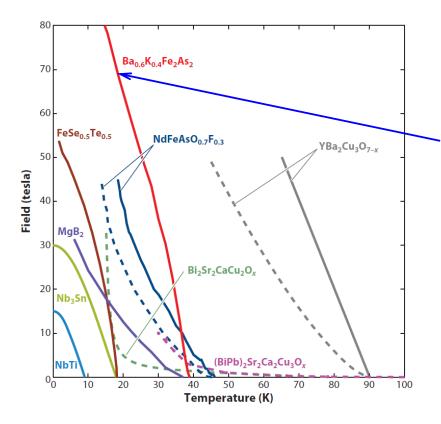
- **1** Background on Fe-based superconductor
- **2** Strategies to improve Jc in 122 wires
- **3** Fabrication of practical 122 pnictide wires
- 4 Scaling up to the first 100-m-class 122 wire
- **5** Conclusions

### **Main known Fe-based superconducting families**

Among them, the three phases most relevant for wire applications are 1111, 122, and 11 types with a  $T_c$  of 55, 38 and 8 K, respectively.



## The extremely high $H_{c2}$ in IBS



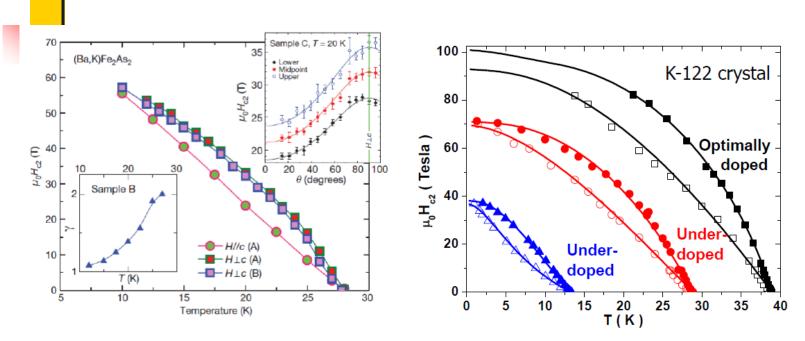
Gurevich, Nature Mater. 10 (2011) 255

At 20 K, the  $H_{c2}$  can be >70 T where IBS outperform both MgB<sub>2</sub> and Bi-2223.

- Interesting FBS have T<sub>c</sub>: 38-55 K
   > Nb-Ti and Nb<sub>3</sub>Sn
- Operation at 4K >20T or 10-30 K at >10 T would be very valuable

The extremely high  $H_{c2}$  in IBS shows a great potential for applications in high field magnets, e.g., H > 20 T, which cannot be achieved via LTS and MgB<sub>2</sub>.

## **122 IBS** - small anisotropy γ



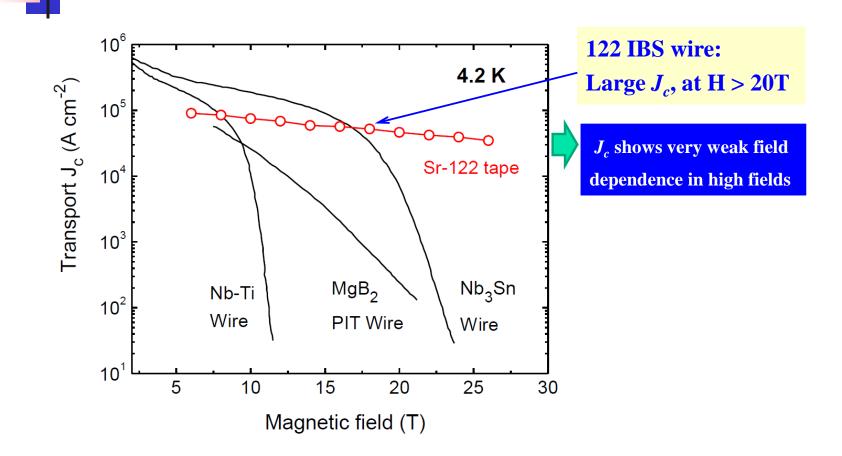
Yuan et al. Nature 457, 565 (2009)

Tarantini et al. PRB 86, 214504 (2012)

#### γ ~1.1 for K-122, nearly isotropic

γ is almost 1, clearly, vortices are much more rigid than in any cuprate-much easier to prevent depinning of any GB segment

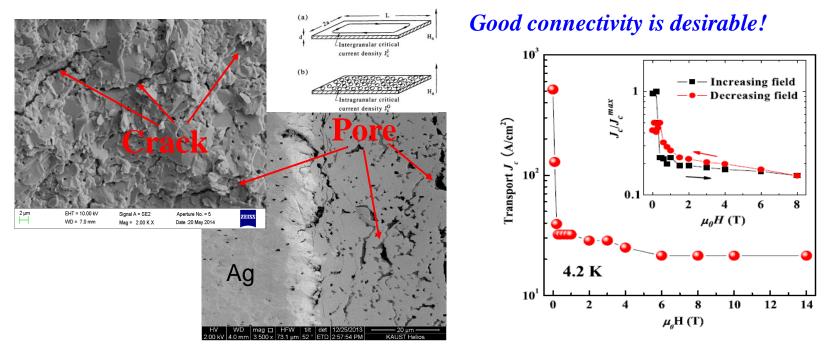
# The J<sub>c</sub> of IBS wires: Very weak field dependence in high-field region





- **1** Background on Fe-based superconductor
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### **Key problems for PIT wires: Impurity, Low density and** Weak-link behavior

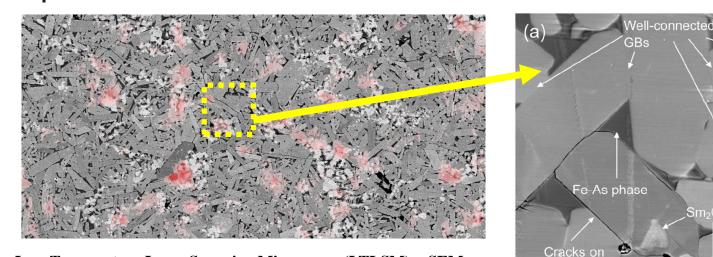


Low density: cracks and porosity

Hysteresis in transport  $J_c$ : signature of weak links

- Impurity and low density (porosity) always lead to poor grain connection, so suppress J<sub>c</sub> in polycrystalline wires!
- A hysteretic phenomenon observed for transport  $J_c$  in an increasing and a decreasing field indicated a **weak-linked behavior**, similar to that of the cuprates.

**Q1:** Early efforts at wire development suffered from many impurity phases (such as Fe-As) that wet the grain boundaries, largely decreasing Jc



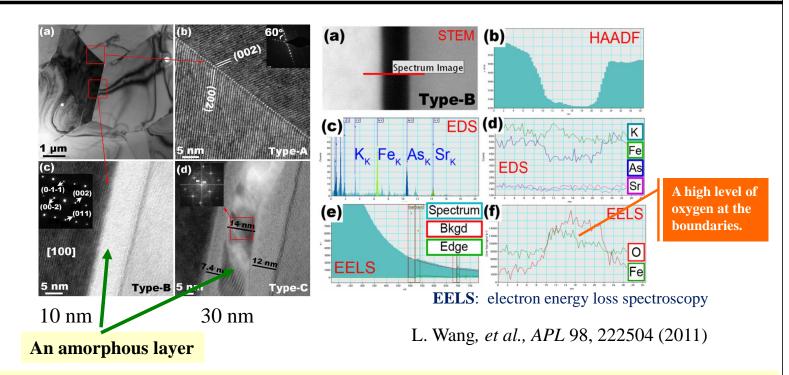
Low Temperature Laser Scanning Microscopy (LTLSM) + SEM

Kametani et al., APL. 95, 142502 (2009)

GBs

Dissipation is clearly localized in impurity-rich regions.
Fe-As phase covers the grains causing a current blocking effect.

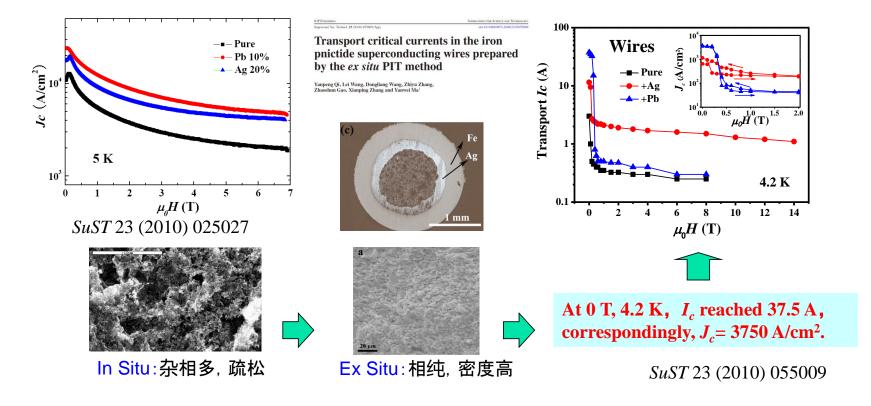
# **TEM-EELS studies:** Grain boundaries in the Sr122 polycrystals are usually coated by impurity amorphous layers (10-30 nm), which show significant oxygen enrichment



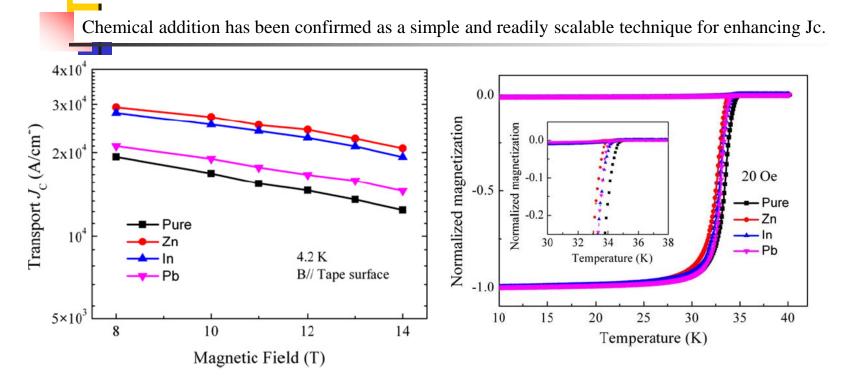
- Obviously, these O-rich amorphous layers are strongly related to the introduction of O<sub>2</sub> during fabrication.
- These oxygen-rich layers undoubtedly obstructed many grain boundaries, consequently resulting in a poor grain connection.

# **Solutions:** *Ex-situ* + *Addition* PIT method → removed the impurity phases in Ag-cladded 122 wires

- **Ex-situ PIT method:** fewer impurity phases as well as a high density of the superconducting core for the final wires.
- Ag or Pb addition to improve the grain connectivity, hence the enhanced  $J_c$ .



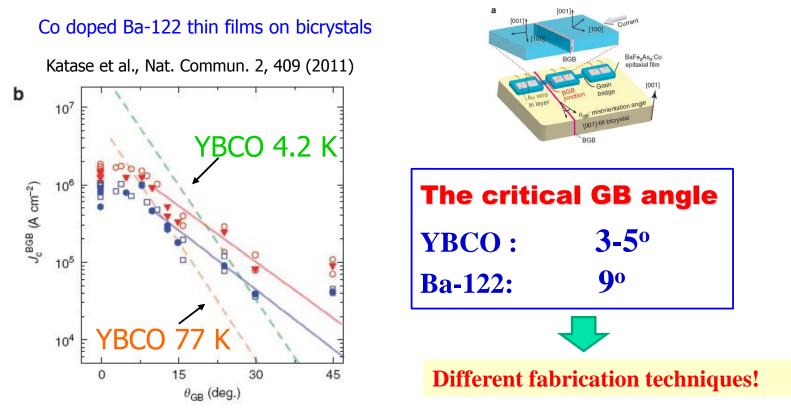
## Zn and In additions are effective to enhance the $J_c$ -B of 122/Ag tapes



- The additions do not significantly affect the temperature transition Tc, and the Tc decreased only 0.4 K.
- the Jc enhancement in In or Zn-added samples may be attributed to the improved phase uniformity as well as the good grain connectivity

Lin et al., Scripta Mater. 112 (2016) 128

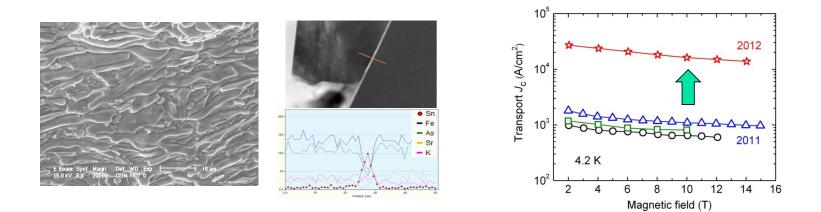
#### **Q2: Weak-link problem -** Intrinsic nature of dissipation



- **⊃** J<sub>c</sub> decreases exponentially with GB angle .
- **\bigcirc** Weak link effect, the GBs do not degrade the  $J_c$  as heavily as YBCO.
- Advantageous GB over cuprates! This is the reason why we can use the PIT method to make the pnictide wire and tapes, but PIT can not work for YBCO.

## **Texturing process of PIT Sr122 tapes**

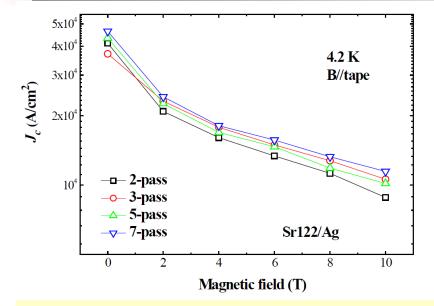
 One effective method is to engineering textured grains to minimize deterioration of Jc across high-angle grain boundaries, like the Bi2223.

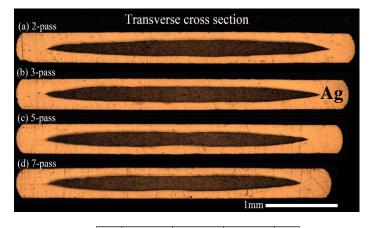


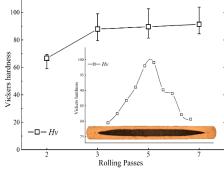
The above rolling strongly improved c-axis texturing, and effectively reduce the large angle GBs, thus Jc was enhanced by **an order of magnitude, from 10^3 to**  $10^4$  A/cm<sup>2</sup> °

Physica C 471 (2011) 1689; Sci. Rep. 2 (2012) 998

## **Optimized rolling process for** Sr<sub>0.6</sub>K<sub>0.4</sub>Fe<sub>2</sub>As<sub>2</sub> /Ag tapes: 3-pass deformation is best



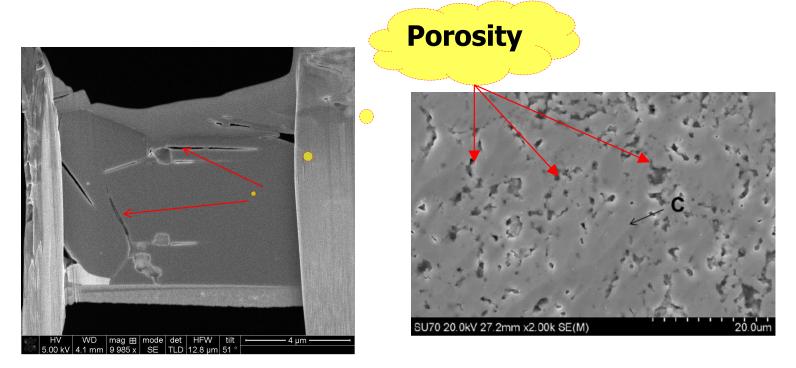




- The width of the tapes and the area of superconducting cores increase with decreasing the rolling pass, but the transport J<sub>c</sub> seems close.
- We can fabricate tapes with 3 rolling passes to get the uniform and high- $J_c$  122 tapes.

*Physica C* 525 (2016) 94

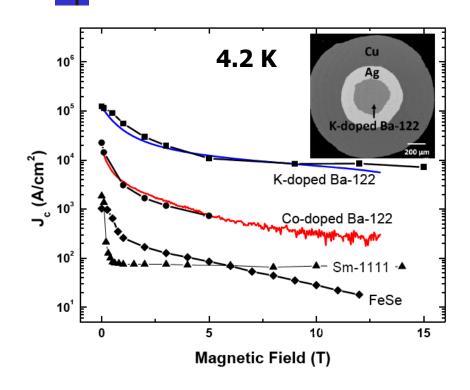
# **Q3:** Densification is another dominant factor that determines the $J_c$ of PIT pnictide wire



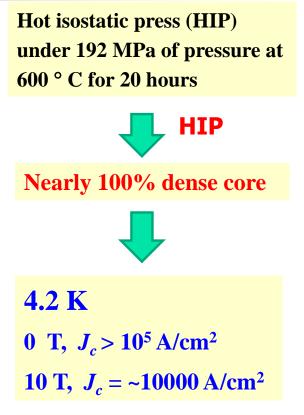
**Cracks and voids are one of the important reasons for low critical current density values** 

#### NHMFL

## New synthesis method (HIP) increased $J_c$ in Ba-122 round wire

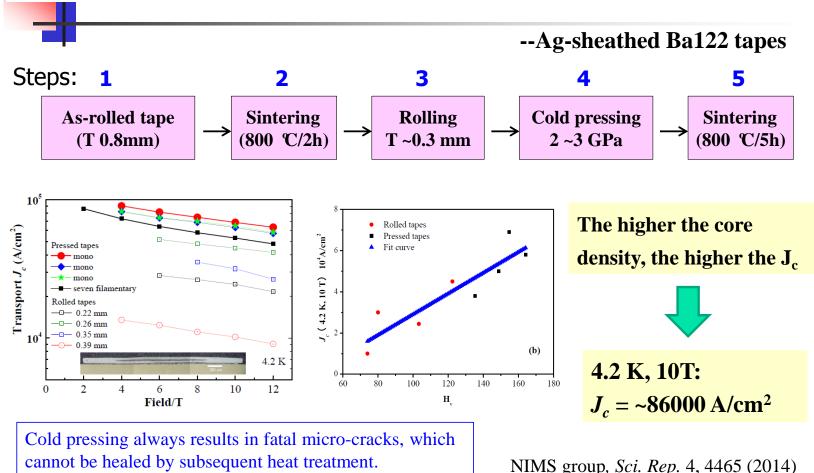


J. Weiss et al., Nature Mater. 11, 682 (2012)



#### NIMS

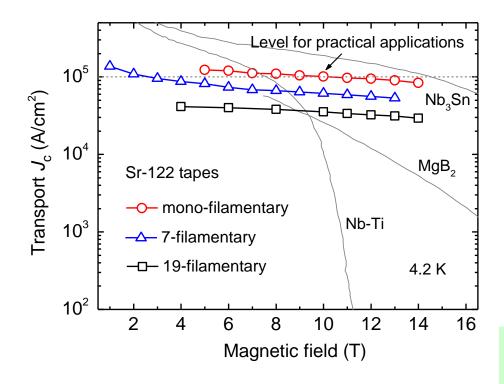
# Thin tapes by combined the rolling, cold pressing and sintering process-- Denser core yields higher $J_c$



#### **Breakthrough work** --Sr<sub>0.6</sub>K<sub>0.4</sub>Fe<sub>2</sub>As<sub>2</sub>

Very high transport  $J_c$  were achieved in Sr122/Ag tapes:  $J_c > 10^5 \text{ A/cm}^2 (4.2 \text{ K}, 10 \text{ T})$  - by hot pressing

First to reach practical level  $J_c$ !



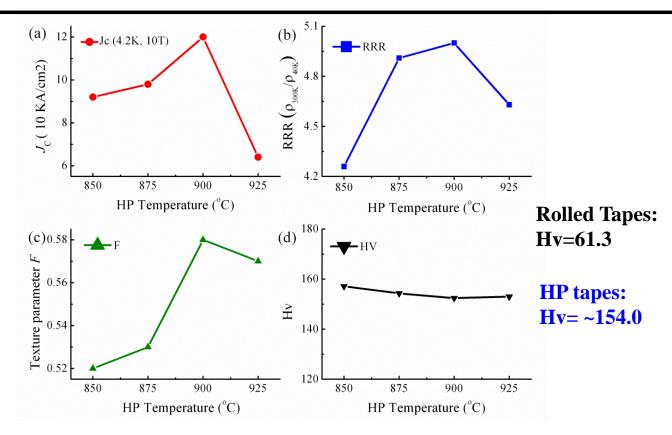
The threshold for practical application:  $J_c=10^5 \text{ A/cm}^2@10 \text{ T}$ 

Later achieved At 10 T,  $J_c = 1.2 \times 10^5 \text{ A/cm}^2$ even in 14 T,  $J_c = \sim 10^5 \text{ A/cm}^2$ 

The superior  $J_c$  can be attributed to higher grain texture and improved densification.

Zhang et al., *APL* 104 (2014) 202601 Lin et al., *Sci. Rep.* 4 (2014) 6944

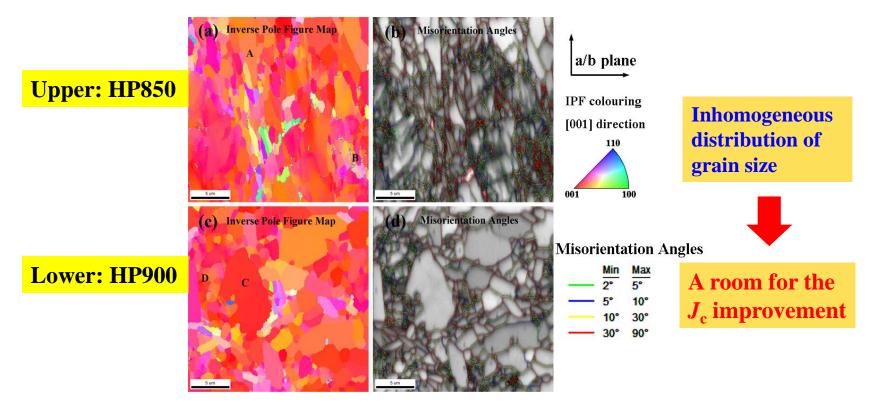
#### **Reasons for high transport** $J_c$ in HP900 tapes



- **\bigcirc** The variation tendency of  $J_c$  values was qualitatively similar to those of F and RRR values.
- **The hardness was almost saturated as soon as the hot pressing was applied.**
- ⇒ The J<sub>c</sub> increase for HP900 tapes was mainly attributed to enhanced grain connectivity (high density) and higher degree of c-axis texture.

## **EBSD images: HP samples are highly textured**

A useful tool to clarify the grain size, local orientation of the grains and misorientation angles between grains.

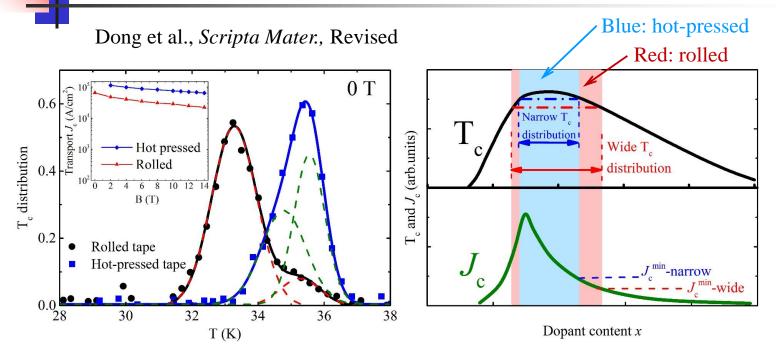


- ➡ The large fraction of small misorientation angles between 2–10°C (HP850 tapes 23.3%, HP900 tapes 26.2%).
- A large proportion of misorientation angles of the GBs were smaller than 9° such that the transport Jc would not be greatly reduced upon encountering these GBs.

Latest result --Ba<sub>0.6</sub>K<sub>0.4</sub>Fe<sub>2</sub>As<sub>2</sub>/Ag tapes New record transport  $J_c$  up to  $1.5 \times 10^5$  A/cm<sup>2</sup> @ 4.2 K, 10 T was obtained by Hot Pressing At 18 T,  $J_c = \sim 10^5 \,\text{A/cm}^2$  $I_c = 437 \text{ A}$ I // Tape surface 10<sup>5</sup> At 27 T, Fransport J<sub>c</sub> (A/cm<sup>2</sup>)  $J_c = 5.5 \times 10^4 \,\text{A/cm}^2$  $10^{4}$ **Small anisotropy** Nb<sub>3</sub>Sn NbTi MgB<sub>2</sub> Fransport  $J_{c}$  (A/cm<sup>2</sup>)  $10^{3}$ 10 4.2 K Batch 1 Batch 2 At 10 T  $10^{2}$ 8 12 16 20 24 28 Angle (°) 10 2 10 12 14 0 4 6 8 Magnetic Field (T) Magnetic Field (T) **2MO3-05** Huang et al., arXiv:1705.09788

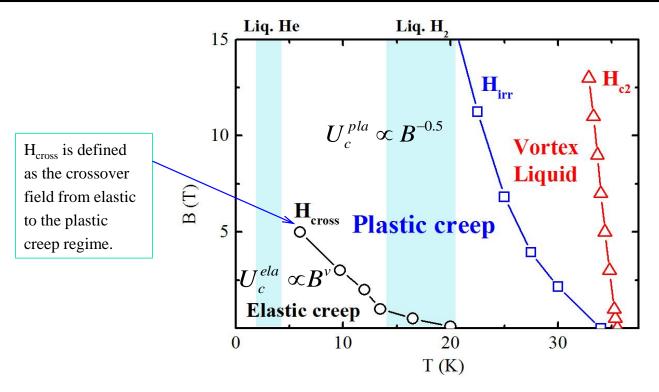
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# **Specific heat measurement: Calorimetric evidence for enhancement of homogeneity**



- The Schottky anomaly that is obvious in the specific heat of the rolled Sr122 tape disappears in the hot-pressed Sr122 tape.
- the hot-pressed tape has a higher fraction of superconductivity and a narrower distribution of superconducting transition temperature than the rolled tape.

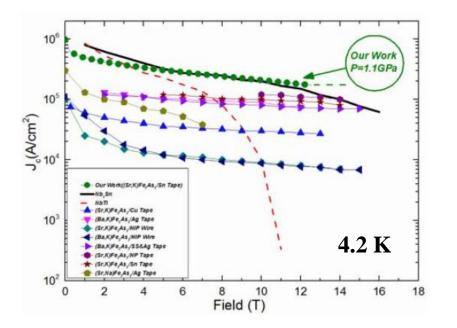
### Vortex phase diagram of high-J<sub>c</sub> HP-122 tapes



- $\square$  More robust field dependence of  $J_c$  in the elastic creep regime.
- $\square$  Weak field dependence of  $J_c$  in the liquid helium region, but  $J_c$  quickly decrease in the liquid hydrogen region.
- **To further increase flux pinning force: increase point pinning sites,** *e.g.* irradiation or nano-particle inclusion.

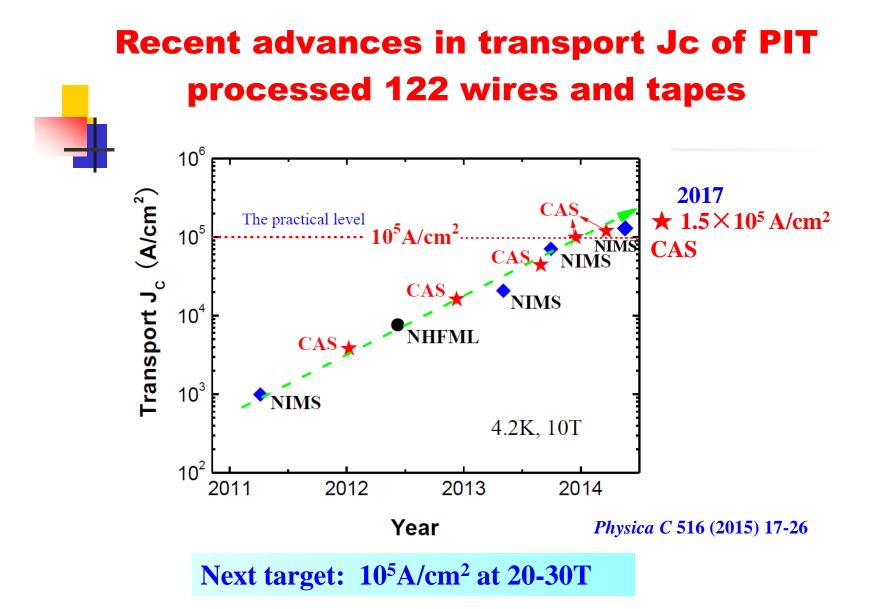
Dong et al., JAP 119 (2016) 143906

## Magnetic $J_c$ up to $3 \times 10^5$ A/cm<sup>2</sup> @ 4.2 K, 10 T can be achieved under Hydrostatic Pressure



- -- Collaborated with Prof. Xiaolin Wang, S. X. Dou, Wollongong Univ.
- ✓ Using PPMS, HMD high pressure cell and Daphne
   7373 oil as the medium for applying hydrostatic pressure on Sr-122/Ag tape samples.
- ✓ Tape samples were measured under pressure.
- The hydrostatic pressure of 1GPa can significantly enhance  $J_c$  in Ag-clad Sr<sub>0.6</sub>K<sub>0.4</sub>Fe<sub>2</sub>As<sub>2</sub> tapes at different temperatures, e.g., ~2×10<sup>5</sup>A/cm<sup>2</sup> at 13T, 4.2 K.
- Pressure can improve the grain connectivity and increase the pinning number density.
- The result shows that the current IBS tapes/wires should have plenty of room to raise their  $J_c$  or  $I_c$  to higher levels.

Shabbir et al., Phys. Rev. Mater., 2017

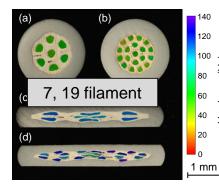


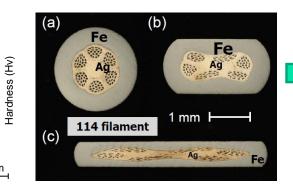
## **Outlines**

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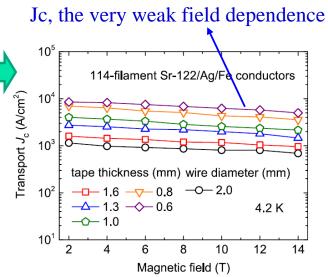
## Fabrication of 114-filament Sr-122/Ag/Fe wires by the drawing and rolling

> The fabrication of multifilamentary wires and tapes is an indispensable step, to reduce the AC loss and avoid the flux jump.





- 114-core round wires:  $J_c = 800 \text{ A/cm}^2$ .
- When they are flat rolled into tapes, the  $J_c$  grows with the reduction of tape thickness. the  $J_c = 6.3 \times 10^3 \,\text{A cm}^{-2}$  in 0.6 mm thick tapes.
- 7-core tapes:  $J_c = 1.5 \times 10^4 \text{ A/cm}^2$ .
- This  $J_c$  degradation can be ascribed to the sausage effect.

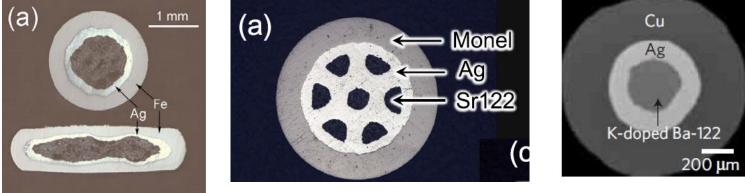




Yao et al., JAP 118 (2015) 203909

## **Fabrication of high-strength IBS wires**

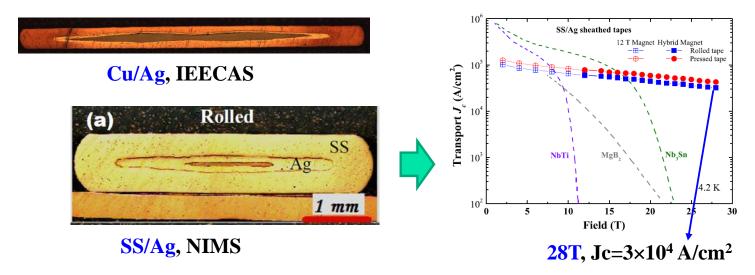
Ag may be also used in combination with an additional outer sheath made of Fe, Cu, and stainless steel to **reduce costs and improve the mechanical strength.** 



Fe/Ag, IEECAS

Monel/Ag, IEECAS

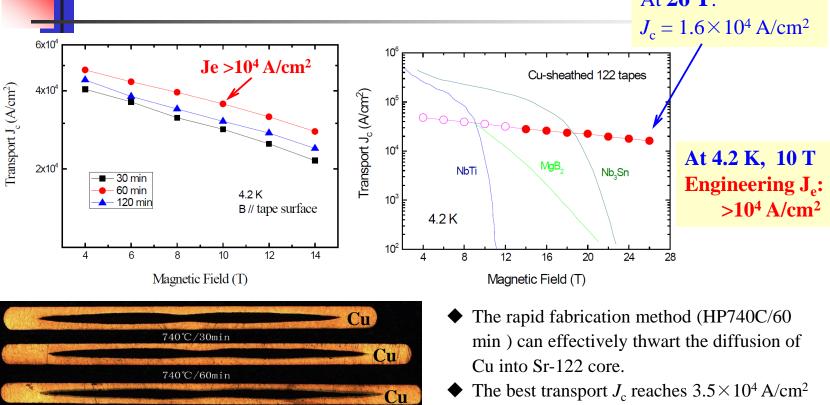
Cu/Ag, Florida



#### **Copper sheath material**

740°C/120min

# High J<sub>c</sub> in Cu-sheathed Sr-122 tapes at low temperature 740°C



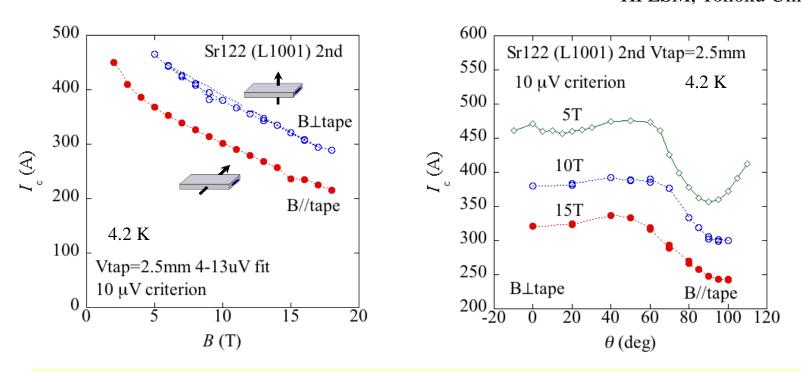
at 10 T and keeps  $1.6 \times 10^4$  A/cm<sup>2</sup> at 26 T.

Lin et al., *SuST* 29 (2016) 095006

**Small anisotropy** 

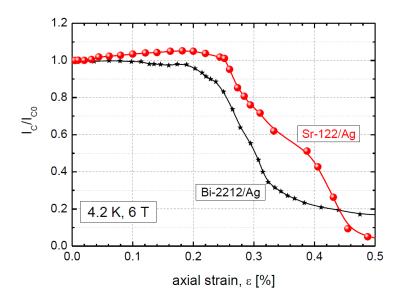
Awaji et al., SuST 30 (2017) 035018

#### J<sub>c</sub> properties at 4.2 K for HP Sr-122/Ag tapes -- Measured by Prof. Awaji HFLSM, Tohoku Univ.



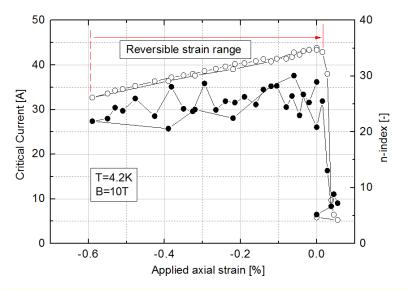
- The  $I_c$  in applied magnetic fields is slightly higher in the perpendicular field  $(I_c^{\perp})$  than in the parallel field  $(I_c^{\prime\prime})$ .
- The anisotropy ratio ( $\Gamma = I_c^{\perp}/I_c^{\prime\prime}$ ) is less than 1.5, quite small, very promising for applications.

## Sr122/Ag tape – Strain-stress properties



#### The first strain measurements

Reversible critical currents under a large compressive strain of  $\varepsilon = -0.6$  %



At 4.2 K, 10 T:  $I_c > 125A$ Irreversible strains:  $\varepsilon = 0.25\%$ 

Comparable to Bi2212

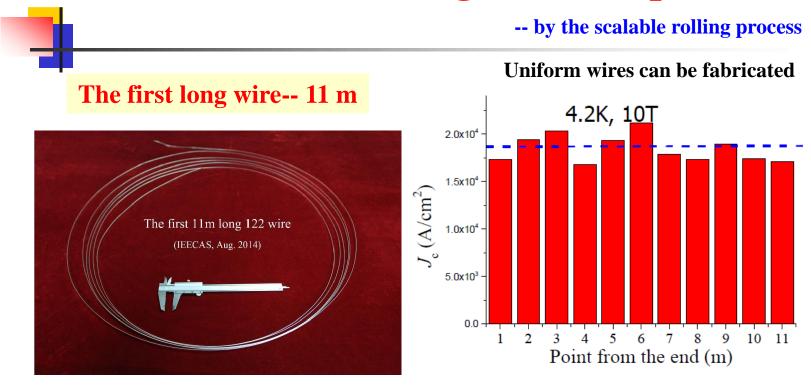
Kovac et al., SuST 28 (2015) 035007

The critical current of Sr-122 tape exhibits less strain sensitivity than that of the Nb<sub>3</sub>Sn, which is an important for ITER application.

Liu, SuST 30 (2017) 07LT01

Ma, Physica C 516 (2015) 17

### The first 11m long Sr-122 tape



The minimum  $J_{\rm c}$  ~1.7 × 10<sup>4</sup>A/cm<sup>2</sup>

The average  $J_c$  of this long Sr122/Ag wire is ~ 18400A/cm<sup>2</sup> The fluctuations of the  $J_c$  is ~5% Significant breakthrough!

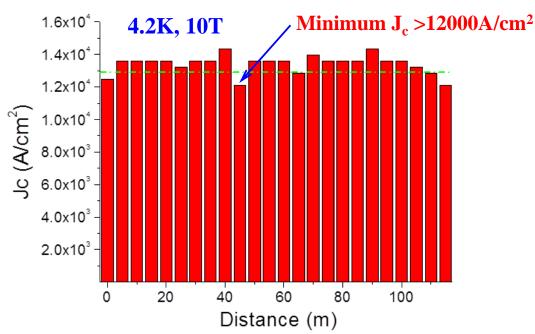
In Aug., 2016

### The world's first 100 meter-class iron-based superconducting wire



115 m long **7-filamentary** wire





At 4.2 K, 10 T, transport Jc distribution along the length of the first 100 m long 7-filament Sr122 tape

http://snf.ieeecsc.org/pages/new-paper-and-result-highlights

X. P. Zhang et al., IEEE TAS 27 (2017) 7300705

-- Presented at ASC2016, Denver

## Conclusions

- Currently, Fe-based wires and tapes are in the rapid development stage of research and development.
- ✓ The transport J<sub>c</sub> values are already extremely high, maximum J<sub>c</sub>=1.5×10<sup>5</sup> A/cm<sup>2</sup> at 10 T and 4.2 K, surpassing the widely accepted threshold for applications.
- ✓ In particular, the world's first 100-m-class 122-type Febased wire was achieved, which demonstrates the great potential for large-scale manufacture.
- ✓ Further improvement in J<sub>c</sub> can be expected upon either introducing more pinning centers, or enhancing grain texturing.
- ✓ We believe that Fe-based wires are very promising for applications using high magnetic fields, e.g. >20 T at 4.2 K or >10 T at 20-30 K.

#### **Contributors:**

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**Institute of Electrical Engineering, CAS** 

### **Collaborators:**

S. Awaji (Ic-B measurement) Institute for Materials Research, Tohoku University, Japan

Hai-hu Wen (Pinning properties) Nanjing University, China

Jianqi Li (HRTEM) Institute of Physics, Chinese Academy of Sciences

Xiaolin Wang, S. X. Dou (Pinning and hydrostatic studies) Wollongong University, Australia

W. K. Kwok, U. Welp (Irradiation) Argonne National Laboratory, USA

**T. Kiss** (Characterizing local microstructure and homogeneity of wires) **Kyushu University, Japan** 



# Thank you for your attention